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# IDENTIFICATION OF ADSORBATE FT-IR BANDS USING IN-SITU TECHNIQUES: Pd SPECIATION AND ADSORPTION CHEMISTRY OF Pd-ZEOLITES FOR PASSIVE NOx ADSORPTION

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> Robert Bruce Pace III, Student Dr. Mark Crocker, Major Professor Dr. Yinan Wei, Director of Graduate Studies

## IDENTIFICATION OF ADSORBATE FT-IR BANDS USING IN-SITU TECHNIQUES: Pd SPECIATION AND ADSORPTION CHEMISTRY OF Pd-ZEOLITES FOR PASSIVE NOX ADSORPTION

## DISSERTATION \_

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A dissertation submitted in partial fulfillment of the requirements for the degree of Doctor of Philosophy in the College of Arts and Sciences at the University of Kentucky

By

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Lexington, Kentucky

Co- Directors: Dr. Mark Crocker, Professor of Chemistry

and Dr. John P. Selegue, Professor of Chemistry

Lexington, Kentucky

2020

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#### ABSTRACT OF DISSERTATION

## IDENTIFICATION OF ADSORBATE FT-IR BANDS USING IN-SITU TECHNIQUES: Pd SPECIATION AND ADSORPTION CHEMISTRY OF Pd-ZEOLITES FOR PASSIVE NOX ADSORPTION

To meet increasingly stringent automotive emissions standards, further improvements in catalytic converter design are necessary. Current automotive catalyst systems are effective at eliminating emission of nitrogen oxides  $(NO<sub>x</sub>)$  once the catalyst reaches operational temperature ( $\sim$ 200 °C). NO<sub>x</sub> emitted at lower catalyst temperatures now comprises most of the  $NO<sub>x</sub>$  released during a typical test cycle. Referred to as "the cold start problem" this issue has come to the forefront of automotive catalyst development, as mitigating these emissions is necessary to further reduce automotive emissions. Passive  $NO<sub>x</sub>$  adsorbers present an appealing solution to the cold start problem, these being a class of materials that chemisorb exhaust components such as  $NO<sub>x</sub>$ , carbon monoxide (CO) and hydrocarbons at near-ambient temperatures, and then desorb these compounds once the downstream catalyst has reached operational temperature. An effective passive  $NO<sub>x</sub>$  adsorber must have several properties: high  $NO<sub>x</sub>$  adsorption at near-ambient temperatures, near-complete  $NO<sub>x</sub>$  desorption at temperatures within the operational range, high thermal stability, and resistance to automotive exhaust components at high temperatures. The potential environmental impact of such a system is substantial, as  $NO<sub>x</sub>$  emissions currently result in the formation of millions of tons of smog and acid rain each year.

Pd-exchanged zeolites have shown promise for deployment as Passive  $NO<sub>x</sub>$ adsorbers, though much remains to be understood about their adsorption chemistry and deactivation. In-situ Diffuse Reflectance Infrared Fourier Transform Spectroscopy (DRIFTS) provides a convenient probe of adsorbed species, most automotive exhaust components possessing IR-active chemical bonds. By examining the evolution of IR bands under various pretreatments and adsorbates, the overall Pd-speciation and adsorptive zeolite sites of each material can be characterized, and the identities of IR bands can be deduced. In this work, microreactor-MS analysis of the adsorption and desorption behavior of these materials was also examined, these results being coupled with in-situ DRIFTS temperature programmed desorption (TPD) to correlate desorption events with specific adsorbed species.

A pair of zeolite frameworks of similar Si/Al ratio but differing pore size were examined, Beta zeolite (BEA) and Chabazite (CHA) representing a medium- and smallpore framework, respectively. The effect of Pd-loading on BEA was examined, as well as the various deactivation pathways and active sites of each material.

KEYWORDS: Passive NOx Adsorber, Automotive Catalysis, Environmental Catalysis, Palladium, Zeolite, DRIFTS

Robert Bruce Pace III

11/16/2020

Date

# IDENTIFICATION OF ADSORBATE FT-IR BANDS USING IN-SITU TECHNIQUES: Pd SPECIATION AND ADSORPTION CHEMISTRY OF Pd-ZEOLITES FOR PASSIVE NO<sub>X</sub> ADSORPTION

By Robert Bruce Pace III

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Yinan Wei Director of Graduate Studies

11/16/2020

Date

DEDICATION

To my wife Stephanie, without whom none of this would have been possible.

#### ACKNOWLEDGMENTS

<span id="page-7-0"></span>This dissertation represents an individual work, though without the contributions of others it would never have come about. First and foremost, I would like to thank my committee chair, Dr. Mark Crocker for his insight, wisdom, and patience in guiding me through my time as a graduate student. I would also like to thank my co-chair Dr. John Selegue for his unerring ability to always find that one reference that helps my data make sense. I would also like to thank Dr. Marcelo Guzman and Dr. Yang-Tse Cheng for their insight and commentary that has helped keep me on course to graduation.

I would also like to thank Mr. Trevor Lardinois and Dr. Raj Gounder for their significant intellectual and experimental contributions to this work, their efforts in material preparation and characterization serve as the foundation for the work presented here. I would also like to thank Dr. Yaying Ji for her contributions to this work, her initial studies have been invaluable for the interpretation of data collected in the later stages of this effort. This project involved many collaborations, and I would specifically like to thank Prof. Alexis T. Bell, Dr. Jeroen Van der Mynsbrugge, Dr. Andrew "Bean" Getsoian, Dr. Joe Theis, Dr. Christine Lambert, and Dr. Olivier Heinz who all contributed to this work either intellectually or through experimental data. I would also like to thank Dr. Eduardo Santillan-Jimenez for his advice and guidance. Finally, I would like to thank the U.S. Department of Energy for funding this work.

# TABLE OF CONTENTS



<span id="page-9-0"></span>

# LIST OF TABLES



# LIST OF FIGURES

<span id="page-11-0"></span>

Figure 2.11 DRIFT spectra of CO adsorption onto 0.7% Pd-CHA after pretreatment at 500 °C for 1 h with Ar wetted by bubbler, untreated, or rigorously dried by cold trap. The material was cooled in dried Ar to 25  $\degree$ C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C. Top: Ionic Pd range. Bottom: metallic Pd range. ............. 45 Figure 2.12 0.6% Pd-CHA: DRIFT spectra of CO adsorption after sequential pretreatment at 500 °C for 1 h with dried air, air wetted by bubbler, and finally dried air. Only the as-prepared and re-oxidized spectra are shown for clarity. The material was cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C between each pretreatment. Top: ionic Pd range. Bottom: metallic Pd range. ... 47 Figure 2.13 DRIFT spectra of 1.4% Pd-BEA pretreated at 500  $\degree$ C for 1 h in Ar then cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min followed by adsorption of 1-2% water in Ar wetted by bubbler. Top: Ionic Pd range. Bottom: metallic Pd range. ... 50 Figure 2.14 DRIFT spectra of CO adsorption onto 1.4% Pd-BEA for 10 min after water adsorption for 10 min. The material was pretreated at 500 °C for 1 h in Ar then cooled in dried Ar to 25 °C prior to adsorption. 1-2% water in Ar wetted by bubbler was adsorbed for 10 min followed by 1000 ppm CO in Ar. Top: Ionic Pd range. Bottom: metallic Pd range.. 51 Figure 3.1 DRIFT spectra of CO adsorption onto 1.4% Pd-BEA after sequential pretreatment at 500 °C for 1 h with dried air, 1000 ppm CO in Ar, and finally dried air. The material was cooled in dried Ar to 25  $^{\circ}$ C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C between each pretreatment. Top: ionic Pd range. Bottom: metallic Pd range. ... 55 Figure 3.2 DRIFT spectra of CO adsorption onto 0.7% Pd-CHA after sequential pretreatment at 500 °C for 1 h with dried air, 1000 ppm CO in Ar, and finally dried air. The material was cooled in dried Ar to 25  $\degree$ C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C between each pretreatment. Top: ionic Pd range. Bottom: metallic Pd range. ... 57 Figure 3.3 DRIFT spectra of CO adsorption onto 1.4% Pd-BEA after sequential pretreatment at 500 °C for 1 h with dried air,  $10\%$  H<sub>2</sub> in Ar, and finally dried air. The material was cooled in dried Ar to 25  $^{\circ}$ C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C between each pretreatment. Top: ionic Pd range. Bottom: metallic Pd range... 59 Figure 3.4 DRIFT spectra of CO adsorption onto 0.7% Pd-CHA after sequential pretreatment at 500 °C for 1 h with dried air,  $10\%$  H<sub>2</sub> in Ar, and finally dried air. The material was cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C between each pretreatment. Top: ionic Pd range. Bottom: metallic Pd range... 61 Figure 3.5 1.4% Pd-BEA (top),  $0.7\%$  Pd-CHA (bottom). NO<sub>x</sub> TPD conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$  concentrations. Both materials were pretreated in 10%  $O_2$  in He, 10%  $H_2$  in He, and finally 10%  $O_2$  in He. 1000 ppm NO in 10%  $O_2$ /He was adsorbed for 10 min at 50 °C followed by a 1 h purge in  $10\%O_2$ /He. TPD was carried out in 10% O2/He at a ramp rate of 10 °C per min. .. 64



1000 ppm NO in He,  $10\%$  O<sub>2</sub>/He, or 1-2% water  $10\%$  O<sub>2</sub>/He was adsorbed for 10 min at 50 °C followed by a 1 h purge in  $10\%O_2$ /He. TPD was carried out in  $10\%$  O<sub>2</sub>/He at a ramp rate of 10 °C per min. Tabulated values were calculated from integrated  $NO<sub>x</sub> MS$ signals. .. 88 Figure 4.14 NOx TPD conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$  concentrations Right: total  $NO<sub>x</sub>$  adsorbed on H and Pd-BEA, left: deconvolution of  $NO<sub>x</sub>$  storage events on H-BEA. Materials were pretreated in  $10\% O_2$  in He for 1 h prior to adsorption. 1000 ppm NO in 10% O<sub>2</sub>/He was adsorbed for 10 min at 30, 50, or 100 °C followed by a 1 h purge in 10%O<sub>2</sub>/He. TPD was carried out in 10% O<sub>2</sub>/He at a ramp rate of 10 °C per min. Tabulated values were calculated from integrated NOx MS signals................................. 89 Figure 4.15  $NO<sub>x</sub>$  TPD conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$  concentrations. Materials were pretreated in 10%  $O_2$  in He for 1 h prior to adsorption. 1000 ppm NO in 10% O<sub>2</sub>/He was adsorbed for 10 min at 50 °C followed by a 1 h purge in  $10\%$ O<sub>2</sub>/He. TPD was carried out in 10% O<sub>2</sub>/He at a ramp rate of 10 °C per min. Tabulated values were calculated from integrated NOx MS signals... 90 Figure 5.1 NOx TPD conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$  concentrations. Materials were pretreated in  $10\%$  O<sub>2</sub> in He or He for 1 h prior to adsorption. 1000 ppm NO in 10% O<sub>2</sub>/He or He was adsorbed for 10 min at 50  $\degree$ C followed by a 1 h purge in 10% O<sub>2</sub>/He or He. TPD was carried out in 10% O<sub>2</sub>/He or He at a ramp rate of 10 °C per min. O2 was either included (top) or excluded (bottom) during all steps......................... 94 Figure 5.2  $NO<sub>x</sub>$  TPD of 1.4% Pd-BEA conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$  concentrations. Materials were pretreated in 10%  $O<sub>2</sub>$  in He or He for 1 h prior to adsorption. 1000 ppm NO in 10% O<sub>2</sub>/He or He was adsorbed for 10 min at 50 °C followed by a 1 h purge in 10%  $O_2$ /He or He. TPD was carried out in 10%  $O_2$ /He or He at a ramp rate of 10 °C per min... 96 Figure 5.3 H-BEA (top) 0.1% Pd-BEA (second) 0.6% Pd-BEA (third) and 1.4% Pd-BEA (bottom): DRIFTS spectra of NO<sub>2</sub> desorption after pretreatment at 500  $^{\circ}$ C for 1 h with rigorously dried Ar. NO<sub>2</sub> was adsorbed for 10 minutes at 50  $\degree$ C before the material was purged for 10 minutes in Ar prior to ramping to remove gas phase  $NO<sub>2</sub>$ . Temperature was ramped at 10 °C/min to 500 °C with spectra being collected every 20 °C, highlighted spectra represent 50 °C (orange spectrum), and 100 °C (dark blue), 200 °C (yellow), 300 °C (light blue), 400 °C (navy), and 500 °C (black). .. 100 Figure 5.4 1.4% Pd-BEA: DRIFTS spectra of NO desorption after pretreatment at 500 °C for 1 h with rigorously dried Ar. NO was adsorbed for 10 minutes at 25 °C before the material was purged for 10 minutes in Ar prior to ramping to remove gas phase NO. Temperature was ramped at 10  $\degree$ C/min to 500  $\degree$ C with spectra being collected every 20 °C, highlighted spectra represent 25 °C (orange spectrum), and 100 °C (dark blue), 200 °C (yellow), 300 °C (light blue), 400 °C (navy), and 500 °C (black). ............................ 102 Figure  $5.5$  NO<sub>x</sub> TPD on CHA materials conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$  concentrations. Materials were pretreated in 10%  $O<sub>2</sub>$  in He or He for 1 h prior to adsorption. 1000 ppm NO in 10% O<sub>2</sub>/He or He was adsorbed for 10 min at 50 °C followed by a 1 h purge in 10%  $O_2$ /He or He. TPD was carried out in 10%  $O_2$ /He or He at a ramp rate of 10 °C per min. O<sub>2</sub> was either included (top) or excluded (bottom) during all steps.. 105



#### <span id="page-16-0"></span>CHAPTER 1. INTRODUCTION

#### <span id="page-16-1"></span>1.1 Automotive Catalytic Conversion Systems

Internal combustion engines have facilitated the rapid development of the industrialized world for over a century, though only now are the full implications and negative impact of this technology being adequately understood. Combustion of hydrocarbon fuels leads to the formation of a distribution of undesirable byproducts, including particulate matter, partially combusted hydrocarbons, carbon monoxide (CO), carbon dioxide (CO<sub>2</sub>), an assortment of nitrogen oxides (NO<sub>x</sub>), along with other byproducts such as hydrogen  $(H_2)$  and sulfur compounds. Some of these species are especially problematic; annually millions of tons of  $NO<sub>x</sub>$  is emitted and these species react with ozone and hydrocarbons, and with water, to generate photochemical smog  $(NO<sub>2</sub>)$  and acid rain  $(HNO<sub>3</sub>)$ , respectively.<sup>1</sup> To reduce the emission of these species, automotive catalytic converters were developed to convert  $NO<sub>x</sub>$ , CO, and hydrocarbons to more benign  $CO<sub>2</sub>$ , water, and  $N<sub>2</sub>$  under the necessary operational conditions of the vehicular exhaust stream. Since a wide variety of engine types and fuels exists, a wide range of catalytic converter technologies have also been developed to meet the specific needs of each vehicle and fuel type. However, factors in automotive development such as greater fuel-efficiency, and thus lower operating exhaust temperatures, are driving these technologies to their functional limits.<sup>2, 3</sup>

One of the most important aspects of catalytic converter design is the air-to-fuel ratio the intended engine system requires, three typical ranges being established: a reducing or rich regime where oxygen is the limiting reactant, an oxidizing or lean regime where CO and hydrocarbons are the limiting reagent, and finally a stoichiometric regime in which the quantity of oxygen is stoichiometrically adequate to convert all of the available CO and hydrocarbons to  $CO<sub>2</sub>$ . Each of these conditions has advantages, and some catalyst technologies even require various stoichiometries to provide for catalyst regeneration. Generally speaking, two strategies have been employed in both engine development and  $NO<sub>x</sub>$  reduction strategy, one being the lean-burn diesel engine coupled with a diesel oxidation catalyst (DOC) followed by lean  $NO<sub>x</sub>$ -trap (LNT) and or selective

catalytic reduction (SCR) systems, in addition to stoichiometric gasoline engines that employ a three-way catalyst (TWC) to reduce  $NO<sub>x</sub>$ . Stoichiometric gasoline engines represent the bulk of small, general-use vehicles on the road today.

<span id="page-17-0"></span>1.1.1 Modern Technologies

#### 1.1.1.1 LNT-SCR systems

<span id="page-17-1"></span>Lean-burn diesel vehicles employ a series of catalytic conversion technologies, each performing a different function in the exhaust treatment process. First, a diesel oxidation catalyst (DOC) uses supported platinum group metals to oxidize CO, hydrocarbons and particulate matter to  $CO<sub>2</sub>$ , while also having the disadvantageous side effect of oxidizing most NO to  $NO<sub>2</sub>$ .<sup>4-6</sup> Since the oxidation catalyst eliminates any reducing species from the exhaust stream, the problem of  $NO<sub>x</sub>$  reduction can be approached in two ways; either by adding a reductant to the gas stream to be used in the reduction of  $NO_x$ , or by storing  $NO_x$  to be reduced under a deviation to rich conditions. The former method represents the process implemented in a SCR system, while the latter describes a LNT system. 7, 8

SCR systems are employed primarily on larger vehicles due to the necessity of carrying an additional quantity of reductant which is introduced to the exhaust from an external tank. Commonly called diesel exhaust fluid (DEF), this reductant is either aqueous ammonia or urea.<sup>8</sup> Three reactions (Equations 1-3) are involved in the SCR process, the first being the most prevalent:

$$
4NH_3 + 4NO + O_2 \rightarrow 4N_2 + 6H_2O \tag{1}
$$

$$
2NH_3 + NO + NO_2 \to 2N_2 + 3H_2O \ (fast)
$$
 (2)

$$
8NH_3 + 6NO_2 \to 7N_2 + 12H_2O \ (very slow)
$$
 (3)

Depending on the application, various catalysts have been employed in this process. Stationary lean-burn engines have typically been outfitted with catalysts consisting of vanadium oxide supported on titania with a tungsten oxide promoter.<sup>9</sup> While this catalyst achieves excellent  $NO<sub>x</sub>$  reduction at intermediate temperatures, at

higher temperatures it can lead to the oxidation of ammonia, formation of  $SO<sub>3</sub>$ , as well as  $N_2O$ .<sup>10</sup> Conversely, at lower temperatures this system has insufficient  $NO<sub>x</sub>$  conversion to meet modern emission standards. While conversion can be improved by increasing vanadium loading, this also yields a corresponding decrease in selectivity to  $N_2$ .<sup>8</sup> Other metal oxides have been investigated for their activity in this process, as  $NO<sub>x</sub>$  reduction at lower temperatures would also be advantageous for stationary internal combustion systems.<sup>11</sup> Manganese oxide materials have received special attention, though their usefulness is limited by the heavily phase-dependent nature of their activity.<sup>11, 12</sup> The introduction of various metal oxides as dopants served to remediate some of these issues; iron, copper and ceria all stabilize the active phase of the catalyst, and thus improve the low-temperature  $NO<sub>x</sub>$  conversion.<sup>13, 14</sup>

More recently, metal-exchanged zeolites have attracted significant attention, Fe-ZSM-5 and Cu-SSZ-13 being shown to have excellent high- and low-temperature  $NO<sub>x</sub>$ conversion, respectively. These materials have higher  $NO<sub>x</sub>$  conversion efficiencies than metal oxide catalysts, making them candidates for meeting future emissions targets.<sup>10, 15</sup> Some metal-zeolite formulations have other advantageous properties such as reduced ammonia slip, or ammonia released by the exhaust, as well as enhanced low-temperature activity when copper is employed as the active metal.<sup>16, 17</sup> Further, the preparation technique was found to exert influence over the activity of the materials, sublimation of FeCl<sub>3</sub> being found to produce a more resilient active phase in Fe-based formulations.<sup>18</sup>

The differing active temperature ranges of SCR catalysts can be understood by examining the mechanistic pathways that have been proposed to occur on these materials. Two potential pathways have been proposed for the metal oxide-type catalysts, one being a Langmuir-Hinshelwood mechanism involving ammonia and adsorbed nitrate species, and the other being the more generally accepted Eley-Rideal mechanism, in which ammonia is adsorbed at Lewis acid sites that react with gaseous NO and NO2. Mechanistic considerations for metal-loaded zeolites are more complex, reactions being intermediated by  $N_2O_3$  in the case of an active  $MnO_x$  phase, while other materials possess active copper or iron ions as well as metal oxide oligomers.<sup>19-22</sup>

As mentioned above, much of the NO present in the exhaust feed is converted to  $NO<sub>2</sub>$  by the DOC upstream from the SCR system. Since reduction of  $NO<sub>2</sub>$  only occurs through the fast SCR reaction, and since this reaction is kinetically favorable at low temperatures, special attention must be paid to the mechanisms through which this reaction proceeds.<sup>23</sup> Zeolites have been shown to have high fast SCR activity through a variety of intermediates depending on the metal present.<sup>19, 20</sup> However, SCR catalysts alone still present several disadvantages in the engineering and upkeep required to maintain the catalyst.

LNT catalysts were initially developed as an alternative to SCR technologies. LNT catalysts must exhibit several functionalities.<sup>24, 25</sup> The operation of these systems can be broken down into three steps, the first being oxidation of NO to NO2, though this reaction is kinetically limited at low temperatures.<sup>26, 27</sup> Enhancement of oxidation activity is achieved through addition of Pt or Rh, the prototypical LNT formulation being Pt/BaO/Al<sub>2</sub>O<sub>3</sub>.<sup>28</sup> The next step in the process is storage of NO<sub>2</sub> as nitrates or nitrites on a basic metal oxide such as BaO. While enhancing the Ba loading to increase the capacity of these materials is attractive, increased Ba loading has been shown to result in interactions with platinum and decreased oxidation activity either by blocking active Pt sites, or by stabilization of Pt oxides.<sup>29, 30</sup> The oxidation of NO to NO<sub>2</sub> is of critical importance as only nitrates appear on these materials in spectroscopic studies at temperatures above 200  $\degree$ C, an indication that NO does not interact with these materials at operational temperatures; nitrites (indicative of NO adsorption) only being observed at lower temperatures.<sup>31-33</sup> However, when an oxidative component is added to these materials, nearly identical behavior is observed whether NO or  $NO<sub>2</sub>$  is supplied at high temperatures when the oxidation reaction is not kinetically limited.<sup>34</sup> Additionally, Pt and Ba have been found to interact to provide additional  $NO<sub>x</sub>$  storage capacity at the Pt-Ba interface, nitrites being preferentially converted to nitrates in the presence of oxygen.<sup>27</sup> Unfortunately, other exhaust species are problematic for these materials and the initial state of the trapping medium also exerts an influence, the reactivity of Ba species to  $NO<sub>2</sub>$  being in the order oxide > hydroxide > carbonate.<sup>35</sup> This presents a significant challenge to the operation of these materials as  $CO$ ,  $CO<sub>2</sub>$ , and water all

compete for adsorption sites or contribute to the loss of  $NO<sub>2</sub>$  through alternative reduction pathways.<sup>36</sup>

The final step of the LNT process is the desorption and reduction of stored  $NO<sub>x</sub>$ , decomposition of nitrates achieved by the equilibrium change associated with deviation to rich conditions, followed by reduction of the released  $NO<sub>x</sub>$  achieved by reductants adsorbed at precious metal sites.<sup>27</sup> The most challenging aspect of this process is the rapid reduction of a large quantity stored  $NO<sub>x</sub>$ , though these materials have demonstrated remarkable effectiveness in carrying out this reduction. The mechanism by which this reaction proceeds is still debated.<sup>30</sup>

LNT catalysts also present problems with deactivation, increasing  $NO<sub>x</sub>$  and ammonia slip being observed as the catalyst is aged.<sup>37</sup> To resolve these issues, tandem LNT-SCR systems were demonstrated to have a high degree of effectiveness while still eliminating the need to introduce an external reductant for SCR activity.<sup>7, 24, 37</sup> In this configuration, the LNT catalyst produces ammonia, which is adsorbed on the SCR catalyst under rich conditions and serves as a  $NO<sub>x</sub>$  reductant in lean mode, thus reducing  $NO<sub>x</sub>$  that fails to adsorb on the LNT material. Cu-chabazite SCR materials are especially advantageous for this configuration as they can also reduce  $NO<sub>x</sub>$  using adsorbed hydrocarbons as reductant while also limiting N<sub>2</sub>O slip.<sup>7, 38, 39</sup>

#### <span id="page-20-0"></span>1.1.1.2 TWC systems

Stoichiometric gasoline engines require different pollutant mitigation strategies due to the nature of the fuel and exhaust stream. The only catalytic system capable of meeting modern emissions standards in gasoline vehicles is the three-way catalyst (TWC), so called because of the three functionalities the catalyst possesses.  $40, 41$  These materials are composed of a high-surface-area alumina material washcoated onto a monolith, which serves as a support for the active metals, while Pt, Pd, and or Rh comprise the active supported metals, and finally Ce/Zr mixed oxides act to store oxygen for release under rich conditions.<sup>40, 42-44</sup> These catalysts are active in three catalytic processes: reduction of  $NO<sub>x</sub>$  to  $N<sub>2</sub>$ , oxidation of CO to CO<sub>2</sub>, as well as oxidation of hydrocarbons to CO<sub>2</sub>. These catalysts are also active in the steam reforming and water

gas shift reactions that produce CO and  $H_2$ , and  $H_2$  and CO<sub>2</sub> respectively.<sup>40, 41</sup> Stoichiometric conditions provide H2, partially combusted hydrocarbons, and CO as reducing agents with oxygen and  $NO<sub>x</sub>$  providing the oxidative counterbalance. The airfuel equivalence ratio is commonly denoted by λ, or the ratio of the current air-fuel ratio to the stoichiometric value. An air/fuel ratio of 14.7 represents a stoichiometrically equal quantity of oxidants and reductants by weight, while the formula used to derive  $\lambda$  is presented in Equation 4.<sup>45</sup> A  $\lambda$  value of 1 therefore represents stoichiometric conditions, while a  $\lambda$  of less than 1 or greater than 1 indicates rich and lean conditions, respectively. The  $\lambda$  of the automotive exhaust is measured by sensors both before and behind the catalyst bed and is controlled by the fuel injection system through a feedback loop that oscillates slightly above and below true stoichiometry.40 This oscillation is beneficial to maintaining the catalyst's  $NO<sub>x</sub>$  reduction functionality by preventing total oxidation of the  $NO<sub>x</sub>$  reduction component, while still achieving oxidation under rich conditions through the oxygen storage function of the Ce/Zr oxides incorporated in the catalyst washcoat. 46, 47

$$
\lambda = \frac{\text{Air/Full}_{current}}{\text{Air/Full}_{stoichiometric}} \quad (4)
$$

The noble metal or metals deposited on the catalyst determine the functions the catalyst can perform. The most important metal for the  $NO<sub>x</sub>$  reduction process is Rh, this metal's ability to dissociatively adsorb NO into adsorbed N and O atoms being a key feature of the NOx reduction process. The adsorbed N atoms then combine to form  $N_2$ , while adsorbed C and O combine to form  $CO<sub>2</sub>$ . Molecularly adsorbed NO is also capable of reacting with the adsorbed N atoms to form  $N_2$  and adsorbed O.<sup>48-50</sup> The adsorbed O atoms can then combine with adsorbed CO to form  $CO<sub>2</sub>$  via a Langmuir-Hinshelwood mechanism.<sup>51</sup> As the exhaust composition fluctuates from rich to lean conditions, Rh is oxidized from its metallic state to form Rh oxides, which have been shown to be less effective at adsorbing NO<sub>x</sub> and CO.<sup>51</sup> As a result, preventing over-oxidation of Rh is of prime importance to maintaining  $NO<sub>x</sub>$  conversion efficiency, thus limiting this technology to operation under stoichiometric conditions. Addition of Pt to the catalyst such that both Pt and Rh sites are present greatly enhances the conversion of CO over these materials by enhancing oxidation activity.<sup>52</sup> Recent developments in this

technology have focused on the development of Pd-only TWCs due to the increasing costs of Pt and Rh. The development of Pd-catalysts accompanied various engine design changes that allowed the catalyst to be placed closer to the engine, and therefore achieve a higher operating temperature. This advantage serves to mitigate the negative impacts of removing Rh and Pt from the catalyst and has led to widespread adoption of this technology.<sup>53</sup> Additional developments in washcoat components and trimetallic formulations have also been explored.<sup>53</sup>

Various undesirable side reactions can also occur on these materials, including the formation of ammonia and  $SO_3$ . These products are yielded by oxidation of  $SO_2$  and by over-reduction of  $NO<sub>x</sub>$  with  $H<sub>2</sub>$ . Ammonia is undesirable due to its noxious odor while SO3 is corrosive and can damage exhaust components, though trimetallic catalyst formulations have shown excellent selectivity toward  $N_2$  and  $SO_2$ <sup>40, 44, 51, 53</sup> Automotive catalysts must show exceptional durability due to the long lifetimes and harsh conditions demanded by their operational conditions. Various deactivation paths have been studied over exhaust treatment materials, with metal particle agglomeration, heavy metal poisoning if leaded gasoline is used, support degradation by thermal stress and fouling all being important considerations. Fuel contaminants such as sulfur, fuel and engine oil additives, and corrosion products can all harm the function of the catalytic converter.<sup>54</sup> As a specific example, high fuel concentrations of sulfur can lead to the formation of sulfates on ceria that either remain adsorbed, thus blocking adsorption sites, or are eliminated as H2S, a foul-smelling gas. The blocking of adsorption sites by sulfates was found to be limited when zirconia is employed as an additive to ceria.<sup>43, 55</sup>

<span id="page-22-0"></span>1.1.2 The Cold Start Problem

Catalytic converter technologies that reduce  $NO<sub>x</sub>$  have achieved nearly universal adoption, leading to a 44% decrease in anthropogenic  $NO<sub>x</sub>$  emissions in the United States between 1991 and 2011.<sup>56</sup> Modern catalysis technologies are capable of achieving excellent results in both  $NO<sub>x</sub>$  reduction and CO oxidation, nearly complete conversion being achieved under ideal operating conditions. However, emissions standards continue to become more stringent and in the face of an ever-growing fleet of vehicles and stationary  $NO<sub>x</sub>$  sources, near-total elimination of  $NO<sub>x</sub>$  from exhaust streams is of

increasing importance. Considering the effectiveness of modern catalytic systems, further improvement is an especially difficult challenge and must involve enhancement of  $NO<sub>x</sub>$ conversion outside of the optimal performance regions of current technologies.

The most attractive area for catalyst improvement is in the initial phases of vehicle startup. During this period, that concerns only the first seconds of vehicle operation, the catalyst is in a temperature regime where the reactions it typically catalyzes are kinetically limited, thus the catalyst is largely ineffective at  $NO<sub>x</sub>$  reduction. This difficulty is commonly referred to as "the cold-start problem" or, the need to bring the catalyst to its operating temperature as quickly as possible. Catalyst heaters and operation under rich conditions (additional combustion leading to higher temperatures) have both been evaluated to achieve faster temperature increase, but additional engineering controls and operation with excess fuel are undesirable in the context of efforts to improve fuel economy. A more attractive pathway for improvement is to enhance the low-temperature effectiveness of the catalyst system, though other materials must be considered. LNT systems are somewhat effective at lower temperatures, and thus present an appealing starting point for further development, though the reactivity limitations at near-ambient temperatures suggest passive adsorption as a more realistic option. Such a passive  $NO<sub>x</sub>$ adsorber (PNA) material would ideally be placed upstream from the current catalyst and adsorb  $NO<sub>x</sub>$  and other pollutants at low temperature, then desorb these species at elevated temperatures, at which point the downstream catalyst is active.

#### <span id="page-23-0"></span>1.1.3 Passive  $NO<sub>x</sub>$  Adsorbers

The PNA concept was described as early as 1997, initial efforts focusing on the employment of mixed metal oxides.<sup>57</sup> This concept is advantageous in several respects, as it requires no additional monitoring or controls and is broadly applicable to a variety of engine systems. However, the usual durability requirements of automotive catalysts still apply to any material to be deployed as a PNA. Specifically, any PNA material candidate must exhibit high hydrothermal stability, high  $NO<sub>x</sub>$  storage at near ambient temperatures, operability in the presence of competing adspecies such as water, and durability under typical operating conditions.<sup>58</sup> Also of importance to PNA function is the ideal  $NO<sub>x</sub>$ desorption temperature range, in which little  $NO_x$  desorbs below ~200 °C, while all  $NO_x$ 

must desorb by the maximum operating temperature of the system. Any species that remains adsorbed above this temperature would effectively poison the PNA active sites, and thus decrease the longevity and effectiveness of the material by blocking adsorption sites during the next cold-start event.<sup>59</sup> Further, any PNA must also have a high-rate of  $NO<sub>x</sub>$  adsorption, this being necessary to handle the large volume and high flow rate of the exhaust gas while limiting the quantity and thus cost of the PNA material. Early PNA formulations included Pt and Pd on alumina, ceria, ceria/zirconia mixed oxides, as well as alumina-supported copper oxide, nickel oxide and chromium oxide.<sup>57, 60-64</sup> More recent formulations have included the use of silver as the active storage component, in addition to Pd-promoted tungstated zirconia, along with other metal support combinations.<sup>59, 62, 63,</sup> 65-67

Studies of Pt and Pd on ceria and alumina have found overall greater  $NO<sub>x</sub>$  storage over Pt formulations due to the ability of Pt to store  $NO<sub>x</sub>$  as both nitrites and nitrates, while mainly nitrites are observed on Pd formulations.<sup>62, 64, 68</sup> Nitrates are more thermally stable and are thus undesirable as their desorption point is near the maximum of ideal catalyst operating temperatures.<sup>69</sup> Pd-based catalysts therefore achieve greater total NO<sub>x</sub> desorption and received further attention, lean environments being found to greatly enhance  $NO<sub>x</sub>$  storage capacity while also maintaining the active sites on the ceria support.<sup>70</sup> However, ceria materials show a high sensitivity to poisoning by sulfur through formation of sulfates that are more thermally stable than nitrates.<sup>71</sup> This is problematic for the use of metal oxide supports as regeneration from sulfur poisoning is difficult, although increasing catalyst acidity through the use of rare-earth promoters has been shown to improve sulfur resistance on  $MnO<sub>x</sub>$ -based materials.<sup>67</sup>

Another group of materials that has attracted significant attention for employment as PNA's are Pd-loaded zeolites, the high  $NO<sub>x</sub>$  desorption efficiency of Pd and high sulfur tolerance of the zeolite support being especially appealing. Rather than acting as a support for either metallic or metal oxide particles, zeolites provide sites where metal ions can ion exchange directly at zeolite Al-O(H)-Si species, providing higher metal dispersion and thus greater  $NO<sub>x</sub>$  storage capacity than is achievable on other materials.<sup>72</sup> Other properties of the zeolite are also desirable for these applications, the

aluminosilicate framework providing a high degree of thermal stability, especially in small-pore zeolites whose internal cavities are too small for dealumination reactions to readily occur in the presence of water.<sup>73</sup> The Si/Al ratio of the zeolite is an important factor in determining the speciation of the supported metal as well as the overall stability of the framework. Zeolites with lower Si/Al ratio have a greater abundance of Al sites, and therefore a greater abundance of aluminum sites that occupy next-nearest-neighbor (NNN) and next-next-nearest neighbor (NNNN) positions to other Al atoms and are therefore able to support divalent metal cations.<sup>72, 74</sup> Since dealumination reactions are the primary pathway of hydrothermal degradation, low Si/Al ratio materials are also more susceptible to deactivation. Conversely, high Si/Al ratio materials show a greater abundance of isolated Al sites that can support a monovalent metal ion, though the overall abundance of metal-exchange sites is also lower in these materials.<sup>75</sup> Therefore, intermediate Si/Al zeolites are desirable for this application as these materials balance the number of active sites with the stability of the material. However, the distribution of sites and metal speciation is more complex on these materials, and thus their mechanism of action is more ambiguous. 76

Another important parameter in the metal-zeolite interaction is the crystallographic nature of the framework and the number of distinct positions within the framework that Al can occupy. Relatively simple frameworks such as chabazite (CHA) possess only a single crystallographically unique Al or "T" site, so called because of the tetrahedral coordination geometry of the trivalent Al atom.<sup>77</sup> More complex materials have an abundance of distinct T sites; for example, beta (BEA) zeolite has nine such sites. The pore size is also important as this parameter influences reactivity through the limitations placed on the size of molecules that can be included within the zeolite framework. Discussion of the specific details of the chemistry that occurs over these materials is facilitated by an introduction of the various characterization techniques that are most prevalently employed to make these assessments.

#### <span id="page-26-0"></span>1.2 Analytical Techniques

An array of analytical techniques has been used to study metal-loaded zeolites employed as PNA materials and on metal-loaded zeolites more broadly. Typical catalyst characterization techniques are applicable to these materials, including pore volume analysis by  $N_2$  physisorption coupled with the Brunauer-Emmett-Teller (BET) adsorption isotherm, metal reducibility analysis by  $H_2$  temperature-programmed reduction (TPR), as well as a wide array of in-situ adsorption and desorption techniques. Various detectors are employed in these experiments, including infrared spectrometers (FT-IR), mass spectrometers (MS), and thermal conductivity detectors (TCD). FT-IR and MS techniques are especially useful as these methods allow for the identification of specific molecules that adsorb and desorb from the experimental material, temperatureprogrammed desorption (TPD) being the primary method by which  $NO<sub>x</sub>$  storage is quantified. Other techniques are useful in the study of the zeolite framework, including  $^{27}$ Al magic-angle spinning nuclear magnetic resonance spectroscopy (MAS-NMR), a method that allows the identification of variously coordinated Al within the zeolite framework. X-ray diffraction (XRD) is a useful probe of the crystallinity of the zeolite framework and provides a convenient tool to assess the extent of zeolite degradation.

Analyzing the supported metal is especially important in the design of catalytic systems, Pd being appealing for use as a PNA metal for reasons enumerated above. Generally, X-ray techniques have been used to study Pd speciation, especially X-ray photoelectron spectroscopy (XPS), though these studies are often complicated by the high degree of band overlap between the various Pd species. Further complicating differentiation of Pd 3d XPS bands is the low concentration of Pd typically supported by the zeolite framework, and thus low intensity of the Pd 3d bands. Another useful, if underutilized, technique for analyzing Pd speciation is electron paramagnetic resonance (EPR) spectroscopy, which can be used to ascertain the presence and quantity of paramagnetic Pd ions such as  $Pd^+$  and  $Pd^{3+}$ .

By far, the most convenient probe of the reactivity of Pd-loaded zeolites is in-situ diffuse-reflectance FT-IR spectroscopy (DRIFTS), a technique that allows direct observation of surface adsorbates that have IR-active chemical bonds. Fortunately, the IR

spectra of  $NO<sub>x</sub>$ , CO, and a plethora of other species have been extensively characterized by this method, CO adsorption being instrumental in the elucidation of Pd speciation. DRIFTS is the primary analytical technique employed in this contribution, though each of the techniques described above will be used to provide further context and information to aid in deriving DRIFTS band assignments.

#### <span id="page-27-0"></span>1.2.1 In-situ DRIFTS and Microreactor Studies

Specific experimental procedures employed for this work are provided in Appendix 1, though generally experiments in both the DRIFTS and microreactor systems can be described by three steps: pretreatment in either inert or active atmosphere at high temperature followed by cooling to near-ambient temperatures, adsorption of a desired probe gas at near-ambient temperatures, and finally desorption of the probe molecule via TPD with a ramp rate of 10  $\degree$ C/min. The literature is rich with IR spectroscopy studies of adsorbate molecules on zeolites and Pd-loaded zeolites specifically, though much disagreement remains as to the identities of many of the observed IR bands. Discussion of these apparent inconsistencies will be outlined in detail below, especially revolving around the adsorption of CO and NO as probe molecules.

#### 1.2.1.1 Adsorption of Carbon Monoxide on Pd-loaded Zeolites

<span id="page-27-1"></span>The identification of active metal and zeolite sites using CO as a probe molecule dates to the early 1970s, with analysis of many zeolite IR-active bonds also being of paramount significance.<sup>78, 79</sup> A seminal study by Naccace et al. examined the behavior of Pd-Y zeolite under various conditions and established the presence of Pd in oxidation states of  $+1$ ,  $+2$ , and  $+3$ , the odd-numbered ions being examined by EPR. The zeolite in this study was of a Si/Al ratio of  $\sim$ 4, so most of the Pd can be assumed to be Pd<sup>2+</sup> due to the high abundance of proximal Al sites. This led the authors to conclude that the  $Pd^+$  and  $Pd^{3+}$  observed by EPR occurred in concentrations too low to account for any of the observed CO bands.<sup>78</sup> The observed  $Pd^{n+}$ (CO) bands occurred at 2135 and 2110 cm<sup>-1</sup> and were assigned by the authors as CO adsorbed at two different  $Pd^{2+}$  sites. Upon a temperature increase, these species are found to shift to lower wavenumbers, this shift being proposed to result from the formation of  $Pd^{2+}(CO)_2$  species at elevated

temperature.<sup>78</sup> Additional studies on low Si/Al ratio zeolites (Si/Al <10) have been conducted by Khivantsev et al.,  $Pd^{n+}(CO)$  bands of the type  $Pd^{2+}(CO)_2$  (2193-2214 cm<sup>-1</sup>),  $Pd^{2+}({\rm CO})$  and  $Pd^{2+}({\rm CO})({\rm OH})$  (2115-2150 cm<sup>-1</sup>), and  $Pd^{+}({\rm CO})$  at 2075 cm<sup>-1</sup> all observed, although a later contribution amended this last assignment to  $Pd^0(CO)$  species, an assignment more consistent with other reports.<sup>72, 74</sup> The identities of the  $Pd^{2+}(CO)$ <sub>2</sub> bands bear further examination, the CO stretching frequencies of this species being substantially above the gas-phase stretching frequency of the CO molecule. This Pd species is classified as a "super-electrophile" by Khivantsev et al., analogous to the inorganic complex  $[Pd(CO)_4][Sb_2F_{11}]_2$  synthesized by Willner et al. that also expresses symmetric and asymmetric CO stretches at 2212 and 2189  $cm^{-1}$  respectively.<sup>74, 80</sup> Such a label indicates that the Pd in question is adequately electrophilic to eliminate  $\pi$ -backbonding with the coordinated CO molecules. In its inorganic complex form, this Pd species is highly reactive to water, though Khivantsev et al. observed a remarkable stability of this Pd even when water is present in a small pore SSZ-13 zeolite with Si/Al ratio of 6.<sup>74</sup> The super-electrophilic Pd is found to dominate the FT-IR spectrum after CO adsorption onto low Si/Al SSZ-13 materials, the stability of this species being proposed to result from the small pore diameter of this material.<sup>74</sup> The connectivity of this species to the framework is still unclear, though a "pore confinement" effect has been proposed, rather than a more classical exchange at Si-O(H)-Al sites. In previous studies, CO bands in this region were assigned to species associated with  $Pd^{3+}$  ions, though EPR studies have shown inadequate quantities of  $Pd^{3+}$  to support this assignment considering the intensity of the observed Pd(CO) bands.<sup>75, 76, 81, 82</sup>

Intermediate Si/Al zeolite materials show a more complex CO band structure than those observed for low Si/Al materials, super-electrophilic  $Pd^{2+}(CO)$  bands still being observed, though they are often less stable and less abundant.<sup>76, 82</sup> A greater number and intensity of  $Pd^{n+}(\text{CO})$  bands occurs in the range 2110–2185 cm<sup>-1</sup>, the identities of these species being variously assigned to  $Pd^+(CO)$ ,  $Pd^{2+}(CO)$ , and  $Pd^{2+}(CO)(OH)$ , though disagreement exists as to the exact range and behavior of each of these proposed species.<sup>74, 76, 78, 82</sup> CO-TPD analysis of various zeolite frameworks with similar Si/Al showed a pair of  $Pd^{n+}(CO)$  bands that remain to higher temperatures, though these were not observed to decrease in wavenumber upon CO desorption as shown on the low Si/Al

material described above.<sup>76</sup> As a result the authors assign these bands to CO adsorbed on "naked"  $Pd^{+/2+}$  ions exchanged onto the zeolite framework, while the other  $Pd^{n+}(\text{CO})$ bands in this range are assigned to Pd with active oxygen ligands, presumably  $[Pd^{2+}(OH)]^{+}$ . Conversely, a study by Aylor et al. does observe a wavenumber decrease during TPD and assigns this pair of bands to  $Pd^{2+}(CO)$ <sub>2</sub> in agreement with Naccace et al., the reason for these contradictory results not being immediately obvious.<sup>78, 82</sup> Pd<sup>+</sup> species are generally described as possessing lower CO stretching frequencies than  $Pd^{2+}$ , Pd<sup>+</sup>(CO) bands being assigned in the range 2110–2130 cm<sup>-1</sup> while Pd<sup>2+</sup>(CO) bands are assigned to the range  $2130-2185$  cm<sup>-1,76,82</sup> The exception to this consensus is the work of Khivantsev et al. who describes all species in the range  $2110-2185$  cm<sup>-1</sup> as being associated with  $Pd^{2+}$ , either in the form of  $Pd^{2+}(CO)$  or  $Pd^{2+}(CO)(OH)$ , results that are supported by computational and other spectroscopic methods.<sup>72, 74, 83</sup> Given that  $Pd^+$  and  $Pd^{3+}$  are paramagnetic ions, analysis by EPR would be of great benefit. Unfortunately, EPR studies present in the literature are confined to zeolite materials of very low  $(Si/A<sub>1</sub> < 10)$  and very high Si/Al ratios  $(Si/A<sub>1</sub> > 20)$ , situations in which Pd<sup>+</sup> would be unlikely to exist given the high abundance of proximal Al sites and the low total abundance of Al sites respectively, though some  $Pd^+$  is still observed in either case.<sup>75, 81</sup> This limits the current understanding of the quantities of  $Pd<sup>+</sup>$  present on intermediate  $Si/Al$  zeolites, these materials also presenting the strongest evidence of  $Pd<sup>+</sup>$  by CO adsorption.

CO also adsorbs on metallic Pd particles, the CO adsorption behavior over metallic Pd being significantly better understood than that of the  $Pd^{n+}(CO)$  bands. CO adsorption on Pd metal occurs in four distinct manners; singly adsorbed CO molecules on Pd defect sites (2095–2110 cm<sup>-1</sup>), singly adsorbed CO molecules on Pd edge sites (2060– 2095 cm<sup>-1</sup>), doubly bridging CO of the type  $(Pd^{0})_{2}(CO)$  (1850–1990 cm<sup>-1</sup>) and finally, triply bridging (Pd<sup>0</sup>)<sub>3</sub>(CO) (1780–1830 cm<sup>-1</sup>)<sup>74, 82, 84-86</sup>. Previous studies on Pt-loaded zeolites have suggested a linear relationship between CO stretching frequency and metal particle size, the addition of each Pt atom being found to increase the frequency of the CO band by approximately  $7 \text{ cm}^{-1}$ , a behavior also proposed to be relevant to Pd-loaded materials.<sup>86, 87</sup> Of special note is the appearance of sharp  $(Pd^0)_2(CO)$  bands in the

presence of a narrow distribution of Pd particle sizes, whereas broad bands indicate a wider distribution of Pd particle sizes.<sup>84</sup>

The zeolite acid sites can also adsorb CO to a lesser degree, bands appearing on various H-form materials associated with CO adsorbed at Brønsted acid OH occurring at  $\sim$ 2175 cm<sup>-1</sup>. Zeolites also have a pair of distinct Lewis acidic sites that interact with CO, bands appearing at 2195–2200 and 2225–2230 cm<sup>-1</sup>. These bands are proposed to represent CO adsorbed to extra-framework alumina debris (EFAl) and partiallyhydrolyzed framework Al (P-HAl) respectively.<sup>79, 88</sup> These bands are typically much lower in intensity than those resulting from CO adsorption at Pd.

1.2.1.2 Adsorption of Nitrogen Oxides on Pd-loaded Zeolites

<span id="page-30-0"></span>The intended application of these materials also makes the IR examination of adsorbed  $NO<sub>x</sub>$  species such as  $NO$  and  $NO<sub>2</sub>$  pivotally important, a wealth of such studies appearing in the literature. Both  $NO$  and  $NO<sub>2</sub>$  have been extensively explored as adsorbates, dating to early studies by Hadjivanov et al. and Descorme et al. in the 1990s that dealt with Cu-and Pd-loaded materials, respectively.<sup>89, 90</sup> Bands in the NO stretching region typically fall into three categories: bands representing  $NO<sup>+</sup>$  species that occur in the range 2100–2200 cm<sup>-1</sup>, nitrosyl bands of the type  $Pd^{n+} (NO)$  appearing in the range  $1800-1900$  cm<sup>-1</sup>, and finally molecularly adsorbed NO<sub>2</sub> and nitrate bands in the range 1550–1660 cm<sup>-1</sup>.<sup>59, 90-92</sup> There is broad agreement as to the identities of these various band groups; however, the identities of the nitrosyl bands are still debated to some degree. Early studies by Descorme et al. suggested that the two bands observed in the nitrosyl region represent  $Pd^+(NO)$  and  $Pd^+(NO)(NO_2/water)$  respectively, though more recent studies ascribe these species to a higher frequency feature representing  $Pd^{2+}(NO)$ and a lower frequency feature representing  $Pd<sup>+</sup>(NO)$ .<sup>74, 90</sup> A pair of bands assigned to  $Pd^{n+}(NO)$ <sub>2</sub> is also observed at higher frequencies than the respective mono-nitrosyl bands, though this species is proposed to play no role in PNA applications as it decays readily under vacuum.<sup>74</sup> In contrast, Mihai et al. assign bands in the 1700-1800 cm<sup>-1</sup> range to dinitrosyl complexes  $(Pd^{n+}(NO)_2)$  according to the works of Hess et al.<sup>91, 93</sup> Some studies propose that NO reduces  $Pd^{2+}$  and or  $[Pd^{2+}(OH)]^{+}$  to  $Pd^{+}$  upon adsorption, while others propose that the super-electrophilic  $Pd^{2+}$  is the species that is reduced.<sup>74, 91</sup> The former

supposition is supported by Mihai et al. who explored the effect of  $Si/A1$  ratio on  $NO<sub>x</sub>$ adsorption by NO-DRIFTS, showing that the band associated with  $Pd^{2+}(NO)$  is extremely weak at high Si/Al (38, 300), both materials presenting largely isolated Al sites, and thus a greater relative abundance of  $Pd^{+,91}$ 

The behavior of NO bands in the range  $2100-2200$  cm<sup>-1</sup> has also been extensively examined, initially being assigned to  $NO^{2+}$  by Descorme et al., but more recently being ascribed to NO<sup>+</sup> that results from the reduction of  $Pd^{2+}$  to  $Pd^+$ , and also from interaction of NO with the zeolite Brønsted acid sites, thus explaining the appearance of these bands on H-form zeolite materials.<sup>91, 94</sup> These bands are also more prominent when oxygen is present, due to enhanced formation of higher nitrogen oxides under these conditions.<sup>74, 95</sup> However, the stability of these species varies from study to study, being found to decompose readily in the presence of water and under vacuum, while under dry inert gas these bands are stable and persist to higher temperatures.<sup>91, 94, 95</sup>

IR bands related to the higher nitrogen oxides are also well documented,  $NO<sub>2</sub>$  $NO<sub>3</sub><sup>-</sup>, N<sub>2</sub>O<sub>3</sub>$ , and an HNO<sub>2</sub><sup>\*</sup> activated complex all being reported.<sup>92</sup> A number of nitrate features are often apparent upon adsorption of  $NO<sub>2</sub>$ , this molecule being proposed to adsorb at cationic zeolite sites such as P-HAl and EFAl.<sup>59, 92, 96</sup> A trio of coordination modes is available to this species as evidenced by a triplet of FT-IR bands representing bridging (1600–1650 cm<sup>-1</sup>), bidentate (1585–1500 cm<sup>-1</sup>), and monodentate nitrates  $(1530-1480 \text{ cm}^{-1})$ . These species have high thermal stability and represent a NO<sub>x</sub> storage pathway that is less desirable for PNA materials as a result.  $N_2O_3$  bands appear at 1950 and 1570 cm<sup>-1</sup> on BEA, a result that indicates N<sub>2</sub>O may also be stabilized by the zeolite BEA pores.<sup>94</sup> The NO stretch of  $HNO_2$ <sup>\*</sup> has been observed at 1621 cm<sup>-1</sup> on ZrO<sub>2</sub>, this species forming in the presence of  $H_2$  over zeolites.<sup>92, 97</sup>

#### 1.2.2 Pd-loaded zeolites as passive  $NO<sub>x</sub>$  adsorbers

<span id="page-31-0"></span>The properties of zeolites as potential PNAs has been explored for various materials, but many questions remain.. 59 Several factors must be considered in developing PNA materials: the total  $NO<sub>x</sub>$  storage capacity, the utilization efficiency of precious metals (NO<sub>x</sub>/Pd ratio), as well as the rate of NO<sub>x</sub> uptake, a parameter that has a heavy influence on the  $NO<sub>x</sub>$  storage efficiency, that is, the percentage of the total  $NO<sub>x</sub>$ that is removed from the feed.

The  $NO<sub>x</sub>$  to precious metal ratio has been evaluated over a wide range of materials using both Pt and Pd alone and in combination as the active metal.  $NO<sub>x</sub>/Pd$  ratios as high as 1 have been reported for materials with near total atomic dispersion of Pd, values of 0.5–0.7 being more typical.<sup>59, 72</sup> Unfortunately, these high ratios are only achievable to loadings of 1–2 wt%, after which the metal utilization decreases, suggesting saturation of exchange sites occurs near 1 wt.% of active metal.<sup>59, 72</sup> The NO<sub>x</sub>/Pd ratio was shown by Ryou et al. to be improved after hydrothermal treatment, indicating that during preparation of the material both ion-exchanged Pd and  $PdO<sub>x</sub>$  clusters are formed, the latter being re-distributed to ion-exchanged positions after hydrothermal treatment.<sup>98</sup> This and other contributions indicate high mobilities of small  $PdO<sub>x</sub>$  and Pd metal species that can be increased by the presence of water and high temperatures, while other studies suggest the formation of pore-confined metal particles that are stabilized when they are larger than the pore openings, in a "ship in a bottle" process.<sup>76, 84, 99, 100</sup>

Unfortunately,  $PdO_x$  clusters do not adsorb CO or NO to any great degree, thus rendering this Pd population invisible to IR examination. These results show the importance of Pd speciation to  $NO<sub>x</sub>$  storage and demonstrates the high sensitivity of these materials to synthesis and pretreatment conditions. The overall rate of  $NO<sub>x</sub>$  storage over these materials has not been extensively explored, though the near-ambient temperature activity of  $NO<sub>x</sub>$  adsorption suggests a low activation barrier and thus a process that is driven primarily by mass diffusion effects.<sup>59</sup> Several zeolite materials have exhibited NO<sub>x</sub> storage efficiencies near 100 % for periods up to 60 s, such behavior being sufficient to mitigate most cold start NO<sub>x</sub> emissions.<sup>76, 97, 101, 102 Of further consideration is the</sup> behavior of  $NO<sub>x</sub>$  in the presence of other automotive exhaust components, the reaction chemistry in these cases being more complex.

The  $NO<sub>x</sub>$  desorption temperature is also a key parameter, as adsorbates that remain after exposure to the maximum catalyst temperature effectively act as catalyst poisons. The observed behavior is found to substantially depend on the ramp rate employed during TPD studies, though a rate of  $10 \degree C/\text{min}$  is most common and allows

adequate differentiation of adsorbates for kinetic evaluation.<sup>59</sup> However, such a slow ramp rate is unrealistic under real application due to the rapid temperature changes that occur during a cold start, further investigation revealing a non-linear relationship between ramp rate and desorption temperature, though above rates of  $\sim$ 100 °C/min the changes in desorption temperature are less pronounced.<sup>102-104</sup> Generally, either one or two desorption events are observed, one at lower temperatures (100–250 °C) and another at higher temperatures (200–500 °C).<sup>59</sup> Desorption temperatures vary widely depending on the zeolite framework, metal dispersion, and presence of other exhaust species, though generally any desorption below 200  $\degree$ C is undesirable, as this will lead to incomplete NO<sub>x</sub> reduction on the downstream catalyst.<sup>24</sup>

The  $NO<sub>x</sub>$  desorption behavior is heavily influenced by the speciation of adsorbed  $NO<sub>x</sub>$ , six distinct adsorption pathways being identified in the literature by Gu et al. that lead to three distinct groups of adsorbates. <sup>59</sup> The species that exhibit the lowest temperature desorption are NO<sup>+</sup> and NO<sub>2</sub><sup>+</sup> weakly interacting with zeolite acid sites, these species comprising much of the  $NO<sub>x</sub>$  that desorbs below optimal temperatures.<sup>105</sup> Nitrosyl species are found to comprise  $NO<sub>x</sub>$  that desorbs near and somewhat above the optimum temperature, though the desorption range of these species is again heavily dependent on the preparation and pretreatment conditions, as well as the presence of other exhaust species.<sup>74, 83, 91</sup> Finally, species related to  $NO<sub>2</sub>$  adsorbed as nitrates desorb at intermediate to high temperatures (300–500  $^{\circ}$ C), these species representing a less desirable adsorption pathway due to the potential failure of these species to completely desorb. 106

## 1.2.2.1 Effect of Reductants

<span id="page-33-0"></span>The effects of CO on PNA materials have been thoroughly explored during all parts of the PNA process on a variety of zeolite frameworks. The presence of CO during high-temperature pretreatment has been found to be especially problematic for the longterm activity of these materials, CO being found to lead to greater  $Pd<sup>0</sup>$  particle mobility and agglomeration with concurrent loss of PNA activity.<sup>36, 103, 107</sup> The agglomeration of large  $Pd<sup>0</sup>$  particles outside the zeolite framework is proposed to be responsible for the loss of PNA activity, these particles failing to re-disperse to ion-exchanged Pd upon oxidative

regeneration.<sup>5, 100, 107, 108</sup> During low-temperature adsorption of  $NO<sub>x</sub>$ , CO is found to have a promotional effect on PNA behavior both through increased  $NO<sub>x</sub>$  storage capacity and elevation of  $NO_x$  desorption temperature.<sup>76, 103</sup> These results were initially proposed to be a result of the reduction of Pd to lower oxidation state, i.e.  $Pd^+$ , that binds NO more strongly. Other studies have suggested that this promotional effect results from the formation of a  $Pd^{2+}(NO)(CO)$  co-adsorbed complex that allows for greater quantities of  $NO<sub>x</sub>$  to adsorb while also representing a more thermally stable species than NO adsorbed alone.83 Other pathways have also been proposed, including the formation of an NCO species like that observed on ceria supported metals, though this band occurs in a similar range to the CO stretch of the reported co-adsorbed complex and thus differentiation between these two species is difficult. $97$ 

The effect of  $H_2$  on Pd-loaded zeolites is more benign, reduction at high temperatures in  $H_2$  demonstrated to be more reversible due to the formation of smaller Pd particles that are more readily re-dispersed to exchanged Pd ions.<sup>100, 108</sup> Theis et al. found  $H_2$  to promote  $NO_x$  adsorption capacity, and suggested the formation of  $HNO_2^*$  species, hydrocarbons having a similar effect.<sup>97</sup> The reaction of hydrocarbons over Pd-loaded zeolite materials is more complex, with the participation of water gas shift and steam reforming reactions being significant. $40, 41$  The presence of hydrocarbons at high temperature leads primarily to combustion under lean conditions, though ethylene was shown to promote  $N_2O$  formation at high temperatures on Pd-loaded ceria/zirconia through an unselective  $NO<sub>x</sub>$  reduction pathway.<sup>58</sup> The formation of alkyl nitrites is also proposed to promote  $NO<sub>x</sub>$  adsorption on ceria-based materials, though these last reactions are not in evidence for zeolite-based materials.<sup>58, 97</sup>

#### 1.2.2.2 Effect of Oxidants

<span id="page-34-0"></span>Oxidizing agents typically consist of  $O_2$  and  $NO_2$ , the presence of these species exerting influence in all aspects of PNA reactivity. The presence of  $O_2$  in the gas feed leads to enhanced oxidation of NO to  $NO<sub>2</sub>$ , thus many of these effects are also interrelated. This oxidation pathway is found to enhance  $NO<sub>x</sub>$  storage capacity through the formation of nitrates at EFAI sites in addition to the formation of NO<sup>+</sup> and NO<sub>2</sub><sup>+</sup>, these reactions being active even at near-ambient temperatures.<sup>76, 90, 95</sup> However, NO

oxidation competes with the process of  $NO<sub>x</sub>$  adsorption and desorption, though the specifics of this relationship are still poorly understood.<sup>59</sup> Oxidation of NO to NO<sub>2</sub> is observed both during and after  $NO<sub>x</sub>$  storage and is proposed to occur through three mechanisms:

$$
Z^{-} - Pd^{2+} - OH + NO \rightarrow NO_2 + H_2O + Z^{-} - Pd^{+} \quad (5)
$$
<sup>90</sup>  

$$
Z^{-} - Pd^{2+} - O - Pd^{2+} - Z^{-} + NO \rightarrow NO_2 + 2[Z^{-} - Pd^{+}] \quad (6)
$$
<sup>76,109</sup>  

$$
PdO_2 + 2NO \rightarrow NO_2 + NO - PdO \quad (7)
$$
<sup>76,109</sup>

In each case Pd is reduced during  $NO<sub>2</sub>$  formation, thus indicating that the role of  $Pd<sup>+</sup>$  species is significant, as NO is proposed to adsorb at the reduced Pd sites. The presence of lower oxidation state Pd after  $NO<sub>x</sub>$  adsorption was confirmed by FT-IR and XPS spectroscopy performed by Zheng et al. but again, quantitative analysis of these species is challenging.<sup>76</sup> Furthermore, direct storage of NO at  $Pd^{2+}$  is proposed in many cases, and as a result the ratio of NO removed from the feed to NO<sub>2</sub> formed cannot readily be differentiated, preventing deconvolution of adsorption and oxidation activity.<sup>59</sup> An additional complication to understanding these processes is the weak interaction between  $NO<sub>x</sub>$  and the zeolite Brønsted acid sites that results in formation of  $NO<sup>+</sup>$  and  $NO<sub>2</sub><sup>+</sup>$  even in the absence of active metals.<sup>82</sup> The oxidizing effects of NO<sub>2</sub> on ionic Pd species has not been well characterized by FT-IR experiments, though an equimolar quantity of NO and  $NO<sub>2</sub>$  has been shown to enhance the activity of Cu-zeolite SCR catalysts and other results on Pd zeolites suggest this is the case for these materials as well.76, 110

#### 1.2.2.3 Effect of Water

<span id="page-35-0"></span>Water has been shown to influence various aspects of PNA behavior. Treatment of these materials in the presence of water at high temperatures is referred to as hydrothermal aging, this process being shown to impact both the zeolite and the exchanged metal. Several studies have explored the hydrothermal stability of assorted zeolite frameworks under relevant conditions, in addition to Pd on ceria/zirconia and
SCR catalysts.<sup>19, 70, 73, 98</sup> Dealumination of the zeolite is an important deactivation pathway for these materials and is facilitated by water. Elimination of Al sites from the framework has been shown to lead to metal particle agglomeration and loss of zeolite  $NO<sub>x</sub>$  storage sites.<sup>100, 111</sup> Generally, smaller pore zeolite frameworks are more resilient to dealumination by hydrolysis due to the smaller volume available for the reaction to take place, though the hydrophobicity of the framework is also important for the lability of the Al species involved in this reaction.<sup>94</sup> The Si/Al ratio is also important to the dealumination process, as lower Si/Al ratio materials simply have more sites at which this process can occur and are more hydrophilic in nature. Hydrothermal aging has also been shown to facilitate Pd migration in small pore zeolites such as SSZ-13, initial Pd deposition leading to the formation of  $PdO_x$  particles due to diffusion limitations within the zeolite framework.<sup>98, 111</sup> These  $PdO_x$  particles are redispersed into ion-exchange sites upon hydrothermal aging, as demonstrated by the appearance of a new  $NO<sub>x</sub>$  desorption feature and through  $H_2$  TPR of the treated and untreated materials. The effect of water on Pd speciation is less thoroughly understood, though the formation of hydrated  $Pd^{n+}$ complexes is proposed to decrease  $NO<sub>x</sub>$  storage.<sup>59, 102</sup>

Water also has an impact on the  $NO<sub>x</sub>$  adsorption behavior, as water interacts strongly with the zeolite framework at near-ambient temperatures and thus competes for adsorption sites during the initial seconds of PNA operation.<sup>76</sup> This result is a discouraging one for the implementation of these materials as water has been demonstrated to dramatically suppress  $NO<sub>x</sub>$  adsorption at 100 °C and below.<sup>76, 112</sup> This has been proposed to be a result of either the inhibition of NO oxidation, or of the blocking of zeolite Brønsted acid sties.<sup>102, 112</sup> However, the inhibitive effects of water have been shown to be tempered somewhat by the presence of other exhaust components such as CO and  $H_2$  at low temperatures.<sup>76, 103</sup> The presence of ethylene was found to inhibit  $NO<sub>x</sub>$  storage to a greater degree than water alone on Pd ceria/zirconia materials, while CO and  $H_2$  as reductants failed to produce a similar result.<sup>113</sup> These results are explained by the influence of water on three disparate reactive intermediates, including alkyl nitrates and nitrites, NCO, and finally  $HNO<sub>2</sub><sup>*</sup>$ . These intermediates are shown to be produced by ethylene, CO, and H<sub>2</sub> respectively, though these species have not been observed on zeolite-based materials.<sup>113</sup>

#### 1.3 Dissertation Aims and Scope

From the preceding discussion, several key areas of investigation become apparent. First and foremost, the identification of FT-IR bands through careful experimental observation is clearly indicated in the absence of adequately corroborating techniques. To achieve this goal, a wide variety of adsorbates and pretreatment conditions must be employed over a consistent set of materials to provide a more complete picture of the effects of individual gas species present in the automotive exhaust stream. The next chapter of this thesis revolves around the Pd speciation present on the two zeolite frameworks chosen for this study, Pd-BEA and Pd-CHA. Pd-BEA was prepared at various Pd loadings to better understand the thermodynamic favorability of Pd species, while also providing much needed information for  $Pd^{n+}(CO)$  band identification. Both of these materials are of intermediate Si/Al ratio (12–15) and as a result are representative of the most complex  $Pd^{n+}(CO)$  band structure as observed by Zheng et al.<sup>76</sup> A further advantage of comparing materials of similar Si/Al is the more facile deconvolution of framework-related effects from other processes, these two frameworks possessing widely different pore size and framework structure complexity. The effects of water on these materials will also be examined at both high and near-ambient temperatures.

The next chapter will explore the effects of oxidizing and reducing pretreatment conditions on the Pd speciation of these materials. The effect of reductants such as  $H_2$  and CO on Pd speciation will be examined to gain better understanding of the deactivation modes of these materials. Next, the role of zeolite sites in PNA activity will be more fully explored through examination of the H-form zeolite materials in comparison to their Pdloaded counterparts to derive better understanding of the FT-IR bands resulting from adsorption of  $NO<sub>x</sub>$  and the zeolite sites responsible for the formation of these species. The role of partially hydrolyzed Al sites on zeolite BEA in  $NO<sub>x</sub>$  storage is examined, as well as their potential interaction with Pd, a special focus being placed on deconvolution of the behavior of NO and  $NO<sub>2</sub>$ .<sup>88</sup> Finally, the effects of NO and  $NO<sub>2</sub>$  on these materials will be examined during and after pretreatment, as well as the adsorption and desorption behavior from these materials pretreated in other reducing and oxidizing environments.

22

#### CHAPTER 2. EFFECT OF Pd LOADING AND WATER ON Pd-SPECIATION

#### 2.1 Introduction

This chapter will focus on a basic description of the zeolite materials used in this work, as well as initial experiments undertaken to derive FT-IR band assignments. The resulting CO band assignments will be presented for both BEA and CHA. The effects of Pd-loading on BEA will also be described, these data providing evidence on the preference of Pd for certain exchange sites. Finally, an examination of the effects of water on these materials will be performed, the effects of water examined both during pretreatment and during CO adsorption. The catalyst materials and data presented in Table 2.1 were contributed by Mr. Trevor Lardinois at Purdue University, while XPS spectra were collected by Dr. Olivier Heintz at the University of Burgundy, France.

### 2.2 Effect of Pd-loading on Pd-BEA

#### 2.2.1 Catalyst Characterization

The catalyst materials were prepared as described in Appendix 2 and characterized by a combination of techniques. The BEA zeolite was obtained commercially, while the CHA zeolite was synthesized at Purdue University according to the methods of Di Iorio et al.<sup>114, 115</sup> Zeolites were initially prepared in Na<sup>+</sup> form, Pd being added by incipient wetness impregnation, followed by drying and calcination at 550 °C. For each material, the Si/Al, Pd wt%, extent of Pd exchange, and crystallinity were evaluated. In total, seven materials were examined in this work, H-CHA and H-BEA, in addition to  $0.1$ ,  $0.6$ , and  $1.4$  wt% Pd-BEA and  $0.6$  wt% Pd-CHA. Lastly, a 1 wt% Pd-Si-BEA material was prepared, a sample in which no Al is incorporated into the BEA framework, providing an example of a material in which all Pd is present as  $PdO_x$ particles. In each case, atomic absorption spectroscopy was used to determine the Si/Al and the Pd wt%, XRD was used to examine the crystallinity of the materials and finally, H2-TPR was employed to assess the extent of Pd exchange. The results of these analyses

are presented in Table 2.1, while the experimental data used to determine these values are presented in Appendix 2.

Sample	$Si/A1*$	Pd wt $\%^*$	$Pd/A1*$	$Pd_{iso}/Pd_{tot}^+$
$1\%$ Pd-Si-BEA	$\infty$	0.96	--	0.00
$0.1\%$ Pd-BEA	12.5	0.13	0.009	0.67
$0.6\%$ Pd-BEA	12.5	0.60	0.04	0.78
$1.4\%$ Pd-BEA	12.5	1.36	0.09	0.75
$0.7\%$ Pd-CHA	14	0.68	0.06	0.30

Table 2.1 Characterization of air-treated (550 °C) Pd-zeolite materials

\*Determined by atomic absorption spectroscopy +Determined by  $H_2$  TPR

The Si/Al ratio of the materials is intentionally similar, to limit the effect of this parameter on the Pd speciation and PNA behavior. The  $Pd_{iso}/Pd_{tot}$  is an especially important parameter, as this is an indication of the extent to which Pd is ion-exchanged rather than present as  $PdO_x$  particles. This determination was made possible by the differing reduction temperatures of exchanged Pd cations and  $PdO<sub>x</sub>$  particles, the former exhibiting a reduction event at  $35-200$  °C and the latter reducing at  $-10-35$  °C. The quantity of ion-exchanged Pd was then estimated assuming a 1:1 stoichiometric ratio of  $H_2$  to Pd<sup>2+</sup> ions, as determined by a calibrated TCD signal. This also assumes the majority of Pd cations are divalent after pretreatment in air, an assumption shown to be valid by DRIFTS results presented in the next section.  $H_2$  consumption was also observed at  $-40$ °C on both the H- and Pd-form materials, this being a result of Ar desorption from the zeolite micropores during TPR rather than a true reduction event.<sup>116</sup>

The high degree of Pd ion-exchange on the BEA materials is notable, showing that ion-exchanged Pd represents greater than two-thirds of Pd present. Meanwhile, for the 0.7% Pd-CHA material only about 30% of Pd was found to be present in an ionexchanged state, while the remainder is present as  $PdO_x$  and  $Pd<sup>0</sup>$  particles. This result is consistent with other reports on CHA and SSZ-13 frameworks, the extent of Pd dispersion being limited by the rate of diffusion through the small pores of these materials.<sup>100</sup> The framework structure of each material was found to be consistent with XRD analyses published elsewhere, thus confirming the framework structures remained intact through Pd impregnation and calcination.<sup>117, 118</sup> The pore size of the two

24

frameworks is substantially different, CHA having a pore diameter of  $3.7 \times 3.7$  Å, while BEA, a partially disordered material, has both straight and sinusoidal channels with diameters of  $6.6 \times 6.7$  Å and  $5.6 \times 5.6$  Å, respectively. The framework complexity of these materials is also substantially different, CHA consisting of 8-, 6-, and 4-membered rings and a single T site, while BEA is constituted of 4-, 5-, 6-, and 12-membered rings with 9 distinct T sites.<sup>77</sup>

# 2.2.2 CO-DRIFTS

To examine the Pd speciation, each of the BEA materials was pretreated in air or Ar for 1 h at 500 °C to remove water from the zeolite prior to adsorption. CO was then adsorbed at room temperature and a concentration of 1000 ppm in Ar on each of the BEA materials for ten minutes, as shown in Figure 2.1. The band assignments and associated references are shown in Table 2.2.



Figure 2.1 DRIFT spectra of CO adsorbed on Pd-BEA pretreated at 500 °C for 1 h with air or Ar rigorously dried by cold trap. To visualize Pd speciation, 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C. Top: ionic Pd range. Bottom: metallic Pd range.

Band $(cm^{-1})$	Assignment	Position	References
1790	(Pd <sup>0</sup> ) <sub>3</sub> (CO)	External Pd <sup>0</sup>	82, 119
1950	(Pd <sup>0</sup> ) <sub>2</sub> (CO)	Pore-confined $Pd^0$	82, 119
1990	(Pd <sup>0</sup> ) <sub>2</sub> (CO)	External Pd <sup>0</sup>	82, 119
2078	$Pd^0(CO)$	$Pd0$ defect	82, 119
2098	$Pd^0(CO)$	$Pd0$ defect	82, 119
2113	$Pd^+(CO)$	Isolated Al	75, 76, 78, 82, 102
2125	$Pd^{2+}(CO)(OH)$	Isolated Al	78, 102
2132	$Pd^+(CO)$	Isolated Al	78, 102
2144	$Pd^{2+}(CO)(OH)$	Isolated Al	75, 76, 78, 102
2154	$Pd^{2+}(CO)$	Paired Al	82, 102
2160	$Pd^{2+}(CO)$	Paired Al	82, 102
2169	$Pd^{2+}(CO)$	Paired Al	82, 102
2183	$Pd^{2+}(CO)$	Paired Al	82, 102
2189	Super-electrophilic $Pd^{2+}(CO)_2$	Pore-confined ion	74,80
2212	Super-electrophilic $Pd^{2+}(CO)_2$	Pore-confined ion	74,80

Table 2.2 Pd(CO) DRIFTS band assignments and references for BEA materials

These spectra were collected under rigorously dry conditions, a cold trap being used to remove residual water from the pretreatment gas. As expected, CO adsorption on the H-BEA material is weak, the only obvious feature being a CO stretch related to CO adsorbed at Lewis acid sites at 2185 cm<sup>-1</sup>. A pair of much weaker and broader features is also observed in the ranges 2075–2140 and 2140–2180  $cm^{-1}$ , these representing CO condensed in the zeolite pores and CO adsorbed at Brønsted acid sites, respectively.<sup>88, 120</sup> The 0.1 wt% Pd material shows the presence of the various  $Pd^{n+}(\text{CO})_x$  bands, all bands observed on the more highly Pd-loaded materials being in evidence. This suggests the thermodynamic stability of the various Pd species is not highly differentiated, though the most intense bands occur at frequencies typically associated with Pd<sup>+</sup>(CO) (2119 and  $2134 \text{ cm}^{-1}$ ), indicating a slight preference for the filling of isolated Al sites at low Pd loading.<sup>76</sup> The high-frequency  $Pd^{n+}({\rm CO})$  bands (>2150 cm<sup>-1</sup>) can be identified as  $Pd^{2+}$ (CO) species (2154, 2160, 2169, and 2183 cm<sup>-1</sup>) and the symmetric and asymmetric stretches of super-electrophilic  $Pd^{2+}(CO)_2$  species (2189, 2212 cm<sup>-1</sup>). The literature broadly agrees on the assignment of these bands as CO adsorbed at divalent Pd cations, though the true nature of the super-electrophilic complex was only recently described by Khivantsev et al.<sup>74, 76</sup> The abundance of  $Pd^{2+}(CO)$  bands is indicative of the BEA framework complexity, five electronically distinct  $Pd^{2+}$  species being observed. Under these rigorously dry conditions, only very weak metallic  $Pd^0(CO)$  bands are observed at frequencies consistent with bulk Pd particles, suggesting that only limited auto-reduction to  $Pd<sup>0</sup>$  occurred during inert treatment.<sup>82</sup>

On the 0.6 wt% Pd material,  $Pd<sup>+</sup>(CO)$  bands are again the most intense, though the frequency of the band at 2134 cm<sup>-1</sup> increases to 2136 cm<sup>-1</sup>, an observation that suggests the presence of multiple  $Pd^{n+}(CO)$  species of similar CO frequency, the increase or decrease in abundance of one species leading to a shift of the local band maximum. This band occurs in an area of overlap between  $Pd^+(CO)$  and  $Pd^{2+}(CO)$  bands, and thus the assignment of this species is ambiguous, based on the available literature.<sup>74, 76, 82</sup> The Arpretreated 1.4% Pd material continues this trend, the band appearing at 2140 cm<sup>-1</sup>, though at this point  $Pd^{2+}(CO)$  bands become dominant in intensity. The frequency increase of this band occurs simultaneously with the increasing abundance of  $Pd^{2+}$ , the first indication that the high-frequency contributor to this band represents a divalent Pd cation.

The air-pretreated 1.4% Pd-BEA further reinforces this conclusion, as this band shifts again to 2144 cm<sup>-1</sup> with the concurrent appearance of a shoulder to the  $Pd<sup>+</sup>(CO)$  band  $(2119 \text{ cm}^{-1})$  at a frequency of approximately 2125 cm<sup>-1</sup>. The air-pretreated 1.4% Pd-BEA shows a predominance of  $Pd^{2+}(CO)$  species and points to the conclusion that most Pd is present as  $Pd^{2+}$  in this state, thus demonstrating the validity of the 1:1 H<sub>2</sub> to Pd stoichiometry used in H<sub>2</sub>-TPR calculations. These results reveal a complex  $Pd^{n+}(\text{CO})$ band structure in the range  $2110-2145$  cm<sup>-1</sup>, at least four distinct Pd species exhibiting CO bands in this range.

An EPR study of Pd-BEA zeolite carried out by Stokes et al. revealed that more than one  $Pd<sup>+</sup>(CO)$  complex is formed during CO adsorption, due to a heterogeneity of EPR signals upon exposure of Pd-BEA to CO, though the CO coordination number per Pd was not determined and  $Pd^+$  was not quantified.<sup>75</sup> EPR studies have been conducted on other Pd-loaded zeolites, establishing the presence of  $Pd<sup>+</sup>$  species on a variety of frameworks, though none of these studies provide an example of an intermediate Si/Al material, and thus the quantities of  $Pd^+$  observed are always low.<sup>75, 78, 81</sup> Taking these studies into account, it can be inferred that some of the  $Pd^{n+}(CO)$  bands observed in the region 2110–2145 cm<sup>-1</sup> represent  $Pd^+(CO)$  bands, while the others likely represent additional divalent Pd species.

More information can be gleaned from the TPD of CO from air-pretreated 1.4% Pd-BEA as shown in Figure 2.2, spectra being recorded every 100 °C. As the temperature is increased from 25 °C to 100 °C, the bands in the range 2110–2145 cm<sup>-1</sup> become readily distinguishable into four features at 2144, 2132, 2125, and 2119 cm<sup>-1</sup>, though this last band is only evident as a weak shoulder to the band at  $2125 \text{ cm}^{-1}$ . A pair of bands also appears in the metallic  $Pd^0(CO)$  frequency range at 100 °C (2078, 2098 cm<sup>-1</sup>), these representing CO at  $Pd<sup>0</sup>$  particle defect sites. By 200  $^{\circ}$ C nearly all the CO has desorbed, but a pair of bands remains at 2132 and 2113  $cm^{-1}$ , these disappearing by 300 °C. The shifting of bands to lower frequency upon TPD has been observed previously and has been explained either as an effect of decreasing CO coverage, or as a more thermally stable  $Pd^{n+}({\rm CO})$ <sub>2</sub> species.<sup>78, 82</sup> However, neither of these explanations withstands scrutiny; CO coverage effects are only described over metal particles and a polycarbonyl species

29

would be expected to occur at higher wavenumber, the opposite direction of the observed behavior. Further, Zheng et al. do not observe this behavior on any of the Pd-loaded frameworks examined in their study.<sup>76</sup> The appearance of  $Pd^0(CO)$  bands with increasing temperature provides a valuable clue however, as this is indicative of Pd reduction by CO. Since  $Pd^+(CO)$  bands are generally agreed to be at lower frequency than  $Pd^{2+}(CO)$ bands, the observed decrease in band frequency can therefore be interpreted as the reduction of  $Pd^{2+}$  ions to  $Pd^+$  with the simultaneous appearance of Pd metal particles that likely results from further reduction of Pd<sup>+</sup>. Due to the entangled effects of Pd reduction and CO desorption, the extent to which the high-frequency  $Pd^{2+}$  features are reduced in this process cannot be readily discerned, though the frequencies of these bands remain relatively unaffected during desorption.



Figure 2.2 1.4% Pd-BEA: DRIFT spectra of CO desorption after pretreatment at 500 °C for 1 h with air which has been rigorously dried by cold trap. CO was adsorbed for 10 minutes at 25 °C before the catalyst was purged for 10 minutes in Ar prior to ramping to remove gas-phase CO. Temperature was ramped at 10 °C/min to 500 °C with spectra being collected every 100 °C.

These observations suggest the presence of three distinct populations of  $Pd^{2+}$  ions, the super-electrophilic  $Pd^{2+}$  ions, a group of  $Pd^{2+}$  ions exchanged at the zeolite framework of intermediate electrophilicity, and finally a group of easily reducible  $Pd^{2+}$  species that have electrophilic character similar to that of  $Pd^+$ . The CO stretching frequencies of these

bands can be used as a relative measure of the metal electrophilicity due to the effects of π-backbonding, in which electron density is transferred from the filled metal d-orbitals to the empty antibonding orbital (LUMO) of CO, thus lengthening the CO bond and altering its stretching frequency. These observations point to the possibility of a  $Pd^{2+}$  species that is already ligated by another species,  $[{\rm Pd}^{2+}(\rm OH)]^+$  being proposed to also present CO bands in this region.<sup>74, 94</sup> Examination of the OH region of the FT-IR spectra would theoretically reveal the presence of a Pd(OH) stretch, though zeolite OH bands also occur in this region and exert convoluting effects during CO adsorption. To facilitate observation of changes in the zeolite OH region relative to the H-form material, the spectrum of the H-form material was employed as the background spectrum for the dehydrated Pd-loaded materials such that any bands that rise above the baseline would indicate the development of additional OH species relative to the H-form zeolite. The results of this analysis are presented in Figure 2.3.



Figure 2.3 BEA: DRIFT spectra of the OH region of the dehydrated 1.4%, 0.6%, 0.1% Pd-BEA and H-BEA materials after pretreatment at 500  $\degree$ C for 1 h with Ar rigorously dried by cold trap. The spectrum of H-BEA was used as the background for these spectra. Materials were cooled in dried Ar to 25 °C prior to collection. Spectra of H-BEA pretreated in air at 500 °C and 1.4% Pd-BEA treated in  $H_2$  at 500 °C are also shown.

Zeolite-exchanged transition metal OH bands are not well documented in the literature, though recent identification of a Cu(OH) band on Cu-SSZ-13 at 3660 cm<sup>-1</sup> provides some indication of where such species may occur.121 In fact, examination of the variously loaded Pd-BEA materials shows increasing intensity of a pair of OH bands at  $3604$  and  $3660 \text{ cm}^{-1}$  with increasing Pd loading, reaching maximum intensity on the airtreated 1.4% Pd-BEA material. These bands mirror the behavior of the bands at 2144 and  $2125 \text{ cm}^{-1}$ , providing evidence that these CO bands represent the CO stretch of a  $Pd^{2+}(\text{OH})(CO)$  complex, this species readily reducing to  $Pd^{+}(\text{CO})$  at elevated temperature.

To further confirm the identity of the OH bands, the spectra of the air pretreated H-BEA material and the H<sub>2</sub>-reduced 1.4% Pd-BEA material are also included. The putative Pd(OH) bands overlap both zeolite Brønsted acid and EFAl(OH) bands though the air-

treated H-BEA shows no analogous intensity increase in these areas, discounting the possibility that these bands result from changes in the zeolite OH distribution during pretreatment. Further, the  $H_2$ -reduced 1.4% Pd-BEA also fails to show these bands, thereby confirming their association with ion-exchanged Pd. The CO band assignments presented above provide a consistent explanation of the available literature results, the frequency shift observed upon TPD by Naccace et al. and Aylor et al. being explained by reduction of  $[{\rm Pd}^{2+}({\rm OH})]^+$  to  ${\rm Pd}^+$ , while the failure of Zheng et al. to observe this behavior can be explained by the absence of  $[{\rm Pd}^{2+}({\rm OH})]^+$ , the  ${\rm Pd}^{n+}({\rm CO})$  bands observed in that case resulting from "bare"  $Pd^{n+}$  ions in agreement with the conclusions of the authors.<sup>76,</sup>  $78,82$  These differences in speciation can be explained by the dependence of this parameter on the preparation methods employed in each case.

CO adsorption was also carried out on the 1% Pd-Si-BEA material to characterize the interaction of CO with  $PdO_x$  particles, shown in Figure 2.4. No sharp Pd(CO) bands appear, only the pair of broad features previously described on H-BEA that represent CO condensed in the zeolite pores and CO interaction with zeolite OH sites.



Figure 2.4 1% Pd-Si-BEA: DRIFT spectra of CO adsorption after pretreatment at 500 °C for 1 h with Ar which has been rigorously dried by cold trap. CO was adsorbed for 10 minutes at 25 °C.

To summarize,  $Pd^+$  and  $Pd^{2+}$  of intermediate electrophilicity fill the zeolite first (presumably at isolated and paired Al sites, respectively), followed by an increasing abundance of super-electrophilic  $Pd^{2+}$  with increasing Pd loading. Finally, oxidation of Pd<sup>+</sup> to the least electrophilic Pd<sup>2+</sup> ions (2154 cm<sup>-1</sup>) and  $[{\rm Pd}^{2+}({\rm OH})]$ <sup>+</sup> at isolated Al sites is observed under air treatment. The band at 2154 cm<sup>-1</sup> is likely representative of  $Pd^{2+}$  at NNN Al sites, this species possessing greater stability than its higher frequency counterparts (that would represent NNNN Al sites) as described in section 2.4. Meanwhile, isolated Al sites support  $Pd^+$  ions under inert pretreatment, these species being largely converted to  $[{\rm Pd^{2+}(OH)}]^+$  upon oxidizing treatment. The high intensity of the super-electrophilic  $Pd^{2+}(CO)_2$  bands is also notable, these bands being much weaker on BEA zeolites shown in the literature, while these species are shown to predominate on Pd-SSZ-13 of low  $Si/A1^{74,76}$  Of further note is the near-total absence of metallic Pd(CO) bands under rigorously dry conditions, an observation to be discussed extensively in section 2.3.

### 2.3 Comparison to Pd-CHA

The 0.7% Pd-CHA material was treated in Ar and compared to 1.4% Pd-BEA as shown in Figure 2.5, a listing of band assignments being presented in Table 2.3. Even though 0.7% Pd-CHA presents a much lower degree of Pd ion exchange, the Pd(CO) bands are much more intense than those observed on 1.4% Pd-BEA. This suggests either that the extinction coefficients of these bands differ substantially from framework to framework, or the lower abundance of differing sites leads to more intense bands on CHA. Using the band assignments for Pd-BEA and the similarity of CO band structure between the two materials, a set of assignments can be derived for 0.7% Pd-CHA. Again, the super-electrophilic  $Pd^{2+}$  bands are intense, though the symmetric/asymmetric stretch ratio differs from that observed on Pd-BEA. This, coupled with a slight shift of these bands to lower frequency, suggests the smaller CHA pore is imposing geometric constraints on this species that increase the extent of  $\pi$  back-bonding with CO. The number of putative NNNN and NNN Al-sited  $Pd^{2+}$  bands is lower than that observed on BEA, only a pair of bands being observed at 2175 and 2154 cm<sup>-1</sup> in this region. This is

reflective of the less complex CHA framework, a narrower distribution of these sites being available for exchange.

The low-frequency region of the 0.7% Pd-CHA spectrum also presents an intriguing feature. A very sharp band appears at 1890 cm<sup>-1</sup> representing (Pd<sup>0</sup>)<sub>2</sub>(CO) adsorbed on a group of Pd metal particles with a very narrow particle size range.<sup>119</sup> This band likely represents CO adsorbed at Pd metal particles trapped within the zeolite framework that consist of a very few Pd atoms as described by Sheu et al.<sup>84</sup> The appearance of such a species on CHA is again a result of the smaller CHA pore size, BEA apparently having less propensity to form such species.

Band $(cm^{-1})$	Assignment	Position	References
1790	$(Pd^{0})_{3}(CO)$	External $Pd^0$	[27, 28]
1890-1920	$(Pd^0)_2(CO)$	Pore-confined $Pd^0$	[27, 28]
1990	$(Pd^{0})_{2}(CO)$	External $Pd^0$	[27, 28]
2113	$Pd^+(CO)$	Isolated Al	[3, 16, 20, 27, 29]
2125	$Pd^{2+}(CO)(OH)$	Isolated Al	[3, 20]
2132	$Pd^+(CO)$	Isolated Al	[3, 20]
2142	$Pd^{2+}(CO)(OH)$	Isolated Al	[3, 16, 20, 29]
2154	$Pd^{2+}(CO)$	Paired Al	[3, 16]
2175	$Pd^{2+}(CO)$	Paired Al	[3, 16]
2188	Super-electrophilic $Pd^{2+}(CO)_2$	Pore-confined ion	[17, 30]
2212	Super-electrophilic $Pd^{2+}(CO)_{2}$	Pore-confined ion	[17, 30]

Table 2.3 Pd(CO) DRIFTS band assignments and references for CHA materials



Figure 2.5 DRIFT spectra of CO adsorbed on 0.7% Pd-CHA and 1.4% Pd-BEA pretreated at 500 °C for 1 h with Ar rigorously dried by cold trap. To visualize Pd speciation, 1000 ppm CO in Ar was adsorbed for 10 min at 25°C. Top: Ionic Pd range. Bottom: metallic Pd range.

 $Pd^{2+}$ -(OH)(CO) bands are also observed in conjunction with  $Pd^+(CO)$  bands, these occurring at frequencies nearly identical to those observed on Pd-BEA. The desorption behavior of 0.7% Pd-CHA was also analyzed, shown in Figure 2.6. Again, four distinct  $Pd^{n+}$ (CO) species appear in the range 2110–2145 cm<sup>-1</sup>, bands at 2113, 2125, 2132, and 2142 cm<sup>-1</sup> being observed at a temperature of 120 °C, the band maxima again shifting to lower frequency with increasing temperature. This behavior indicates a similar CO band

structure to that observed on 1.4% Pd-BEA, so the OH region of the CHA zeolite was also examined to determine the presence of any Pd(OH) bands. Shown in Figure 2.7, these data do not reveal the obvious presence of any Pd(OH) bands due to the greater intensity of the overlapping zeolite OH bands on the CHA materials, though this does not preclude the existence of  $[{\rm Pd}^{2+}({\rm OH})]^+$ .



Figure 2.6 0.7% Pd-CHA: DRIFT spectra of CO desorption after pretreatment at 500 °C for 1 h with rigorously dried Ar. CO was adsorbed for 10 minutes at 25 °C before the material was purged for 10 minutes in Ar prior to ramping to remove gas phase CO. Temperature was ramped at 10 °C/min to 500 °C with spectra being collected every 20 °C.



Figure 2.7 DRIFT spectra of the OH region of the dehydrated 0.7% Pd-CHA materials after pretreatment at 500 °C for 1 h with Ar rigorously dried by cold trap. The spectrum of H-CHA was used as the background for these spectra. Materials were cooled in dried Ar to 25  $\degree$ C prior to collection. The spectrum of 0.7% Pd-CHA treated in  $H_2$  at 500 °C are also shown.

#### 2.4 Effect of Water During Pretreatment

As described above, the Pd speciation of Pd-loaded zeolite materials is sensitive to the pretreatment conditions applied. The presence of water during high temperature treatment was shown to enhance Pd dispersion of materials with high concentrations of PdO<sub>x</sub> particles.<sup>98</sup> However, the effect of water on ion-exchanged Pd is less completely understood, though the behavior of metal migration during hydrothermal treatment has been shown to be dependent on the framework structure and pore size of the material.<sup>59,</sup>  $122$  The effect of water on the FT-IR spectrum of oxide materials that contain aluminum (such as zeolites and alumina) further complicates this analysis, as water association with zeolite OH sites leads to broad spectral intensity shifts over much of the IR range. This phenomenon results in the appearance of pseudo-maxima that can easily be misinterpreted as real spectral features.<sup>74, 123</sup> To provide example and comparison for later spectra, the time-resolved adsorption of water over 1.4% Pd-BEA and 0.7% Pd-CHA is shown in Figure 2.8. These fingerprints will be used to identify water-related pseudomaxima throughout this work. It is especially notable that water pseudo-maxima appear

in the NO and CO stretching regions, and even trace quantities of water present in the gas feed leads to the appearance of these features. This is a further indication of the importance of rigorous gas drying in deconvoluting the effects of water from other adsorbates.



Figure 2.8 1.4% Pd-BEA (top), 0.7% Pd-CHA (bottom): DRIFT spectra of water adsorption after pretreatment at 500 °C for 1 h with Ar rigorously dried by cold trap. The materials were cooled in dried Ar to 25 °C prior to adsorption. 1-2% water in Ar was adsorbed for 10 min, Ar being wetted by room temperature bubbler.

#### 2.4.1 1.4 % Pd-BEA

To assess the impact of water on Pd speciation, 1.4% Pd-BEA was subjected to treatment at 500 °C in either Ar or air with varying amounts of water present in the feed gas, followed by CO adsorption at 25 °C (Figure 2.9). The DRIFT spectrum obtained after treatment of 1.4% Pd-BEA in dried Ar shows a predominance of  $Pd^+$  and superelectrophilic  $Pd^{2+}$ , while use of untreated Ar (i.e., Ar fed directly from the gas cylinder) results in a loss of super-electrophilic  $Pd^{2+}$  and  $[Pd^{2+}(\text{OH})]^{+}$  in favor of  $Pd^{+}$  and  $Pd^{0}$ . This trend continues for the material treated in wetted Ar, the only remaining  $Pd^{n+}(\text{CO})$  bands appearing at 2154, 2130, and 2119 cm<sup>-1</sup>, with total elimination of  $[{\rm Pd^{2+}(OH)}]^{+}$  and superelectrophilic Pd<sup>2+</sup>. Metallic Pd<sup>0</sup>(CO) bands are largely absent on the material pretreated in dried Ar. The appearance of  $Pd<sup>0</sup>$  after treatments containing water can be interpreted as the reduction of ion-exchanged Pd to  $Pd<sup>0</sup>$  particles. Notably, the least electrophilic of the  $Pd^{2+}$  species (2154 cm<sup>-1</sup>) is more resilient to water exposure than the other four divalent Pd species. The super-electrophilic bands are totally absent after exposure to water, an indication that these species have undergone reduction to  $Pd<sup>0</sup>$  particles as described in Equation 8.

$$
Pd(II) - (Z)_2 + H_2O \rightarrow Pd(0) + 2H^+Z^- + \frac{1}{2}O_2
$$
 (8)

The fact that the least electrophilic divalent Pd is less susceptible to reduction by water suggests that this process occurs with Pd acting as electrophile and water acting as a nucleophile, though the intricacies of this mechanism remain to be determined. These data also indicate that this Pd is in a substantially different environment than its counterparts, i.e. NNN Al sites vs NNNN Al sites. Additionally, since this reaction produces  $O_2$ , the equilibrium must lie to the right given that the presence of  $O_2$  does not prevent water from reducing Pd (see Figure 2.9). This mechanism also implies the need for Pd site proximity, since two Pd atoms would need to be reduced to produce one  $O_2$ molecule. It is possible that  $PdO_x$  particles are also reduced to  $Pd^0$  by water, though their role cannot be readily ascertained from these data, due to their lack of CO adsorption. However, the role of  $PdO_x$  particles will be discussed more thoroughly in the next section.



Figure 2.9 DRIFT spectra of CO adsorption onto 1.4% Pd-BEA after pretreatment at 500 °C for 1 h with Ar (top) or air (bottom) wetted by bubbler, untreated, or rigorously dried by cold trap. The material was cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C.

The results of pretreatment in air are also shown, the behavior being somewhat different while also reinforcing the above conclusions. The material pretreated in dried air reveals an abundance of  $Pd^{2+}$ ,  $[Pd^{2+}(\text{OH})]^{+}$  and super-electrophilic  $Pd^{2+}$  as well as lesser quantities of  $Pd^+$ . Untreated air produces a broad band centered around 2144 cm<sup>-1</sup> that consists primarily of  $[Pd^{2+}(OH)]^+$ , though this feature also includes contributions in

the range of Pd<sup>2+</sup> (2154 cm<sup>-1</sup>) and Pd<sup>+</sup> (2132 cm<sup>-1</sup>). Super-electrophilic Pd<sup>2+</sup> is absent here, suggesting a greater concentration of trace water in the air cylinder and thus providing an explanation for the appearance of weak  $Pd^0(CO)$  bands in this spectrum while the majority of other Pd is maintained in the divalent state. The spectrum of the material pretreated in wetted air closely resembles the spectrum of the material treated in wetted Ar, a result indicating that while  $O_2$  partial pressure exerts an influence at low water partial pressure by oxidizing Pd<sup>+</sup>, this effect is eventually negated at higher water pressures.

The corresponding spectra of the low-frequency metallic  $(Pd^0)_x(CO)$  bands are shown in Figure 2.10. Both of the wetted feed gases lead to the formation of a strong  $(Pd^{0})_{2}(CO)$  band at 1950 cm<sup>-1</sup> while the dried and untreated spectra show only very weak bands at 1950, 1990, and 1790  $cm^{-1}$ . The band at 1950  $cm^{-1}$  is presumably a species analogous to that observed at  $1890 \text{ cm}^{-1}$  on Pd-CHA, representing CO adsorbed on a pore-confined Pd metal particle. The higher wavenumber of this band on 1.4% Pd-BEA is an effect of the greater BEA pore diameter, allowing a greater number of Pd atoms to agglomerate before becoming trapped, thus shifting the frequency of adsorbed CO to higher wavenumbers.<sup>119</sup> The band is also broader than that observed on Pd-CHA, this again being an effect of the framework structure, as BEA zeolite has irregular channels with a greater diversity of pore diameters that can support a greater diversity of Pd particle sizes. It can therefore be inferred that the band at  $1990 \text{ cm}^{-1}$  represents CO adsorbed on bulk Pd particles outside the framework, as this band appears on both zeolites examined here and on other materials in literature reports.<sup>82, 124</sup> The appearance of small quantities of doubly and triply bridging CO on the material treated under dry conditions can be explained by the work of Vannace et al. that shows the extinction coefficients of metallic Pd(CO) bands increase with decreasing frequency, indicating small populations of  $(Pd^0)_2(CO)$  and  $(Pd^0)_3(CO)$  can be apparent even when  $Pd^0(CO)$ bands are absent.<sup>125</sup>

42



Figure 2.10 DRIFT spectra of CO adsorption onto 1.4% Pd-BEA after pretreatment at 500 °C for 1 h with Ar (top) or air (bottom) wetted by bubbler, untreated, or rigorously dried by cold trap. The material was cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C. Spectra correspond to those presented in Figure 2.9.

#### 2.4.2 0.7 % Pd-CHA

Similar experiments were performed on 0.7% Pd-CHA, the results being shown in Figure 2.11. The Pd speciation of this material is much more stable in the presence of water at high temperatures, even the super-electrophilic  $Pd^{2+}$  remaining after treatment in wetted Ar. This agrees with the result of Khivantsev et al., who have previously reported the stability of this species on Pd-SSZ-13 in the presence of water at room temperature.<sup>74</sup> This is a remarkable result as the  $[Pd(CO)_4][Sb_2F_{11}]_2$  complex that exhibits CO bands similar to this species readily undergoes reduction upon exposure to water as observed for Pd-BEA.<sup>80</sup> Nonetheless, the material pretreated in wetted Ar does show a decrease in the abundance of  $[{\rm Pd}^{2+}({\rm OH})]^+$  and super-electrophilic  ${\rm Pd}^{2+}$ , while  ${\rm Pd}^+$  species grow more abundant along with the appearance of a shoulder at 2090  $cm^{-1}$  corresponding to CO adsorbed at metallic Pd. The low-frequency region also shows an increase in the intensity of the band corresponding to pore-confined  $(Pd^{0})_{2}(CO)$  again indicating reduction by water leads primarily to the production of small, pore-confined Pd metal particles. Other studies have shown the preferential production of such particles over larger particles outside the framework can lead to improved Pd dispersion upon re-oxidation.<sup>59, 107</sup> The spectra of 0.7% Pd-CHA treated in air with various concentrations of water (not shown) are practically identical to those of the material pretreated in Ar, another indication that the Pd exchanged onto this material is more resilient to reduction.



Figure 2.11 DRIFT spectra of CO adsorption onto 0.7% Pd-CHA after pretreatment at 500 °C for 1 h with Ar wetted by bubbler, untreated, or rigorously dried by cold trap. The material was cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C. Top: Ionic Pd range. Bottom: metallic Pd range.

Whether or not  $PdO_x$  particles can be reduced by water is difficult to ascertain, though the higher concentration of these particles on 0.7% Pd-CHA means reduction of  $PdO<sub>x</sub>$  on this material should be more obvious. To further understand the reducing effects of water on Pd-CHA, the material treated in wet air was subsequently treated in dry air at 500 °C for 1 h prior to CO adsorption to ascertain the degree of reversibility of water

reduction. The comparison of this spectrum with the CO adsorption spectrum of the asprepared material pretreated in air is shown in Figure 2.12. Remarkably, full restoration of ionic Pd is observed upon re-oxidation in dried air along with a slight increase in the abundance of super-electrophilic  $Pd^{2+}$  over the as-prepared material. The low-frequency region is also notable as the abundance of pore-confined Pd particles increases as well, along with the appearance of a shoulder below  $1890 \text{ cm}^{-1}$  that indicates the presence of still smaller Pd particles. The increased intensity of both ionic and metallic Pd(CO) bands after re-oxidation suggests that some  $PdO_x$  particles have been reduced to Pd metal by water and then are either re-distributed as ion-exchanged Pd cations or attain sufficient particle size to become trapped and stabilized by the zeolite pore. The  $PdO_x$  particles susceptible to reduction in water likely consist of only the smallest particles in the  $PdO<sub>x</sub>$ population. It is notable that this result is similar to the effects produced by treatment in  $H_2$ , PdO<sub>x</sub> reduction being proposed to occur through a pathway that also yields primarily small Pd particles. 82, 103, 108, 122



Figure 2.12 0.6% Pd-CHA: DRIFT spectra of CO adsorption after sequential pretreatment at 500 °C for 1 h with dried air, air wetted by bubbler, and finally dried air. Only the as-prepared and re-oxidized spectra are shown for clarity. The material was cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C between each pretreatment. Top: ionic Pd range. Bottom: metallic Pd range.

The above results show the negative impact of water on ion-exchanged Pd, even trace quantities of water being demonstrated to reduce cationic Pd on the larger pore BEA zeolite. The super-electrophilic  $Pd^{2+}$  species is shown to be especially susceptible to the presence of water on 1.4% Pd-BEA, while this species is stabilized to a remarkable degree by the smaller CHA pore structure. Further, reduction by water is shown to lead to the preferential formation of small Pd metal particles confined within the zeolite framework, a positive result that indicates much of the reduced Pd can be recovered by re-oxidation. The reduction of Pd by water is shown to be sensitive to the electrophilicity of  $Pd^{2+}$ , the least electrophilic of these species being more resilient to reduction under these conditions. This reaction is also shown to be insensitive to the presence of oxygen above a certain concentration of water, though effects are observable when only trace quantities of water are applied. Water is also proposed to lead to the reduction of small  $PdO<sub>x</sub>$  particles on 0.7% Pd-CHA, the increased CO band intensity after re-oxidation resulting from the re-distribution of these particles as both ion-exchanged Pd and as Pd metal particles that grew large enough to be stabilized by the zeolite pores. Overall, the Pd speciation of the CHA material is much more stable than that of BEA, a behavior again attributable to the smaller CHA pores.

#### 2.5 Effect of Water at Near-ambient Temperature

The observed sensitivity of 1.4% Pd-BEA to water demonstrates the need for evaluation of these reductive processes at temperatures relevant to PNA behavior. 1.4% Pd-BEA pretreated in dried Ar was subjected to sequential adsorption of CO and water to gain insight into the behavior of these adsorbates within this temperature regime. Figure 2.13 shows the results of adsorption of CO followed by adsorption of water for 10 min each in a balance of Ar. When CO is cut off and water was supplied, water pseudomaxima appear immediately, and CO bands begin to diminish in intensity. Interestingly, evidence of  $Pd<sup>+</sup>(CO)$  remains even after 10 minutes of water exposure, suggesting that CO is bound more strongly to  $Pd^+$  than  $Pd^{2+}$ , a supposition supported by the higher CO desorption temperature of these species as described above. The super-electrophilic  $Pd^{2+}(CO)$ <sub>2</sub> bands disappear within five minutes, in conjunction with the appearance of a  $(Pd^{0})_{2}(CO)$  band at 1950 cm<sup>-1</sup> that reaches its maximum intensity at about the same time.

This indicates that the reduction of super-electrophilic  $Pd^{2+}$  by water proceeds even at 25 °C on Pd-BEA, a result that serves to explain the high degree of variability in the intensity of these species in literature reports.<sup>76, 82, 103</sup>

The opposite experiment was also carried out; Figure 2.14 shows the effect of saturating the material with water prior to CO adsorption. In this case, the CO band structure is extremely weak and so the spectrum of the water pseudo-maxima was used as the background. This results in considerable spectral noise as water begins to desorb, though  $Pd^{n+}(CO)$  bands can still be resolved. Under these conditions,  $Pd^{2+}(OH)(CO)$ bands are observed, suggesting that in the previous experiment, water merely displaces CO from these species rather than acting to reduce  $[{\rm Pd}^{2+}(\rm OH)]^+$  to Pd<sup>+</sup>. The Pd<sup>2+</sup>(CO)<sub>2</sub> bands do not appear in this spectrum, confirming that the reduction of this species by water occurs even when this cation is not ligated by CO. The  $(Pd^0)_2$ (CO) band at 1950  $cm<sup>-1</sup>$  also appears in this spectrum, though examination of its behavior over time is complicated by an overlapping water pseudo-maximum.

The observation of differing CO adsorption behavior depending on the order in which adsorbates are applied is intriguing as this suggests the energies of adsorption for these two adsorbates are similar. The observation of the interaction of CO with  $[Pd^{2+}(OH)]$ <sup>+</sup> on the water saturated material is also notable, these species being suggested to be a key adsorption site in low-temperature PNA behavior.<sup>59, 90, 107</sup>



Figure 2.13 DRIFT spectra of 1.4% Pd-BEA pretreated at 500 °C for 1 h in Ar then cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min followed by adsorption of 1-2% water in Ar wetted by bubbler. Top: Ionic Pd range. Bottom: metallic Pd range.



Figure 2.14 DRIFT spectra of CO adsorption onto 1.4% Pd-BEA for 10 min after water adsorption for 10 min. The material was pretreated at 500 °C for 1 h in Ar then cooled in dried Ar to 25  $\degree$ C prior to adsorption. 1-2% water in Ar wetted by bubbler was adsorbed for 10 min followed by 1000 ppm CO in Ar. Top: Ionic Pd range. Bottom: metallic Pd range.

## 2.6 Conclusions

This chapter has focused on the development of a robust, evidence-based set of Pd(CO) band assignments for both Pd-BEA and Pd-CHA that serves to reconcile many of the disparate behaviors observed on Pd-loaded zeolites in other studies. The effect of Pd loading on BEA was examined, low Pd loadings and pretreatment in Ar being shown to favor Pd distribution as  $Pd^+$  while higher loadings and pretreatment in air leads to formation of greater quantities of  $Pd^{2+}$ , including a super-electrophilic  $Pd^{2+}$  species with CO band frequencies of  $\sim$ 2200 cm<sup>-1</sup>. The presence of  $[{\rm Pd}^{2+}({\rm OH})]^+$  was ascertained through the observation of Pd(OH) bands under various conditions, these species being behaviorally associated with CO bands that indicate formation of a  $Pd^{2+}$ -(OH)(CO) complex. This species is reduced to  $Pd<sup>+</sup>(CO)$  at elevated temperatures, while Pd metal particles are also formed. The reducing effect of water on Pd speciation has been examined, 0.7% Pd-CHA shown to be more resilient than 1.4% Pd-BEA to reduction by water at high temperature. The observed sensitivity of Pd speciation to water concentration emphasizes the need to perform DRIFTS measurements under rigorously water-free conditions, these observations explaining some of the discrepancies observed in literature results. This reductive process appears to operate with Pd acting as an electrophile, as the extent of reduction by water is sensitive to the electrophilicity of the Pd being reduced, the most electrophilic Pd species present on Pd-BEA being reducible even at room temperature. The reduction of  $PdO_x$  particles by water was also proposed to occur, these re-distributing to ion-exchanged Pd or as particles that grew large enough to be trapped by the zeolite pore. These results are nonetheless promising, as the formation of small Pd metal particles has been found to be relatively benign to the long-term performance of these materials as such particles are readily re-dispersed.<sup>59, 107</sup> In fact, greater ion exchange can be achieved after such treatment when a large population of  $PdO<sub>x</sub>$  particles is present, these species also being reduced and re-dispersed to ionexchanged Pd,  $Pd^0$ , and  $PdO_x$  particles.

#### CHAPTER 3. EFFECT OF CO AND  $H_2$  ON Pd SPECIATION

Given the sensitivity of the Pd speciation of these materials to water, the impact of more strongly reducing species present in the automotive exhaust stream presents a substantial impediment to the successful real-world implementation of PNA materials, especially under stoichiometric operation.  $H_2$  and CO have already been demonstrated by Ryou et al. to have differing effects on Pd-loaded zeolites;  $H_2$  preferentially forms small Pd particles due to diffusional limitations within the zeolite pore,<sup>108</sup> while CO is proposed to facilitate the mobility of Pd species through the formation of more mobile Pd-carbonyl species that are then deposited as large particles outside the framework, these external Pd particles being more susceptible to sintering.<sup>108</sup> The irreversible nature of this degradation pathway was further confirmed by Gu et al., reduction in CO leading to loss of  $NO<sub>x</sub>$ storage capacity that could not be regenerated even by extended exposure to  $\mathrm{NO_2.^{107}}$ These studies support their conclusions with STEM analysis, this technique being able to resolve pore-confined Pd particles. However, no thorough analysis of this behavior by DRIFTS is present in the literature, though this method presents a convenient probe of Pd metal particle size by CO adsorption. Below, the reduction and re-oxidation behavior of both Pd-loaded zeolite materials is characterized by CO-DRIFTS, reduction in  $H_2$  and CO being considered separately.

## 3.1 Effect of Reduction in CO Followed by Re-oxidation in Air

## 3.1.1 1.4 % Pd-BEA

1.4% Pd-BEA was treated in 1000 ppm CO in Ar for 1 h at 500  $\degree$ C, followed by re-oxidation in air at 500 °C. The original state of the material was assessed by pretreatment in air followed by CO adsorption at 25 °C, this spectrum provided as reference. Shown in Figure 3.1, treatment in CO leads to near-total loss of ionic Pd, though the fact that any remains is somewhat surprising. This result is consistent, however, with the results of Gu et al. who report some populations of ionic Pd are less susceptible to reduction by CO than others, though the identity of this species is not made

clear in that study.<sup>107</sup> The ionic Pd observed after reduction consists primarily of Pd<sup>+</sup> (2119 and 2130 cm<sup>-1</sup>) and the least electrophilic  $Pd^{2+}$  (2154 cm<sup>-1</sup>) species, along with a population of Pd metal particles. Meanwhile the super-electrophilic  $Pd^{2+}$  species is almost totally absent, this again being especially sensitive to reduction. Re-oxidation leads to a recovery of some ion-exchanged Pd, primarily in the form of  $Pd^+$  and the least electrophilic of the Pd<sup>2+</sup> species, while little evidence of  $[{\rm Pd}^{2+}(\rm OH)]^+$  is observed. The super-electrophilic  $Pd^{2+}$  also recovers to some degree, though it is still much less abundant than on the as-prepared material. This also serves as a further confirmation of the identification of CO bands corresponding to  $[{\rm Pd}^{2+}({\rm OH})]^+$  as these species are readily reduced by CO at elevated temperatures, producing a CO-DRIFT spectrum much like that of the Ar pretreated material upon re-oxidation. With this information it can be inferred that the irreversible degradation of  $NO<sub>x</sub>$  adsorption behavior by CO on Pd-BEA can be at least partially explained by the reduction of  $[{\rm Pd}^{2+}({\rm OH})]^+$ , this species failing to regenerate upon re-oxidation in the absence of water. The low-frequency region is also informative, the as-prepared material showing a population of external Pd-metal particles identifiable by CO bands at 1990 and 1790  $cm^{-1}$ . These bands increase in intensity on the CO-reduced material, accompanied by the appearance of a band at  $1950 \text{ cm}^{-1}$  that corresponds to CO adsorbed onto pore-confined Pd particles. After re-oxidation, only water pseudo maxima are observed in this region, suggesting much of the metallic Pd either sintered, was re-distributed as ion-exchanged Pd, or formed  $PdO_x$  particles. Given the complex nature of the Pd speciation of these materials it is more likely that some or all these transformations are active under the conditions employed here.



Figure 3.1 DRIFT spectra of CO adsorption onto 1.4% Pd-BEA after sequential pretreatment at 500 °C for 1 h with dried air, 1000 ppm CO in Ar, and finally dried air. The material was cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C between each pretreatment. Top: ionic Pd range. Bottom: metallic Pd range.

## 3.1.2 0.7 % Pd-CHA

Similar experiments performed on 0.7% Pd-CHA reveal a distinctly different behavior; reduction in CO still leads to the loss of most ion-exchanged Pd, though the species that remain differ as shown in Figure 3.2. In this case,  $[Pd^{2+}(\text{OH})]^{+}$  remains after CO reduction while  $Pd^+$  and un-ligated  $Pd^{2+}$  species are lost. This result indicates that the reactivity of the various Pd species to CO is different on CHA, favoring the reduction of  $Pd^+$  and  $Pd^{2+}$  to Pd metal particles, while  $[Pd^{2+}(\text{OH})]^+$  seems to be more resilient, the opposite of the behavior observed on BEA. The mechanisms through which these differing processes occur is not made clear by these data, since quantitation of IR bands is not possible and the full role of  $PdO_x$  particles cannot be established. Re-oxidation leads to almost complete regeneration of the super-electrophilic  $Pd^{2+}$  species, while all the other ionic Pd species are recovered somewhat, though they fail to approach their original intensity.

The low-frequency IR region shows the presence of primarily bulk Pd metal particles outside the framework after CO reduction, corresponding CO bands observed at 1990 and 1790 cm<sup>-1</sup> for  $(Pd^{0})_{2}(CO)$  and  $(Pd^{0})_{3}(CO)$ , respectively. This result agrees with the results from Pd-BEA and the work of Gu et al. and Ryou et al. in that CO leads to greater Pd mobility and thus enhances agglomeration of large particles outside the zeolite framework.107, 108 It is notable that the pore-confined Pd particles already present on the material are also lost during CO reduction, suggesting these species can also be rendered mobile in the presence of CO. Further, the fact that Pd(CO) bands fail to recover to their original intensity also points to two potential fates for  $PdO_x$  particles during this process, one being that these particles fail to reduce, or that they reduce completely to large metallic Pd particles and sinter, either effect leading to a failure of these particles to redistribute as observed for reduction by water.


Figure 3.2 DRIFT spectra of CO adsorption onto 0.7% Pd-CHA after sequential pretreatment at 500 °C for 1 h with dried air, 1000 ppm CO in Ar, and finally dried air. The material was cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C between each pretreatment. Top: ionic Pd range. Bottom: metallic Pd range.

# 3.2 Effect of Reduction in  $H_2$  followed by Re-oxidation in Air

# 3.2.1 1.4 % Pd-BEA

Results from a series of experiments analogous to those presented in the previous section using  $H_2$  as reductant on 1.4% Pd-BEA are shown in Figure 3.3. The  $H_2$ -reduced material shows a more complete loss of ionic Pd than was observed for reduction in CO, though evidently some  $Pd^+$  remains, possibly as  $Pd^+$  on the surface of Pd metal particles as described by Hess et al.<sup>93</sup> The predominant feature in the ionic Pd frequency range falls at about 2100 cm<sup>-1</sup> corresponding to  $Pd^0(CO)$  species. A weak feature occurs at  $2186 \text{ cm}^{-1}$ , though this band corresponds to CO adsorbed at a zeolite Lewis acid site, this species being masked by Pd(CO) bands under other circumstances.

Re-oxidation leads to the restoration of  $Pd^+$ , while  $[Pd^{2+}(\text{OH})]^{+}$  and superelectrophilic  $Pd^{2+}$  are restored to a lesser extent. Overall, this treatment leads to better recovery of ion-exchanged Pd than is observed after CO reduction, though the lower abundance of  $PdO_x$  particles on BEA prevents improvement of dispersion on reoxidation. The recovery of  $[{\rm Pd}^{2+}({\rm OH})]^+$  after H<sub>2</sub> reduction further indicates that this species is responsible for some of the irreversible degradation of Pd dispersion observed during CO reduction as it fails to re-appear in that case. The low-frequency region is also revealing, bands appearing at 1990 and 1950  $cm^{-1}$  being representative of bulk and poreconfined Pd metal particles, respectively. Especially notable is the total absence of a band at 1790 cm<sup>-1</sup> on the H<sub>2</sub>-reduced material. This band represents  $(Pd^0)_{3}(CO)$  and appears at an especially low frequency for such a species, an indication that this CO is highly coordinated to Pd and may represent a site where a single Pd atom has been lost to an Ostwald ripening-type process.<sup>82, 126</sup>



Figure 3.3 DRIFT spectra of CO adsorption onto 1.4% Pd-BEA after sequential pretreatment at 500 °C for 1 h with dried air, 10% H<sub>2</sub> in Ar, and finally dried air. The material was cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C between each pretreatment. Top: ionic Pd range. Bottom: metallic Pd range.

# 3.2.2 0.7 % Pd-CHA

The reduction and re-oxidation of  $0.7\%$  Pd-CHA in  $H_2$  is shown in Figure 3.4, again showing a differing behavior from the material reduced in CO. The  $H_2$ -reduced spectrum presents a more complete loss of ion-exchanged Pd relative to the CO-reduced material, only a broad feature centered around 2100 cm<sup>-1</sup> being in evidence. Again, CO adsorption at a zeolite Lewis acid site is observed, this time at  $2194 \text{ cm}^{-1}$ . The results of re-oxidation are more remarkable; the super-electrophilic  $Pd^{2+}$  is restored to its original intensity while the other ion-exchanged Pd species show a significant increase in intensity over the asprepared material. This result is similar to that observed upon re-oxidation from water treatment and provides evidence that  $H_2$  successfully reduces  $PdO_x$  particles into a form that can be re-dispersed to ion-exchanged Pd upon re-oxidation, unlike CO.

The behavior observed in the low-frequency region corroborates these results, the H2-reduced material revealing the presence of a distribution of particles at and above the size of the pore-confined Pd particles, though below the bulk Pd particle frequency. Only a weak band representing bulk  $(Pd^{0})_{2}(CO)$  occurs at 1990 cm<sup>-1</sup>. This result indicates that pore-confined Pd particles are mobilized during  $H_2$  reduction, though they are still largely prevented from attaining the size of external Pd particles large enough to exhibit bulk  $Pd^0$ behavior. Again, the  $(Pd^0)_{3}(CO)$  band at 1790 cm<sup>-1</sup> fails to appear. The increased Pd dispersion observed on re-oxidation of this material is further evidence that the CO adsorption site responsible for the  $1790 \text{ cm}^{-1}$  band is indicative of the agglomeration of Pd to large, external, Pd particles. Accordingly, this band fails to appear when sintering is limited, or in the absence of the  $(Pd^{0})_{2}(CO)$  band at 1990 cm<sup>-1</sup> that is also representative of the presence of such particles. Re-oxidation again reveals an increase in the abundance of pore-confined Pd metal particles over the as-prepared material, though this effect is much more substantial than that observed when water was employed as reductant. This indicates reduction of  $PdO_x$  particles to small Pd metal particles that are then accessible for re-dispersion. Upon re-oxidation these small particles presumably begin to shrink and migrate, producing a large population of still smaller Pd particles that can be readily trapped by the zeolite pores or return to ion-exchange sites as cationic species.



Figure 3.4 DRIFT spectra of CO adsorption onto 0.7% Pd-CHA after sequential pretreatment at 500 °C for 1 h with dried air, 10%  $H_2$  in Ar, and finally dried air. The material was cooled in dried Ar to 25 °C prior to adsorption. 1000 ppm CO in Ar was adsorbed for 10 min at 25 °C between each pretreatment. Top: ionic Pd range. Bottom: metallic Pd range.

# 3.2.3 Other Experimental Methods

To corroborate the DRIFTS data presented above, XPS and microreactor-MS studies were employed to probe Pd speciation and  $NO<sub>x</sub>$  adsorption/desorption behavior, respectively. Near-ambient pressure XPS was performed on 1.4% Pd-BEA after various pretreatments mirroring those performed in the DRIFTS studies. The atomic concentrations measured at the material surface are presented in Table 3.1, while the deconvoluted spectra are present in Figure S1 (supplementary data are included in Appendix 3). It is immediately apparent from the data that the amount of Pd observed is consistently less than 10 % of the total Pd present, while metallic Pd seems to be overrepresented. For instance, only very weak metallic Pd(CO) bands were detected on airpretreated 1.4% Pd-BEA and yet metallic Pd accounts for nearly one quarter of Pd observed by XPS under similar conditions. This indicates large, external Pd particles exert an outsized influence when this method is applied, ion-exchanged Pd being segregated into the zeolite crystallite and thus less visible to XPS. These data nonetheless serve to confirm that substantial differences exist between the material pretreated in Ar and air at 500 °C, these treatments showing 24.0 % and 77.5 % of Pd present in ionic form, respectively.

A Pd  $3d_{5/2}$  band corresponding to super-electrophilic  $Pd^{2+}$  is also observed at 339.7 eV, this result being consistent with the reports of Khivantsev et al. who showed the presence of an identical band on a Pd-SSZ-13 material.<sup>74</sup> This band has been previously observed on Pd-BEA, though it was then attributed to  $Pd^{4+}$  ions due to the extremely high binding energy of this species.<sup>76</sup> This feature is absent from the BEA material pretreated at 750 °C, suggesting that super-electrophilic  $Pd^{2+}$  either decomposes, becomes more sensitive to trace water, or migrates away from the surface at this temperature.  $H_2$ reduction leads to the formation of primarily metallic Pd, though a small amount of ionic Pd remains, corroborating the DRIFTS observation of  $Pd^+$  after  $H_2$  reduction, again likely present on the surface of Pd metal particles.<sup>93</sup> This conclusion is supported by the observed maximum binding energy of the ionic Pd band shifting from 337.35 eV to 336.84 eV, suggesting the average oxidation state of this population of Pd has decreased.

62

Pd-BEA			carbon oxygen aluminum silicon palladium			Oxidized Pd $%$	Metallic Pd $%$
As-Prepared 4.8		64.9	2.0	28.3	0.08	36.7	63.3
$500$ °C air	0.4	66.1	2.0	31.4	0.09	77.5	22.6
$750 °C$ air	1.6	66.1	2.1	30.1	0.10	59.2	40.8
$500 °C$ Ar	3.9	63.9	1.9	30.2	0.07	24.0	76.1
500 °C $\rm H_2$	7.4	61.5	1.9	29.1	0.09	16.2	83.8

Table 3.1 Atomic concentrations (%) as determined by XPS. Oxidized and metallic concentrations are derived from deconvolution of the Pd 3d spectra.

To examine the  $NO<sub>x</sub>$  adsorption and desorption behavior in a semi-quantitative manner, microreactor-MS experiments were performed, these procedures mirroring those of the DRIFTS experiments except that the flow rate was maintained at 100 sccm due to system requirements. In each case, 1000 ppm NO in 10 %  $O_2$  in He was adsorbed at 50 °C for 10 min followed by a purge for 1 h to remove weakly adsorbed species. This was followed by a temperature ramp at 10  $^{\circ}$ C per minute to 500  $^{\circ}$ C. The desorption of NO from both 1.4 % Pd-BEA and 0.7% Pd-CHA after  $H_2$  reduction and subsequent reoxidation is shown in Figure 3.5, the  $NO<sub>x</sub>$  concentration in the gas phase being monitored by a calibrated MS signal. The 1.4 % Pd-BEA material shows a dramatic loss of  $NO<sub>x</sub>$ capacity after reduction in  $H_2$ . Pd metal particles not only fail to adsorb  $NO_x$ , but also serve to block  $NO<sub>x</sub>$  adsorption at zeolite sites, the same being true of 0.7 % Pd-CHA as  $NO<sub>x</sub>$  desorption appears to be less intense from all adsorption sites when compared with the H-form materials (see section 4.4 ). For 1.4% Pd-BEA, re-oxidation leads to a slight enhancement of NO<sub>x</sub> storage over the as-prepared material ( $\sim$ 36%), a result that suggests the small quantity of  $PdO_x$  present on this material has been re-dispersed as species active for NO<sub>x</sub> storage. These data also suggest that the super-electrophilic  $Pd^{2+}$  species and some of the other  $Pd^{2+}$  species observed on Pd-BEA do not participate in NO<sub>x</sub> adsorption, as these are largely lost after re-oxidation as shown in Figure 3.3. As a result it can be inferred that Pd<sup>+</sup> and  $[Pd^{2+}(OH)]$ <sup>+</sup> are the primary participants in NO<sub>x</sub> storage on BEA, these species being regenerated preferentially, as observed by DRIFTS.



Figure 3.5 1.4% Pd-BEA (top), 0.7% Pd-CHA (bottom).  $NO<sub>x</sub>$  TPD conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$  concentrations. Both materials were pretreated in  $10\%$  O<sub>2</sub> in He,  $10\%$  H<sub>2</sub> in He, and finally  $10\%$  O<sub>2</sub> in He. 1000 ppm NO in 10% O<sub>2</sub>/He was adsorbed for 10 min at 50 °C followed by a 1 h purge in  $10\%O_2$ /He. TPD was carried out in 10% O<sub>2</sub>/He at a ramp rate of 10 °C per min.

The 0.7% Pd-CHA material again presents a dramatically different result upon reoxidation, a substantial improvement in the  $NO<sub>x</sub>$  storage capacity being observed, along with a shift of desorption to higher temperatures that are more optimal for PNA behavior. This agrees with the DRIFTS observation that Pd-dispersion is improved upon reduction

and re-oxidation in H2, though the reason for the disappearance of the low-temperature desorption event associated with zeolite sites is less clear. These data do suggest preferential formation of a site that desorbs  $NO<sub>x</sub>$  at higher temperatures, comparison with DRIFTS results indicating that  $Pd^+$  or  $[Pd^{2+}(OH)]^+$  is likely responsible for this behavior due to the increased intensity of these bands upon re-oxidation. A potential explanation for the disappearance of the low-temperature desorption event could be a global change in the zeolite acid-base properties resulting from greater Pd dispersion, similar to that observed by Bisio et al. when  $Cs<sup>+</sup>$  was exchanged onto BEA zeolites, though an extensive computer modeling study would be required to confirm this supposition.<sup>88</sup>

# 3.3 Conclusions

To summarize the results of this chapter, the negative effects of CO on these materials has been demonstrated, CO particle agglomeration and sintering observed in previous studies being supported by in-situ DRIFTS. After re-oxidation from CO reduction,  $Pd^{n+}(CO)$  bands fail to return to their original intensity, suggesting decreased abundance of these species relative to the as-prepared material.  $H_2$  reduction is found to be more benign, producing primarily small, pore-confined metal particles that are readily re-distributed upon re-oxidation as ion-exchanged Pd, better recovery and even improvement of Pd dispersion on re-oxidation being observed. The role of  $PdO<sub>x</sub>$  particles was indirectly explored, re-oxidation of 0.7% Pd-CHA showing more intense Pd(CO) bands than the as-prepared material, suggesting greater Pd-dispersion, this result being confirmed by microreactor-MS studies. It can therefore be concluded that  $PdO<sub>x</sub>$  particles are also reduced to small Pd metal particles by  $H_2$ , a portion of these being converted to ion-exchanged Pd upon re-oxidation. This is also an indication that Pd reduction by water and H2 have similar effects on Pd, as the water-pretreated material also exhibited this behavior.

Of crucial importance,  $H_2$  reduction and re-oxidation leads to the formation of greater amounts of ion-exchanged Pd on CHA and incomplete recovery of ion-exchanged Pd on BEA. This suggests that not all ionic Pd species participate in  $NO<sub>x</sub>$  adsorption on 1.4% Pd-BEA since its  $NO<sub>x</sub>$  TPD behavior upon re-oxidation shows an enhancement of

 $NO<sub>x</sub>$  storage over the as-prepared material, while ionic  $Pd(CO)$  DRIFTS band intensity is lost. The species primarily recovered on re-oxidation are  $Pd^+$  and  $[Pd^{2+}(OH)]^+$ , an indication that these species are primarily responsible for  $NO<sub>x</sub>$  adsorption at ionexchanged Pd on BEA. The behavior of 0.7% Pd-CHA again differs substantially, preferential formation of a high desorption temperature  $NO<sub>x</sub>$  storage site being observed. This leads to an increase in the lowest  $NO<sub>x</sub>$  desorption temperature into a more favorable range while also increasing the overall  $NO<sub>x</sub>$  storage capacity through improved Pd dispersion. Overall, 0.7% Pd-CHA shows greater resistance to Pd reduction than 1.4% Pd-BEA, a result that indicates the favorability of this material as a PNA candidate.

The focus of the next chapters will be on evaluating the effects of  $NO$  and  $NO<sub>2</sub>$  on 1.4 % Pd-BEA and 0.7% Pd-CHA, both as pretreatment gases and as adsorbates. Additionally, the role of zeolite sites in  $NO<sub>x</sub>$  adsorption will be further explored, microreactor-MS studies again being used to corroborate DRIFTS results.

# CHAPTER 4. ROLE OF PARTIALLY HYDROLYZED ZEOLITIC Al IN PASSIVE NOX ADSORPTION

To gain further understanding of the sites available for Pd ion-exchange and their impact on  $NO<sub>x</sub>$  adsorption, a more thorough examination of the zeolite OH band structure was carried out in addition to the examination of additional adsorbates, including  $CO<sub>2</sub>$ , NO2, NO, and NH3.These adsorbates allow for analysis of the formation of carbonates, characterization of the  $NO<sub>x</sub>$  adsorption sites, and characterization of the zeolite acid sites respectively. Emphasis will be placed on the comparison of the H-form materials and Pdloaded materials to provide insight into the specific zeolite sites that Pd and other adsorbates occupy, while also exploring the behavior of P-HAl sites. The role of P-HAl in  $NO<sub>x</sub>$  adsorption and desorption will be explored, the putative adsorbates and desorption events that correlate with the presence of P-HAl being identified. This work focuses on the BEA materials due to the near-absence of P-HAl species on CHA. DRIFTS data will be compared to the fundamental  $NO<sub>x</sub>$  adsorption and desorption behaviors observed for both BEA and CHA, these data being collected by Dr. Yaying Ji in the initial phases of this work. Data provided by Dr. Ji are included in Figures 4.12- 4.15, this work focusing primarily on the effects of  $NO<sub>x</sub>$  adsorption temperature, as well as the rate of  $NO<sub>x</sub>$  adsorption and quantity of  $NO<sub>x</sub>$  adsorbed for the various materials.

## 4.1 Zeolite OH Bands and the Effect of Pd-loading

To identify the various zeolite OH species present on each of the materials prior to Pd exchange, DRIFT spectra of the OH regions of the H-form materials dehydrated at 500 °C for 1 h in flowing Ar were obtained (see Figure 4.1). A band appearing on H-BEA at  $3785 \text{ cm}^{-1}$  has been previously assigned to a P-HAl site of the type presented in Figure 4.2.<sup>127</sup> This species has been proposed to serve as an exchange site for the basic  $Cs<sup>+</sup>$  ion through the Lewis acidity of the under coordinated Al atom, rather than exchanging through a Brønsted acid mechanism.<sup>88, 120, 128</sup> This band is nearly absent from H-CHA, a result that again indicates CHA's greater resistance to dealumination, resulting from its smaller pore size. Other OH-stretching modes are observed as well, terminal and internal Si(OH) bands occurring at 3731 and 3745  $cm^{-1}$  on both materials, EFAl(OH)

67

being evident at 3662 cm<sup>-1</sup> and framework Al-O(H)-Si at 3606 cm<sup>-1</sup>.<sup>88, 120</sup> Lastly, a band at  $3577 \text{ cm}^{-1}$  on H-CHA can be assigned to H-bonded OH groups, while this band is absent on H-BEA.<sup>88, 127</sup> Meanwhile, the P-HAl OH band is one of the most intense features, suggesting a substantial abundance of these sites on this BEA zeolite. The weakness of the Brønsted acid OH band of this material is also suggestive of the conversion of Brønsted acid sites to P-HAl Lewis acid sites as described by Bortnovsky et al., this process being greatly accelerated by steaming.127 For reference, a complete listing of FT-IR bands discussed in this section is presented in Table 4.1.

Band $(cm^{-1})$	Assignment	Position	References
3577	H-Bonded OH	Framework OH	120, 128
3606	$Al-O(H)-Si$	Framework OH	120, 128
3662	EFA1	Non-framework OH	120, 127
3731	Si(OH)	Framework Defect OH	120
3745	Si(OH)	Framework Defect OH	120
3785	P-HA1	Partially Framework OH	88, 120
2125	CO	Pore Condensed	120
2171	CO	Brønsted acid sites	120
2194	CO	Lewis acid sites	120
2206	CO	P-HA1	120
890	P-HAl-O	Partially Framework Al-O	88, 120
940	$Si-O$	Framework	120

Table 4.1 FT-IR band assignments for H-form BEA and CHA



Figure 4.1 H-BEA, H-CHA: DRIFTS spectra of the OH region of the dehydrated materials after pretreatment at 500 °C for 1 h with Ar rigorously dried by cold trap. Materials were cooled in dried Ar to 25 °C prior to collection.



P-HAl(OH) (3785 cm<sup>-1</sup>)

Figure 4.2 Mechanism of P-HAl formation proposed by Bortnovsky et al. along with the relevant OH band frequencies.<sup>127</sup>

The CHA materials show little change in the OH region upon Pd-loading (not shown), no readily discernable trends arising. The effect of Pd-loading on the OH region of the BEA zeolite IR spectrum was also examined; results being shown in Figure 4.3. The low intensity and overlapping development of Pd(OH) bands prevents effects on Brønsted sites from being observed, though the intensity of Si(OH) bands decreases with Pd-loading. This is likely due to Pd exchanging at adjacent Al-O(H)-Si sites, thus shifting the Si(OH) stretch of nearby groups out of the usual frequency range. This behavior cannot be explained by Pd exchange at Si(OH) sites as these are not appreciably Brønsted acidic.<sup>120</sup> The spectrum of 1% Pd-Si-BEA is also included here, this material showing only weak Si(OH) bands.



Figure 4.3 DRIFT spectra of the OH region of the dehydrated BEA materials after pretreatment at 500  $^{\circ}$ C for 1 h with Ar rigorously dried by cold trap. Materials were cooled in dried Ar to 25 °C prior to collection.

The P-HAl(OH) band is also reduced in intensity with Pd loading; however, the nature of this interaction is not well understood. Bisio et al. observed a similar phenomenon when  $Cs<sup>+</sup>$  was impregnated onto BEA zeolites, though it is unlikely that Pd will coordinate similarly given the Lewis acidity of Pd cations, exchange in the former case being proposed to occur due to the basicity of the  $Cs<sup>+</sup>$  ion. Cation exchange is only one of the proposed explanations for this effect, the other being that this site is leached by acid during the impregnation and calcination process, though Bisio et al. also suggest this effect should be inhibited when incipient wetness impregnation is employed (as it was for the materials discussed here), this method also being shown to achieve full quenching of Brønsted acidity upon Cs exchange.<sup>88</sup> While the Lewis acidity of P-HAl is well documented, this site has also been demonstrated to act as a Brønsted acid, though its acid strength is proposed to be intermediate to the essentially non-acidic silanols and the strongly acidic Al-O(H)-Si species.<sup>120</sup> As a result, Pd-exchange at this site could occur

through a Brønsted interaction wherein  $Pd^+$  or  $Pd^{2+}$  exchanges at P-HAl(OH) alone or at P-HAl(OH) and an adjacent Al(OH) site, respectively.

An alternative explanation for the decreased P-HAl(OH) band intensity is that Pd exchanges at a sufficient number of sites adjacent to P-HAl to disrupt the electronic environment and shift this OH stretch outside its normal frequency range as proposed for the silanol OH bands. However, close examination of the changes in this band from material to material suggest this is not the case, the initial change from H-BEA to 0.1% Pd-BEA being the largest. As Pd-loading is increased after 0.1%, the P-HAl(OH) band only decreases slightly in intensity despite the Pd loading eventually being nearly 14 times greater. This observation argues against a long-range effect as observed for the Si(OH) bands, as these bands change substantially and continuously as Pd loading increases. Due to the persistence of the band at  $3785 \text{ cm}^{-1}$ , apparently only a limited population of P-HAl sites is available for exchange, possibly a consequence of the need for an adjacent Al-O(H)-Si site to achieve Pd exchange; if true, divalent Pd cations would participate in this, while monovalent Pd would not; therefore isolated P-HAl sites would presumably be unable to stabilize Pd cations.

To further probe the zeolite acid sites, CO was adsorbed onto both H-form zeolites, formation of weak CO bands and perturbation of zeolite OH bands being observed as shown in Figure 4.4. CO adsorption occurs at both framework Al-O(H)-Si and at EFAl as evidenced by disturbance of the bands at 3606 and 3662  $cm^{-1}$ , respectively, though this effect is much weaker for BEA. CO does not perturb the P-HAl(OH) band appreciably at room temperature, though a weak CO band is observed at approximately  $2206 \text{ cm}^{-1}$  that has previously been assigned to CO adsorbed at these sites.<sup>120</sup>



Figure 4.4 DRIFT spectra of the OH (top) and CO (bottom) regions of the H-BEA and H-CHA materials after pretreatment at 500  $\degree$ C for 1 h with Ar rigorously dried by cold trap. Materials were cooled in dried Ar to 25 °C followed by adsorption of 1000 ppm CO in Ar for 10 minutes.

### 4.2 NH3 and CO2 Adsorption and Treatment on H-BEA

While CO provides examples of the zeolite's behavior in the presence of a weakly basic molecule, the impact of NH<sub>3</sub> is more significant. Pretreatment of H-BEA in NH<sub>3</sub> at 500 °C for 1 h revealed an increase in the intensity of the P-HAl(OH) band and an accompanying increase in the intensity of Si(OH) bands as shown in Figure 4.5. This

suggests that NH<sub>3</sub> is capable of converting Lewis acidic Al present in the framework to P-HAl as described by Bortnovsky et al.<sup>127, 128</sup> To better understand the nature of the interaction of P-HAl with Pd, the behavior of the zeolite framework Al-O and Si-O bands was also examined, these bands representing the IR active bonds present in the zeolite framework. Upon NH<sub>3</sub> treatment, a slight increase in intensity at 890 cm<sup>-1</sup> is observed, (see Figure 4.5 bottom) this band being previously correlated with the formation of P-HAl sites.<sup>88</sup> A band appearing at  $940 \text{ cm}^{-1}$  has been observed to increase in intensity upon dealumination (greater abundance of Si-O), while in this case it decreases in intensity, suggesting loss of Si-O(H)-Al in favor of P-HAl.<sup>120</sup> Furthermore, the band at 940 cm<sup>-1</sup> diverges into two distinct features upon Pd loading (902 and 952 cm<sup>-1</sup>), though no corresponding decrease in the band at  $890 \text{ cm}^{-1}$  is observed. The behavior of the band at  $890 \text{ cm}^{-1}$  is therefore consistent with the continued presence of a similar amount of P-HAl before and after exchange with Pd, indicating that the loss of intensity in the OH region is the result of a direct interaction with Pd, rather than the leaching of these sites by acid during Pd impregnation.

Adsorption of NH3 on H-BEA is also revealing, both zeolite OH and framework stretches being quenched by NH<sub>3</sub> adsorption as evidenced by their decrease in intensity during adsorption as shown in Figure 4.6. The band at 940  $cm^{-1}$  remains after adsorption, an expected result that shows many of the framework Si species are non-acidic. Both the framework and OH bands associated with P-HAl are eliminated during adsorption, suggesting that basic NH3 interacts strongly with this site. This supposition is further confirmed by the fact that recovery of this band does not fully occur until 440 °C upon  $NH_3$ -TPD (not shown). Formation of  $NH_4^+$  at Brønsted acid sites was observed, a band corresponding to these species appearing at  $1490 \text{ cm}^{-1}$  (Figure S2). <sup>88, 129</sup>



Figure 4.5 DRIFT spectra of the OH (top) and framework (bottom) regions of the BEA materials after pretreatment at 500 °C for 1 h with Ar rigorously dried by cold trap or  $1\%$  NH<sub>3</sub> in Ar. The materials were cooled to 25  $\degree$ C prior to the collection of spectra.



Figure 4.6 DRIFT spectra of the OH (top) and framework (bottom) regions of H-BEA after pretreatment at 500 °C for 1 h with Ar rigorously dried by cold trap. The materials were cooled to 25 °C prior to adsorption of 1% NH<sub>3</sub> in Ar for 10 min, spectra being collected every 1 min.

Interaction of H-BEA with a Lewis acidic molecule, namely  $CO<sub>2</sub>$ , was also examined. Shown in Figure 4.7, treatment of H-BEA in  $1\%$  CO<sub>2</sub> in Ar at 500 °C for 1 h leads to a decrease in the intensity of the P-HAl(OH) band, suggesting a decrease in the abundance of these sites over the H-form material, an indication that this site is a strong enough Lewis acid that more weakly Lewis acidic species such as  $CO<sub>2</sub>$  can react as Lewis bases with this site. The results from the framework region are less easily interpreted; however, a shift in the relative intensities of Si(OH) bands is observed with a corresponding shift in the maximum of the framework Si-O band at 940 cm<sup>-1</sup> to 935 cm<sup>-</sup> <sup>1</sup>, this leading to a convoluting influence in the region of the 890 cm<sup>-1</sup> band that produces an apparent increase in intensity of this species.  $CO<sub>2</sub>$  adsorption does show a limited interaction with the P-HAl sites as evidenced in Figure 4.8, EFAL and Brønsted acid OH bands increasing in intensity during adsorption (suggesting elimination of trace water), while the P-HAl(OH) band acts uniquely in becoming less intense with  $CO<sub>2</sub>$  adsorption, again indicating that  $CO<sub>2</sub>$  interacts with this site as it quenches the OH band.



Figure 4.7 DRIFT spectra of the OH (top) and framework (bottom) regions of H-BEA after pretreatment at 500 °C for 1 h with Ar rigorously dried by cold trap or 1% CO<sub>2</sub> in Ar. The materials were cooled to 25 °C prior to the collection of spectra.



Figure 4.8 DRIFT spectra of the OH region of H-BEA after pretreatment at 500 °C for 1 h with Ar rigorously dried by cold trap. The material was cooled to 25 °C prior to adsorption of  $1\%$  CO<sub>2</sub> in Ar for 10 min, spectra being collected every 1 min.

Figure S3 shows the formation of a weak bicarbonate feature at  $1665 \text{ cm}^{-1}$  during  $CO<sub>2</sub>$  adsorption, additional evidence that the interaction of these species is the result of a Lewis acid-base reaction with  $CO<sub>2</sub>$  acting as the base.<sup>130</sup> Therefore, it can be suggested in this case that P-HAl sites are removed by acidic treatment both in solution and in the gas phase at 500 °C, however, as Bisio et al. suggest, this effect is limited when incipient wetness impregnation is employed to deposit Pd. Thus, the effect of acidic species during Pd-loading cannot adequately explain the loss of intensity of this feature, the minimum intensity of the P-HAl(OH) band still being observed on 1.4% Pd-BEA. Given the broad reactivity of P-HAl, its role in  $NO<sub>x</sub>$  adsorption must be more thoroughly explored, especially considering its abundance on BEA.

To summarize the results presented in this section, a more complete examination of the behavior of the zeolite OH-bands with Pd loading was performed, most prominently showing a possible interaction of Pd with P-HAl. Several explanations have been proposed for this behavior, including acid leaching during Pd impregnation, in addition to being a possible result of global changes in the zeolite acidity.<sup>88, 131</sup> To gain

better understanding of the behavior of P-HAl in the presence of acids and bases,  $CO<sub>2</sub>$ and NH3 were employed, respectively. The P-HAl site itself has been shown to be both Brønsted and Lewis acidic, while NH3 reacts with zeolitic Lewis acid Al sites to form additional P-HAl at high temperatures. This result is supported by the behavior of the framework Si-O and P-HAl-O stretches, the latter showing no loss of intensity upon Pd impregnation, thus suggesting the P-HAl(OH) is indeed interacting with Pd directly rather than being lost to acid leaching.  $NH<sub>3</sub>$  adsorption further supports these results, a strong interaction with P-HAl at 25 °C being observed, with concurrent quenching of P-HAl-O and P-HAl(OH) bands. The formation of bicarbonates and  $NH_4^+$  species at these sites is possibly a result of  $CO<sub>2</sub>$  and NH<sub>3</sub> acting as Lewis bases at the strongly Lewis acidic P-HAl site.

# 4.3 NO/NO2-DRIFTS

The adsorption behavior of NO and  $NO<sub>2</sub>$  on zeolites is more consistently described in the literature than that of CO, though some debate remains about the identity of the  $Pd^{n+}(\text{NO})$  bands.<sup>59, 90</sup> Assignment of observed species is therefore an easier task than for the CO spectra, though the precise zeolite sites at which the adsorbates reside is still somewhat ambiguous. To further the understanding of the  $NO<sub>x</sub>$  adsorption behavior of these materials, adsorption of NO and NO2 onto both the H-form materials was conducted. To deconvolute the behaviors of NO and  $NO<sub>2</sub>$ , the NO feed gas was passed through a cold trap, serving to condense both water and  $NO<sub>2</sub>$ . Unfortunately, trace water and NO cannot be removed from the  $NO<sub>2</sub>$  feed gas, though association of bands with their respective adsorption sites is still possible.

Figure 4.9 shows adsorption of NO and NO<sub>2</sub> onto both H-form materials at 50 °C, OH and NO stretching regions being displayed. Several NO stretches are observed, specifically NO<sup>+</sup> bands in the range 2100 to 2200 cm<sup>-1</sup>, while a NO associated feature is observed at  $\sim$ 1870 cm<sup>-1</sup> on both materials and under adsorption of both gases. Only the H-BEA material under exposure to  $NO<sub>2</sub>$  reveals the presence of abundant nitrate-like species, these bands occurring in the range  $1580 - 1650$  cm<sup>-1</sup>, though an extremely weak feature does appear on H-CHA at 1625 cm<sup>-1</sup> corresponding to molecularly adsorbed NO<sub>2</sub>.

The OH region can aid in identifying the zeolite sites that these species occupy through the perturbation of OH species. NO adsorption is shown to result in perturbation of EFAl(OH) bands on both materials, in addition to the Brønsted acid OH bands, indicating that molecularly adsorbed NO and  $NO<sup>+</sup>$  can be attributed to adsorption of these species at EFAI and Brønsted acid sites, respectively, though some  $NO<sup>+</sup>$  may also reside at EFAI sites.<sup>59</sup> Finally, adsorption of NO<sub>2</sub> on the H-BEA material results in a sharp decrease in the P-HAl(OH) band, indicating this is the primary adsorption site for  $NO<sub>2</sub>$  and is also likely responsible for the formation of nitrates. Time-resolved correlation of nitrate formation and P-HAl(OH) disappearance is observed, this being discussed in section 5.4. Interaction of EFAI with  $NO<sub>2</sub>$  is also indicated, though trace NO present in the  $NO<sub>2</sub>$  feed makes deconvolution of the effects of these two species impossible in this instance. Conversely, NO adsorption leads to no such decrease in intensity of the P-HAl(OH) band, suggesting that NO does not interact with this site to any great extent. Further evidence that P-HAl is the primary adsorption site of  $NO<sub>2</sub>$  is the near-absence of both P-HAl(OH) bands and nitrate bands on the H-CHA material, suggesting that the interaction of  $NO<sub>x</sub>$  with EFAL can be explained by NO adsorption rather than  $NO<sub>2</sub>$  adsorption.



Figure 4.9 DRIFT spectra of the NO (top) and OH (bottom) regions of the H-BEA and H-CHA materials after pretreatment at 500 °C for 1 h with Ar rigorously dried by cold trap. Materials were cooled in dried Ar to 50 °C followed by adsorption of 1000 ppm NO or  $NO<sub>2</sub>$  in Ar for 10 minutes.

The role of P-HAl in  $NO<sub>x</sub>$  storage can be further explored by examining the effect of Pd loading on  $NO<sub>2</sub>$  adsorption, these results being presented in Figure 4.10. NO<sup>+</sup> formation is greatly enhanced on the Pd-BEA materials, as evidenced by an increase in the intensity of these bands relative to the nitrate features, this previously being tied to enhancement of low-temperature  $NO<sub>x</sub>$  desorption in TPD studies.<sup>59, 91</sup> Pd-nitrosyl bands

also appear, nearly equal abundances of  $Pd^+$  and  $Pd^{2+}$  being observed for the BEA material (presuming these species have similar extinction coefficients) though there is a slight preference for  $Pd^+$  at low loadings.<sup>74, 91</sup> Pd-CHA shows a more intense and broader  $Pd^{2+}(NO)$  band than observed on Pd-BEA, an indication that a greater variety of  $Pd^{2+}$ species participate in NO adsorption on CHA. Examination of the OH region reveals a similar trend in the intensity of the P-HAl(OH) band to that observed in the spectra of the dehydrated materials, the H-form again showing the greatest abundance of P-HAl. A complete listing of bands resulting from NO and  $NO<sub>2</sub>$  adsorption are shown in Table 4.2 for reference.

Band $(cm^{-1})$ (BEA/CHA)	Assignment	Position	References	
1589	Bidentate NO <sub>3</sub>	P-HA1	89, 92	
1625	Bridging $NO3$	P-HA1	89, 92	
1654	Z(NO <sub>2</sub> )	P-HA1	89, 92	
1832/1804	$Pd^+(NO)$	Isolated Al	74, 91	
1872/1853	$Pd^{2+}(NO)$	Paired Al	74, 91	
1870	Z(NO)	<b>Brønsted</b> Acid/EFA1	89, 92	
2140/2165	$Z(NO)^+$	<b>Brønsted</b>	89, 92	
2171/2190		Acid/EFA1		

Table 4.2 NO and NO2 FT-IR band assignments for BEA and CHA

To provide further support to these results,  $NH<sub>3</sub>$  and  $NO<sub>2</sub>$  were adsorbed sequentially at 50  $\degree$ C on H-BEA, the interaction between NH<sub>3</sub> and P-HAl presumably being stronger than  $NO<sub>2</sub>$  due to the more favorable acid-base properties of that interaction. These results are presented in Figure 4.11 and show that after  $NH<sub>3</sub>$  adsorption the formation of nitrates on H-BEA is prevented. Formation of  $NO<sup>+</sup>$  is apparently also inhibited, while the molecularly adsorbed NO feature at  $\sim$ 1870 cm<sup>-1</sup> does appear. Concurrently, little perturbation of the P-HAl(OH) band is observed, while interaction of

NO2 with EFAl and Brønsted acid sites is observed to a greater degree after NH3 adsorption than when  $NO<sub>2</sub>$  is adsorbed alone.



Figure 4.10 DRIFT spectra of the NO stretching regions of the 0.7 % Pd-CHA, 0.1, 0.6, and 1.4% Pd-BEA materials after pretreatment at 500 °C for 1 h with Ar rigorously dried by cold trap. Materials were cooled in dried Ar to 50 °C followed by adsorption of 1000 ppm  $NO<sub>2</sub>$  in Ar for 10 minutes.



Figure 4.11 DRIFT spectra of the NO stretching regions of the H-BEA material after pretreatment at 500  $\degree$ C for 1 h with Ar rigorously dried by cold trap. Materials were cooled in dried Ar to 50 °C followed by adsorption of 1000 ppm NO2 in Ar for 10 minutes, or sequential adsorption of NH3 for 10 min followed by NO2 adsorption for 10 min.

To summarize the conclusions of this section, NO and NO2 adsorption serves to corroborate the results obtained from CO adsorption regarding Pd speciation on BEA at low Pd-loadings, while the interaction of  $NO<sub>x</sub>$  with P-HAl was examined. P-HAl is found to be associated with the formation of nitrate-like species on both the H- and Pd-form BEA materials, while these species are nearly absent from the CHA materials.  $NO<sub>2</sub>$  is

also found to be necessary for the formation of nitrates, while NO is found to form  $NO<sup>+</sup>$ ,  $Pd^{n+}(\text{NO})$ , and a molecularly adsorbed NO species adsorbed at Brønsted acid, ionexchanged Pd, and EFAl respectively, the behavior of these bands being corroborated by the behavior of the respective OH features. However, the behavior of  $NO<sup>+</sup>$  and the molecularly adsorbed NO cannot be fully deconvoluted and as such these species may reside at EFAI or Brønsted acid sites. Furthermore, NO<sub>2</sub> is found to interact strongly with the P-HAl(OH) band while NO does not behave similarly. Finally, sequential adsorption of NH<sub>3</sub> and NO<sub>2</sub> inhibits the formation of nitrates by blocking P-HA1 sites prior to NO<sub>2</sub> adsorption. In the next section, these results will be compared with DRIFTS and microreactor-MS  $NO<sub>x</sub>$  TPD studies to provide insight into which of the observed  $NO<sub>x</sub>$ species desorb within the range most suitable for PNA behavior.

# 4.4 Microreactor NOx TPD Studies

The basic  $NO<sub>x</sub>$  adsorption properties of the materials were established by Dr. Yaying Ji, the effect of atmosphere on  $NO<sub>x</sub>$  adsorption and desorption being shown in Figure 4.12 for 1.4% Pd-BEA, H-BEA, 0.7% Pd-CHA and finally H-CHA. The presence of  $O_2$ during adsorption and desorption is found to enhance  $NO<sub>x</sub>$  adsorption on both the CHA and BEA materials, while  $H_2O$  serves to suppress it. The BEA materials present a complex  $NO<sub>x</sub>$  desorption profile regardless of adsorption environment. 1.4 % Pd-BEA has three readily discernable desorption events, a low-temperature feature with a maximum at  $\sim$ 135 °C, an intermediate feature with a maximum at  $\sim$ 210 °C and finally a high-temperature feature that has an inconsistent maximum temperature in the range 300–400 °C. H-BEA only possesses a low-temperature feature and an intermediate temperature feature at  $100-150$  °C and 300 °C respectively.

Overall, the CHA materials are demonstrated to adsorb more  $NO<sub>x</sub>$  except when water is added to the adsorption gas, BEA showing superior performance under those conditions. H-CHA exhibits only a low-temperature desorption event at 132  $\degree$ C, while 0.7% Pd-CHA shows the appearance of a desorption feature centered around 220  $^{\circ}$ C in addition to the low-temperature feature when  $O_2$  is present. However, no increase in desorption temperature is evident in the absence of  $O_2$ , only an increase in the intensity of

85

the low-temperature feature over the H-form material. A high-temperature feature centered at 400 °C also appears on 0.7% Pd-CHA, this species being analogous to that observed on 1.4% Pd-BEA with a maximum above 350 °C. The total quantities of desorbed  $NO<sub>x</sub>$  from each of these experiments is shown in Figure 4.13, along with the NO/Pd ratio where applicable.





Figure 4.12 1.4% Pd-BEA (top),  $0.7\%$  Pd-CHA (bottom). NO<sub>x</sub> TPD conducted in microreactor-MS with calibrated NOx concentrations. Materials were pretreated in 10% O2 in He for 1 h prior to adsorption. 1000 ppm NO in He, 10% O2/He, or 1- 2% water 10% O<sub>2</sub>/He was adsorbed for 10 min at 50 °C followed by a 1 h purge in 10%O<sub>2</sub>/He. TPD was carried out in 10% O<sub>2</sub>/He at a ramp rate of 10 °C per min.



Figure 4.13  $NO<sub>x</sub>$  TPD conducted in microreactor-MS with calibrated NOx concentrations. Materials were pretreated in  $10\%$  O<sub>2</sub> in He for 1 h prior to adsorption. 1000 ppm NO in He, 10% O<sub>2</sub>/He, or 1-2% water 10% O<sub>2</sub>/He was adsorbed for 10 min at 50 °C followed by a 1 h purge in  $10\%O_2$ /He. TPD was carried out in 10% O<sub>2</sub>/He at a ramp rate of 10 °C per min. Tabulated values were calculated from integrated  $NO<sub>x</sub> MS$  signals.

According to the NO/Pd ratio, Pd utilization on CHA is observed to be lower than on BEA in agreement with the  $H_2$ -TPR results that indicate lower Pd dispersion on 0.7% Pd-CHA. Pd-CHA also shows a smaller improvement in  $NO<sub>x</sub>$  storage capacity over the H-CHA material than the increase observed in comparing the H- and 1.4% Pd-BEA. This is a result of much of the Pd being present as  $PdO_x$  on CHA, though as shown above,  $NO_x$ storage can be improved after  $H_2$  reduction and re-oxidation. These data also serve to provide a proof of concept for these materials, especially for 1.4 % Pd-BEA, this material trapping a significant amount of  $NO<sub>x</sub>$  even in the presence of water and desorbing a large portion of this  $NO<sub>x</sub>$  near or above 200 °C.

Correlation of  $NO<sub>x</sub>$  desorption events with DRIFTS data will be explored further in the next chapter; meanwhile, additional adsorption parameters were evaluated in the presence of  $O_2$ , given the greater  $NO_x$  adsorption observed in this atmosphere. The effect of adsorption temperature on  $NO_x$  storage capacity of H-BEA and 1.4% Pd-BEA was evaluated, results being presented in Figure 4.14.



Figure 4.14 NOx TPD conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$ concentrations Right: total  $NO<sub>x</sub>$  adsorbed on H and Pd-BEA, left: deconvolution of NO<sub>x</sub> storage events on H-BEA. Materials were pretreated in  $10\%$  O<sub>2</sub> in He for 1 h prior to adsorption. 1000 ppm NO in  $10\%$  O<sub>2</sub>/He was adsorbed for 10 min at 30, 50, or 100 °C followed by a 1 h purge in  $10\%$ O<sub>2</sub>/He. TPD was carried out in 10% O<sub>2</sub>/He at a ramp rate of 10 °C per min. Tabulated values were calculated from integrated  $NO<sub>x</sub> MS$  signals.

NO<sub>x</sub> adsorption was shown to be at a maximum near 50 °C, Pd utilization and NO<sub>x</sub> storage both being greater than at the other two temperatures tested. Examining the easily separable desorption events of H-BEA shows that the high-temperature desorption event is maximized at 50 °C, while the desorption below 200 °C decreases in intensity as temperature increases. This is highly desirable: when adsorbed at 50  $\rm{^{\circ}C},$  most NO<sub>x</sub> desorbs above 200 °C, the minimum temperature for the downstream catalyst to become active. The effect of Pd-loading on the  $NO<sub>x</sub>$  storage capacity of BEA was also examined, the results being shown in Figure 4.15. Generally, as Pd-loading increases, the total quantity of  $NO_x$  adsorbed also increases, while 1% Pd-Si-BEA shows negligible  $NO_x$ adsorption under these conditions, indicating that  $PdO_x$  fails to store  $NO_x$ . Notably, the rate of  $NO<sub>x</sub>$  adsorption was found to be relatively constant for the first two minutes of adsorption regardless of the material tested, a value of  $\sim$ 43  $\mu$ mol/(g\*min) being determined (not shown). Given the above data, a  $NO<sub>x</sub>$  concentration of 1000 ppm and an adsorption temperature of 50  $\degree$ C was chosen for all NO<sub>x</sub>-DRIFTS experiments unless otherwise indicated.



Figure 4.15  $NO<sub>x</sub>$  TPD conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$ concentrations. Materials were pretreated in  $10\%$  O<sub>2</sub> in He for 1 h prior to adsorption. 1000 ppm NO in 10% O<sub>2</sub>/He was adsorbed for 10 min at 50 °C followed by a 1 h purge in  $10\%O_2$ /He. TPD was carried out in  $10\%$  O<sub>2</sub>/He at a ramp rate of 10 °C per min. Tabulated values were calculated from integrated  $NO<sub>x</sub> MS$  signals.

# 4.5 Conclusions

This chapter focused on the role of P-HAI in both Pd-siting and in  $NO<sub>x</sub>$  adsorption, examination of these behaviors being evaluated by in-situ DRIFTS. Increased Pd-loading leads to a reduction in intensity of the P-HAl(OH) band, this behavior being proposed to be a result of interaction with Pd due to the absence of a corresponding decrease in the P-HAl-O vibration as observed upon acid leaching of this species. The behavior of the zeolite OH bands and their corresponding framework vibrations were explored under treatment in both acidic  $(CO_2)$  and basic (NH<sub>3</sub>) gases. Treatment in NH<sub>3</sub> led to an increase in the intensities of the P-HAl(OH) and P-HAl-O bands, suggesting formation of additional P-HAl, while treatment in  $CO<sub>2</sub>$  led to a decrease in the intensity of the P-HAl(OH) band, suggesting a reduced abundance of these sites after treatment. Both species are shown to quench the P-HAl(OH) stretch upon adsorption at room temperature suggesting adsorption at this site, most probably with  $CO<sub>2</sub>$  and NH<sub>3</sub> acting as Lewis bases, a supposition supported by the formation of bicarbonate and  $NH_4^+$ .

NO and NO2 adsorption were also performed on both zeolite materials, emphasis being placed on the differing behaviors of the H-form and Pd-loaded zeolites. Nitrate species are observed only when  $NO<sub>2</sub>$  is adsorbed on the BEA materials, quenching of the P-HAl(OH) band occurring simultaneously. The absence of P-HAl on the CHA materials is accompanied by an absence of nitrate features. Meanwhile, NO adsorption at EFAL is proposed to result in a stretch at  $\sim 1870 \text{ cm}^{-1}$  on both materials, a corresponding OH perturbation again being observed. To further confirm the association of nitrates and P-HAl, NH3-NO2 sequential adsorption was performed on H-BEA, the results showing an absence of nitrate bands upon  $NO<sub>2</sub>$  adsorption when P-HAl sites have been previously saturated by NH<sub>3</sub>. Finally, the basic  $NO<sub>x</sub>$  desorption behaviors are described, 1.4 % Pd-BEA showing the highest  $NO<sub>x</sub>$  adsorption capacity at 50 °C under realistic conditions where water is present. Conversely, the CHA materials show better  $NO<sub>x</sub>$  storage capacity than BEA when water is absent, the presence of  $O_2$  improving capacity in either case. These data serve as a proof of concept for these materials, as 1.4% Pd-BEA exhibits significant  $NO<sub>x</sub>$  storage capacity at near-ambient temperatures in the presence of water, while also desorbing most of the adsorbed  $NO<sub>x</sub>$  within the desired temperature range.

# CHAPTER 5. ROLE OF Pd SPECIATION IN  $NO<sub>X</sub>$  STORAGE

The final chapter of this work focuses on the correlation of DRIFTS and microreactor-MS data along with a further exploration of the formation of  $NO<sub>2</sub>$ , and an examination of the reactivity of NO and  $NO<sub>2</sub>$  with Pd. This work will again primarily focus on the BEA materials due to their better performance during  $NO<sub>x</sub>$  adsorption in the presence of water, and due to the weaker interaction of  $NO<sub>2</sub>$  with CHA, observed by the near-absence of nitrate formation on this material. Special consideration will again be paid to the deconvolution of the effects of NO and  $NO<sub>2</sub>$ , this being a key area of investigation identified by Gu et al. in a recent review of PNA chemistry. <sup>59</sup> The correlation of microreactor  $NO<sub>x</sub>$  desorption events with adsorbate species identified by in-situ DRIFTS and their respective adsorption sites will provide data invaluable to the rational design of PNA materials, as the most desirable adsorption sites and adsorbate species can be identified and selected for the development of future materials. Some microreactor experiments performed for this chapter were conducted by Dr. Yaying Ji as part of her work to define the  $NO<sub>x</sub>$  adsorption and desorption behaviors of these materials in the early stages of this project, these data being included in Figures 5.1, 5.2 and 5.5.

# 5.1 Effect of Atmosphere and Pd-loading on  $NO<sub>x</sub>$  Desorption

# 5.1.1 Microreactor-MS  $NO<sub>x</sub>$ -TPD

 $NO<sub>x</sub>-TPD$  of the Pd-BEA materials was conducted in both inert conditions and in the presence of  $O_2$ , these data being shown in Figure 5.1. The overall quantities of  $NO<sub>x</sub>$ stored on each of the materials is similar whether  $O_2$  is present or not, except in the case of H-BEA, which reveals a substantial increase in  $NO<sub>x</sub>$  storage capacity in the presence of O2. However, the desorption profiles of the materials are greatly dissimilar in the differing atmospheres. The effect of  $O_2$  was also explored by CO-DRIFTS, results showing that when oxygen is present during pretreatment the majority of the Pd present can be inferred to be in the divalent state (Section 2.2.2), while some  $Pd^{2+}$  species present on BEA are shown to not participate in  $NO<sub>x</sub>$  adsorption (Section 3.2.3), these results being especially relevant to the following discussion. During  $NO<sub>x</sub>$  adsorption and

92
desorption in the presence of  $O_2$ , the materials present four differentiable  $NO<sub>x</sub>$  desorption events at 100, 150, 300, and finally 400–500 °C. Given the gradual disappearance of the desorption feature at 300 °C as Pd-loading increases, and the presence of this feature on H-BEA, it can be inferred that this desorption event is associated with the decomposition of nitrates adsorbed at P-HAl. This desorption event is fully extinguished at 1.4% Pdloading, further emphasizing the interaction of Pd with P-HAl, the presence of Pd apparently serving to reduce the desorption temperature of nitrate-like species into a more favorable range ( $\sim$ 300 to  $\sim$ 200 °C), this behavior being correlated with the decreasing intensity of the P-HAl(OH) FT-IR band as Pd-loading increases. The features at  $\sim$ 150 °C and 350–450 °C can be associated with the addition of Pd, as neither of these features appears on the H-form material regardless of atmosphere.



Figure 5.1 NOx TPD conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$ concentrations. Materials were pretreated in  $10\%$  O<sub>2</sub> in He or He for 1 h prior to adsorption. 1000 ppm NO in 10% O<sub>2</sub>/He or He was adsorbed for 10 min at 50 °C followed by a 1 h purge in 10%  $O_2$ /He or He. TPD was carried out in 10%  $O_2$ /He or He at a ramp rate of 10 °C per min.  $O_2$  was either included (top) or excluded (bottom) during all steps.

 $NO<sub>x</sub>-TPD$  performed after NO storage in the absence of  $O<sub>2</sub>$  is also informative. The desorption event at 300 °C increases in intensity from H-BEA to 0.1% Pd-BEA under these conditions, an indication that additional  $NO<sub>2</sub>$  (then stored as nitrates) is formed under inert conditions either as a result of the reduction of  $Pd^{2+}(\text{OH})$  to  $Pd^+$  by NO as described by Descorme et al., or by reaction of NO with  $PdO_x$  to form  $Pd^{0.90}$  This feature is extinguished at a loading of 0.6% under these conditions, suggesting nitrate desorption is influenced by Pd at lower loadings under inert conditions as this desorption event isn't fully extinguished until 1.4% Pd when  $O_2$  is present. This also explains the lack of effect on this desorption event at 0.1% Pd in the presence of  $O_2$ , formation of additional  $NO_2$ upon addition of Pd outweighing the shift of nitrates that interact with Pd to lower desorption temperatures. The low-and intermediate- temperature desorption features are more complex under inert conditions, a new desorption maximum appearing at  $\sim$ 150–170 °C. This feature, as well as the desorption events at 100 and 200 °C, all increase in intensity with Pd-loading. The highest temperature desorption event (450 °C) appears at a Pd-loading of 0.6% and is still more intense at a loading of 1.4%. In either atmosphere, this feature appears simultaneously with the full disappearance of the event at 300  $^{\circ}C$ , with the 450 °C species evidently being strongly favored under inert conditions. That conclusion favors assignment of this desorption event as  $Pd<sup>+</sup>(NO)$ ,  $Pd<sup>+</sup>$  also being proposed to be more abundant on the Pd-BEA materials under inert treatment as described by CO-DRIFTS.

To further clarify the above results, it is necessary to deconvolute the effects of  $O<sub>2</sub>$ during  $NO<sub>x</sub>$  adsorption and  $NO<sub>x</sub>$  desorption, especially regarding the role of  $NO<sub>2</sub>$  during each step of the process.  $NO<sub>x</sub>-TPD$  was therefore performed on 1.4% Pd-BEA with  $O<sub>2</sub>$ added at various stages of the experiment to assess the impact of  $O<sub>2</sub>$  during pretreatment, adsorption, and desorption separately. These results are presented in Figure 5.2, the experiments shown consisting of the full  $O<sub>2</sub>$  and full inert data shown above, as well as the material pretreated in  $O_2$  with adsorption and desorption carried out under inert conditions, and an additional experiment where only the adsorption was carried out under inert conditions.

Remarkably,  $NO<sub>x</sub>$  storage is lowest on the material for which NO was adsorbed in the absence of  $O_2$ , while the pretreatment and desorption were carried out in the presence of  $O_2$ , the combination of  $O_2$  treatment and inert adsorption being particularly unfavorable due to higher abundance of  $Pd^{2+}$  and limited formation of  $NO<sub>2</sub>$  under those circumstances. These suppositions are supported by the effect of inert pretreatment, this leading to an increase in the abundance of the high-temperature desorption feature (450

°C) and a decrease in the abundance of the intermediate- temperature desorption features (150-300 °C) relative to the  $O_2$ -treated material under identical adsorption and desorption conditions. Further, the overall  $NO<sub>x</sub>$  storage capacity is shown to be slightly greater on the BEA materials under inert conditions than under oxidizing conditions, the greater intensity of the high-temperature  $Pd<sup>+</sup>(NO)$  desorption event accounting for much of this additional NO<sub>x</sub> storage capacity. This is a further indication that not all the  $Pd^{2+}$  species observed in the CO-DRIFTS experiments participate in passive  $NO<sub>x</sub>$  adsorption, as a substantially greater change in the intermediate temperature  $NO<sub>x</sub>$  desorption events would be expected from the results of the CO-DRIFTS experiments under air and Ar pretreatment. The presence of  $O_2$  during desorption generally decreases the temperature of all desorption events.



Figure 5.2 NO<sub>x</sub> TPD of 1.4% Pd-BEA conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$  concentrations. Materials were pretreated in 10%  $O<sub>2</sub>$  in He or He for 1 h prior to adsorption. 1000 ppm NO in  $10\%$  O<sub>2</sub>/He or He was adsorbed for 10 min at 50 °C followed by a 1 h purge in  $10\%$  O<sub>2</sub>/He or He. TPD was carried out in 10% O<sub>2</sub>/He or He at a ramp rate of 10 °C per min.

From these data and the corresponding DRIFTS data, presented below, a rough correlation of desorption events and adsorption sites can be made. The lowesttemperature feature observed at 100  $^{\circ}$ C can be assigned to NO<sup>+</sup> and molecularly adsorbed NO at Brønsted acid sites and EFAI, while the event at 300 °C can be assigned to NO<sub>2</sub>

adsorbed at P-HAl sites, both of these desorption events being present on the H-form material and thus associated with zeolite sites. The desorption events at 150 and 250 °C can be assigned to NO adsorbed at  $Pd^{2+}$  and NO<sub>2</sub> adsorbed at Pd-influenced P-HAl sites respectively, while the highest-temperature feature can be assigned to NO adsorbed at  $Pd^+$ . Further, CO-DRIFTS data can be used to infer which  $Pd^{2+}$  species are involved, the most abundant  $Pd^{2+}(CO)$  bands being those related to  $[Pd^{2+}(OH)]^+$ , these species also observed to be easily reducible to Pd<sup>+</sup>.

One inconsistency in these data has yet to be resolved, this being the increase in  $NO<sub>x</sub>$  desorption in the range 150-300 °C during desorption under inert conditions as shown in Figure 5.2. The fact that desorption in this range under inert conditions is also enhanced by pretreatment in  $O_2$  suggests that  $Pd^{2+}$  is responsible for this increase, though that cannot fully explain the increase observed for  $NO<sub>x</sub>$  desorption in  $O<sub>2</sub>$ , similar amounts of  $Pd^{2+}$  theoretically existing in both cases. A possible explanation for this result would therefore be that the presence of  $O_2$  during adsorption helps to maintain Pd in the divalent state, or that an additional desorption feature associated with  $NO<sub>x</sub>$  adsorbed at Pd<sup>+</sup> occurs in the range of interest. The enhanced  $NO<sub>x</sub>$  adsorption under inert conditions would then result from reduction of  $Pd^{2+}$  to  $Pd^+$  during NO<sub>x</sub> adsorption, followed by adsorption of an additional NO as described by Descorme et al. $90$  This observation also points to the difficulty in interpreting these data, as most of these desorption events overlap to at least some degree and the true number of species present cannot be easily determined.

To summarize the results of this section, the effects of  $O_2$  on  $NO<sub>x</sub>$  desorption were analyzed, these results being correlated with CO-DRIFTS data to make inferences about the identities of the various  $NO<sub>x</sub>$  desorption events and also explore the chemistry that occurs during pretreatment, adsorption, and desorption. The presence of  $O_2$  during pretreatment is found to decrease  $NO<sub>x</sub>$  storage on this material, proposed to result from formation of  $Pd^{2+}$  species that do not store NO<sub>x</sub>. The presence of O<sub>2</sub> during adsorption is proposed to enhance  $NO<sub>2</sub>$  formation, leading to additional storage of  $NO<sub>x</sub>$  as nitrates while also inhibiting reduction of  $Pd^{2+}$ . Finally, the presence of  $O_2$  during the desorption step leads to a general decrease in the desorption temperature of all  $NO<sub>x</sub>$  species.  $NO<sub>x</sub>$ desorption event assignments can be resolved from these data,  $NO<sup>+</sup>$  and nitrate desorption

occurring at 100 and 300 °C respectively. Desorption events associated with  $NO<sub>x</sub>$ adsorbed at Pd<sup>n+</sup> sites are also proposed at 100-300 °C (Pd<sup>2+</sup>(NO)), 100-300 °C (Pd<sup>+</sup>(NO)), 200-300 °C (Pd-P-HAl) and finally 350-450 °C (Pd<sup>+</sup>(NO)). The observed changes in desorption under various conditions can largely be explained by changes in the NO<sub>2</sub> concentration resulting from the presence of O<sub>2</sub> or through reduction of Pd<sup>2+</sup> by NO.

## 5.1.2 In-situ DRIFTS NOx-TPD

To corroborate the putative  $NO<sub>x</sub>$  desorption assignments with spectroscopic data, NO2-TPD experiments were performed on the variously loaded materials, desorption behavior being monitored by in-situ DRIFTS. These experiments were carried out in the absence of  $O_2$  and with  $NO_2$  as the adsorbate to ensure the appearance of nitrate species. Immediately upon increasing the temperature, all species begin to decrease in intensity, suggesting that the desorption events overlap to a great degree as observed in microreactor experiments. However, useful inferences can still be derived from the final temperature at which a given band is observed, these values being tabulated in Table 5.1. Due to the convoluting effects of water pseudo-maxima, the behavior of  $NO<sup>+</sup>$  species is not evaluated here.

	$Pd^{2+}(NO)$ (°C) $1876$ cm <sup>-1</sup>	$Pd^+(NO)$ (°C) $1833$ cm <sup>-1</sup>	$Z(NO)_2$ $(^{\circ}C)$ $1650 \text{ cm}^{-1}$	Z(NO) <sub>3</sub> (°C) $1623$ cm <sup>-1</sup>	Z(NO) <sub>3</sub> (°C) $1589$ cm <sup>-1</sup>
<b>H-BEA</b>	N/A	N/A	440	160	420
$0.1\%$ Pd- <b>BEA</b>	N/A	140	420	160	340
$0.6\%$ Pd- <b>BEA</b>	180	220	380	160	300
1.4% Pd- <b>BEA</b>	220	260	380	160	320

Table 5.1  $NO<sub>x</sub>$  desorption temperatures from in-situ DRIFTS NO<sub>2</sub>-TPD





Figure 5.3 H-BEA (top) 0.1% Pd-BEA (second) 0.6% Pd-BEA (third) and 1.4% Pd-BEA (bottom): DRIFTS spectra of NO<sub>2</sub> desorption after pretreatment at 500 °C for 1 h with rigorously dried Ar. NO<sub>2</sub> was adsorbed for 10 minutes at 50 °C before the material was purged for 10 minutes in Ar prior to ramping to remove gas phase  $NO_2$ . Temperature was ramped at 10 °C/min to 500 °C with spectra being collected every 20 °C, highlighted spectra represent 50 °C (orange spectrum), and 100 °C (dark blue), 200 °C (yellow), 300 °C (light blue), 400 °C (navy), and 500 °C (black).

As expected, these final desorption temperatures differ significantly from the desorption maxima observed in microreactor experiments, though the general trends can still be established. Desorption of nitrates is shown to shift to lower temperatures as Pdloading is increased, further suggesting the identities of the microreactor desorption events at 200-300 °C can be associated with decomposition of nitrates, their decreasing desorption temperature explained by their interaction with Pd. The  $Pd<sup>n+</sup>(NO)$  bands also follow the same desorption trend as proposed in the microreactor experiments,  $Pd^{2+}$ desorbing NO at lower temperatures than  $Pd^+$ , though these species disappear at temperatures much lower than expected, this likely being a result of the convoluting influences of trace water, unavoidable when  $NO<sub>2</sub>$  is used as the adsorbate.

To better examine the desorption behavior of the  $Pd^{n+}(NO)$  bands, NO-TPD in Ar purified of water and  $NO<sub>2</sub>$  was performed, NO being adsorbed at 25 °C to achieve optimal band intensity. DRIFTS is especially sensitive to the presence of trace water, water adsorption occurring initially at the surface of the catalyst bed where the IR signal is reflected, leading to rapid interference with DRIFTS signals. As presented in Figure 5.4, these results demonstrate that the removal of trace water from the feed gas yields different desorption behavior, NO<sup>+</sup> and Pd<sup>+</sup>(NO) being still apparent even at 500 °C

during TPD. This result corroborates the assignment of the highest-temperature desorption feature to  $Pd<sup>+</sup>(NO)$ , as this species is observed to remain adsorbed at temperatures higher than nitrates. Of further note is the appearance of nitrate features beginning at 100 °C, these species forming from NO<sub>2</sub> produced either by the reduction of  $[{\rm Pd}^{2+}({\rm OH})]^+$  to Pd<sup>+</sup>, reduction of PdO<sub>x</sub> to Pd<sup>0</sup>, or due to spontaneous oxidation resulting from zeolite pore-confinement effects.<sup>132</sup>. These data also support the presence of two  $Pd^+$  sites with differing desorption temperatures, as the frequency of the  $Pd^+(NO)$  band shifts from  $1828-1808$  cm<sup>-1</sup> with increasing temperature. As with the CO bands, this frequency shift represents a change in the relative abundances of overlapping Pd populations, the more strongly adsorbed NO species eventually becoming dominant as the other desorbs.



Figure 5.4 1.4% Pd-BEA: DRIFTS spectra of NO desorption after pretreatment at 500 °C for 1 h with rigorously dried Ar. NO was adsorbed for 10 minutes at 25 °C before the material was purged for 10 minutes in Ar prior to ramping to remove gas phase NO. Temperature was ramped at 10 °C/min to 500 °C with spectra being collected every 20 °C, highlighted spectra represent 25 °C (orange spectrum), and 100 °C (dark blue), 200 °C (yellow), 300 °C (light blue), 400 °C (navy), and 500 °C (black).

This section can be concluded with the putative microreactor-MS and DRIFTS-TPD assignments for the BEA materials under inert conditions as shown in Table 5.5, this table serving to reconcile nearly all the data presented at this point. However, this information still does not provide sufficient differentiation to deconvolute the  $NO<sub>x</sub>$ desorption features to any degree of quantitative accuracy, as most desorption events overlap the desorption of one or more other species. The favorability of the various Pd species and adsorption sites for PNA applications can nevertheless be qualitatively

assessed based on their observed desorption behavior. These data suggest that the optimal sites for  $NO<sub>x</sub>$  storage on Pd-BEA are Pd<sup>+</sup> sites, as these species exhibit desorption maxima between 200 and 500 °C, at which point  $NO<sub>x</sub>$  will be converted on the downstream catalyst after desorption, while still achieving near complete  $NO<sub>x</sub>$  desorption below 500 °C. The desorption temperatures of these species are further shifted down into the favorable range by the presence of  $O_2$ , greater quantities of  $NO<sub>x</sub>$  being stored as  $NO<sub>2</sub>$ and nitrates at P-HAl sites in that case.  $O_2$  is also proposed to limit reduction of  $[Pd^{2+}(OH)]^{+}$  during treatment; in contrast, improved NO<sub>x</sub> storage is observed under inert pretreatment conditions on Pd-BEA, possibly due to the additional  $NO<sub>x</sub>$  adsorbed after  $[Pd^{2+}(OH)]^{+}$  is reduced to Pd<sup>+</sup>.

	In-situ DRIFTS						Microreactor- <b>MS</b>	
Adsorption Site	CO $\rm (cm^{-}$ $\mathbb{1}_{\lambda}$	NO $(cm^{-1})$	NO <sub>2</sub> $\text{cm}^{-}$ $1\lambda$	<b>OH</b> $(cm^{-1})$	CO End	NO <sub>x</sub> End	NO <sub>x</sub> Range	NO <sub>x</sub> Max
Zeolite <b>Brønsted</b> Acid	2165	$(NO^+)$ 2100- 2250	N/A	3610	25 °C	500 °C	$100 -$ 500°C	100 °C
<b>EFA1</b>	2185	1880	N/A	3665	25 °C	120 °C	50-120 $\rm ^{\circ}C$	$100\text{ °C}$
P-HAl	2208	N/A	1589 1625 1650	3785	25 °C	300 °C 160 °C 380 °C	$200 -$ 350 °C	$250 -$ 300 °C
$Pd^+$	2119 2132	1824 1808	N/A	N/A	300 °C	500 °C	$100 -$ 300 °C $350 -$ 550°C	200 °C 450 °C
$[{\rm Pd}^{2+}({\rm OH})]^+$	2125 2144	1876	N/A	3604 3656	200 °C			
$Pd^{2+}$	2154 2160 2169 2183	1876	N/A	N/A	200 °C	360 °C	$100 -$ 360 °C	200 °C
Super- electrophilic $Pd^{2+}$	2189 2212	1876	N/A	N/A	200 °C			

Table 5.2  $NO<sub>x</sub>$  adsorption sites with DRIFTS band frequencies and desorption endpoints correlated with microreactor desorption ranges and maxima

## 5.2 Comparison with CHA

Similar experiments to those described in the previous section were performed on the CHA materials, the effects of atmosphere and Pd loading again being examined. Figure 5.5 shows the effects of  $O_2$  on adsorption and desorption. The effects on the  $H_2$ reduced and re-oxidized 0.7% Pd-CHA are also considered, given the improvement in NOx storage capacity observed on that material. For the CHA materials, desorption events can be assigned as on BEA. The desorption behavior is easier to interpret here,  $O<sub>2</sub>$ appearing to improve  $NO<sub>x</sub>$  storage in each case. This result suggests that all or nearly all the  $Pd^{2+}$  species participate in  $NO_x$  adsorption on Pd-CHA. Meanwhile, no desorption features associated with nitrates are observed in this case, as anticipated. Desorption of  $NO<sub>x</sub>$  from Pd<sup>+</sup> at high temperatures is also observed, this band shifting its maximum desorption temperature lower on the re-oxidized material. In the case of  $NO<sub>x</sub>$  desorption when  $O_2$  is present, elevation of  $NO_x$  desorption temperature is observed on the reoxidized material, this likely being explained by the preferential formation of a particular Pd species.



Figure 5.5  $NO<sub>x</sub>$  TPD on CHA materials conducted in microreactor-MS with calibrated  $NO<sub>x</sub>$  concentrations. Materials were pretreated in 10%  $O<sub>2</sub>$  in He or He for 1 h prior to adsorption. 1000 ppm NO in  $10\%$  O<sub>2</sub>/He or He was adsorbed for 10 min at 50 °C followed by a 1 h purge in  $10\%$  O<sub>2</sub>/He or He. TPD was carried out in 10% O<sub>2</sub>/He or He at a ramp rate of 10 °C per min. O<sub>2</sub> was either included (top) or excluded (bottom) during all steps.

The DRIFTS  $NO<sub>x</sub>$ -TPD experiments were also replicated under dry conditions, these data being presented in Figure 5.6. Surprisingly, the  $Pd^{2+}(NO)$  band apparently persists at higher temperature than the  $Pd<sup>+</sup>(NO)$  band, this result appearing to be at odds

with the behavior observed in microreactor experiments. Careful examination reveals a potential explanation, as the  $Pd<sup>+</sup>(NO)$  band does not shift to lower frequencies on this material. This is an indication that the  $Pd<sup>+</sup>(NO)$  band observed here consists primarily of the proposed low-desorption-temperature  $Pd^+$  species, while the high-desorbing  $Pd^+(NO)$ band fails to appear due to its lower abundance on CHA. This supposition is supported by comparison of CHA microreactor data with that for the BEA materials, showing that the species responsible for the highest temperature desorption event is less abundant on CHA. Using these data and microreactor experiments, a correlation chart analogous to that for BEA is shown in Table 5.3. These results again point to the greater stability of  $Pd^{2+}$  species on CHA, nearly all of the identified species likely participating in PNA behavior.

Table 5.3  $NO<sub>x</sub>$  adsorption sites on CHA with DRIFTS band frequencies and desorption endpoints correlated with microreactor desorption ranges and maxima

	In-situ DRIFTS						Microreactor- <b>MS</b>	
Adsorption Site	CO $\rm (cm^{-}$ $\mathbf{1}$	NO $(cm^{-1})$	NO <sub>2</sub> $\rm (cm^{-}$ $\mathbf{1}$	OH $(cm^{-1})$	CO End	NO <sub>x</sub> End	NO <sub>x</sub> Range	NO <sub>x</sub> Max
Zeolite <b>Brønsted</b> Acid	2171	$(NO^+)$ 2100- 2250	N/A	3610	25 °C	500 °C	$50-$ 500 °C	100 °C
<b>EFA1</b>	2192	1872	N/A	3662	25 °C	100 °C	$50-$ 100 °C	100 °C
P-HA1	N/A	N/A	N/A	N/A	N/A	N/A	N/A	N/A
$Pd^+$	2113 2132	1803	N/A	N/A	280 °C	320 °C	$100 -$ 350 °C $350 -$ 550 °C	150 °C $350 -$ 450 °C
$[{\rm Pd}^{2+}({\rm OH})]^+$	2125 2142	1855	N/A	N/A	280 °C			
$Pd^{2+}$	2154 2175	1855	N/A	N/A	240 °C 100 °C	400 °C	$150 -$ 400 °C	$100 -$ 280 °C
Super- electrophilic $Pd^{2+}$	2188 2212	1855	N/A	N/A	160 °C			



Figure 5.6 0.7% Pd-CHA: DRIFTS spectra of NO desorption after pretreatment at 500 °C for 1 h with rigorously dried Ar. NO was adsorbed for 10 minutes at 25 °C before the material was purged for 10 minutes in Ar prior to ramping to remove gas-phase NO. Temperature was ramped at 10 °C/min to 500 °C with spectra being collected every 20 °C; highlighted spectra represent 25 °C (orange spectrum), and 100 °C (dark blue), 200 °C (yellow), 300 °C (light blue), 400 °C (navy), and 500 °C (black).

# 5.3 Effect of NO and NO<sub>2</sub> on Pd Speciation of Pd-BEA

Of crucial importance to the implementation of PNA materials is their longevity under exposure to exhaust components, the impact of NO and  $NO<sub>2</sub>$  on Pd speciation at high temperatures still being unclear.<sup>59</sup> These effects were explored by pretreating 1.4% Pd-BEA zeolite under NO or NO<sub>2</sub> in Ar at 500 °C for 1 h, followed by CO or NO<sub>2</sub> adsorption to probe Pd speciation. These results are presented in Figure 5.7, compared with the material pretreated in Ar at 500 °C. These data serve to reinforce the conclusion that NO generally reduces  $Pd^{n+}$  species at high temperatures. Of special note is the apparent reduction of  $[\text{Pd}^{2+}(\text{OH})]^+$  to  $\text{Pd}^+$  at high temperatures, primarily  $\text{Pd}^+$  and  $\text{Pd}^{0}$ being evident after treatment. This reaction likely proceeds similarly to the manner described by Descorme et al. as shown in Equation 9.90

$$
2(Z^{-}[Pd^{2+} - OH]^{+}) + NO \rightarrow 2(Pd^{+}) + NO_2 + H_2O \qquad (9)
$$

Similar reduction occurs when  $NO<sub>2</sub>$  is employed as the treatment gas, this being explainable by the unavoidable inclusion of trace water and NO during this treatment. The greater presence of  $Pd^0$  on the NO<sub>2</sub>-treated material can be explained by reduction of  $Pd^{n+}$  by water, while NO<sub>2</sub> also serves to maintain some ionic Pd in its oxidized state. The species that benefit from NO<sub>2</sub> oxidation are primarily  $Pd^+(2136, 2119 \text{ cm}^{-1})$  and the least electrophilic Pd<sup>2+</sup> (2154 cm<sup>-1</sup>), suggesting a process wherein Pd<sup>2+</sup> is reduced to Pd<sup>+</sup> and  $Pd<sup>0</sup>$  by NO and water respectively, while NO<sub>2</sub> serves to re-oxidize Pd metal to Pd<sup>+</sup> and oxidation of  $Pd^+$  to  $Pd^{2+}$  is less facile. This conclusion is also supported by the effect of inert pretreatment observed in microreactor and CO-DRIFTS experiments, the abundance of  $Pd^+$  apparently being greater when  $O_2$  is absent during pretreatment. A weak feature also appears at  $1790 \text{ cm}^{-1}$ , a concerning result suggesting that NO results in Pd particle agglomeration in a manner analogous to CO. This feature is not present when  $NO<sub>2</sub>$  is employed in the pretreatment, an indication that the presence of  $NO<sub>2</sub>$  may inhibit the agglomeration of large Pd particles.

The NO- and NO<sub>2</sub>-treated 1.4% Pd-BEA were also examined with NO<sub>2</sub> as the adsorbate, these results being shown in Figure 5.8. The intensity of the  $Pd^{2+}(NO)$  band remains relatively constant regardless of treatment, yet another piece of evidence that  $[{\rm Pd}^{2+}({\rm OH})]^+$  and the least electrophilic Pd<sup>2+</sup> sites are the primary Pd<sup>2+</sup>(NO)<sub>x</sub> storage sites on the BEA materials, as the relevant CO bands are the least affected by treatment. However, the intensity of these bands is still low relative to the  $Pd<sup>+</sup>(CO)$  bands as evidenced by the position of the local band maxima at frequencies more aligned with  $Pd<sup>+</sup>(CO)$ . Meanwhile, loss of the super-electrophilic  $Pd<sup>2+</sup>$  is substantial, if this species were participating in NO adsorption it would be expected that the  $Pd^{2+}(\text{NO})$  band would also decrease in intensity from the as-prepared material, though this does not occur. The  $Pd<sup>+</sup>(NO)$  band is weakest on the Ar-treated material where  $Pd<sup>+</sup>$  quantities are relatively

low and  $Pd^{2+}$  is more abundant, then increases dramatically in intensity on the NO<sub>2</sub>treated material, this being consistent with the CO adsorption results that show an increase in Pd<sup>+</sup>(CO) band intensity (2119 cm<sup>-1</sup>) after NO<sub>2</sub> treatment over the as-prepared material. The behavior of the  $NO<sup>+</sup>$  bands is also notable, these species increasing in abundance with the extent of Pd reduction in the order  $NO > NO<sub>2</sub> > Ar$ , loss of ionexchanged Pd leading to greater availability of zeolite Brønsted acid sites that facilitate formation of these species. Additionally, enhanced nitrate adsorption is observed on the NO<sub>2</sub>-treated material, evidence that NO<sub>2</sub> is formed during reduction of  $[{\rm Pd}^{2+}({\rm OH})]^+$  upon NO adsorption at 50 °C (as per eqn. 9). This is supported by the behavior of the P-HAl(OH) band (Figure S4) under these conditions; this band remains relatively constant in intensity, thus indicating the additional nitrate formation is not a result of additional P-HAl formation.



Figure 5.7 1.4% Pd-BEA: DRIFT spectra of CO adsorption after pretreatment at 500 °C for 1 h with rigorously dried Ar, 1000 ppm NO in Ar or 1000 ppm NO<sub>2</sub> in Ar. CO was adsorbed for 10 minutes at 25 °C. Top: ionic Pd range. Bottom: metallic Pd range.



Figure 5.8 1.4% Pd-BEA: DRIFT spectra of NO2 adsorption after pretreatment at 500 °C for 1 h with rigorously dried Ar, 1000 ppm NO in Ar or 1000 ppm NO<sub>2</sub> in Ar. NO<sub>2</sub> was adsorbed for 10 minutes at 50 °C.

To further explore the interaction between NO,  $NO<sub>2</sub>$ , and  $Pd<sup>+</sup>$ , 1.4% Pd-BEA was reduced in H<sub>2</sub> at 500 °C for 1 h and then exposed to NO for ten minutes, followed by NO<sub>2</sub> for ten minutes at 25, 50 and 100 °C (Figure 5.9). Regardless of temperature, only the band assigned to NO adsorbed at EFAl or Brønsted acid sites is apparent upon NO exposure, along with an extremely broad, weak feature in the  $Pd<sup>+</sup>(NO)$  range, this possibly corresponding to a water pseudo-maximum. The formation of  $NO<sup>+</sup>$  is also inhibited under these conditions, only very weak features appearing, as Pd metal particles limit diffusion to Brønsted acid sites.  $NO<sub>2</sub>$  adsorption spectra are more revealing, however,  $Pd^{n+}(NO)$  bands failing to appear at 25 °C, while at 50 °C  $Pd^{+}(NO)$  is formed preferentially, and finally at  $100\text{ °C}$   $\text{Pd}^{2+}$  and  $\text{Pd}^{+}(\text{NO})$  bands are present at nearly equal intensity. The intensity of the nitrate bands also decreases with increasing temperature, a result indicating that much of the NO<sub>2</sub> is consumed during Pd oxidation, nitrate bands not appearing until oxidation is nearly complete. The  $NO<sup>+</sup>$  bands also support this conclusion, these growing less intense as the extent of Pd reoxidation increases and more NO is stored as nitrosyl species.



Figure 5.9 1.4% Pd-BEA: Top: DRIFT spectra of NO-NO<sub>2</sub> sequential adsorption after pretreatment at 500 °C for 1 h with 10%  $H_2$ . NO was adsorbed for 10 min at 25, 50 or 100 °C followed by NO2 adsorption for 10 minutes. Bottom: Timeresolved spectra of  $NO<sub>2</sub>$  adsorption at 100 °C, the black spectrum being that collected after Ar purge for 10 min.

To provide further evidence for these inferences, the time-resolved spectrum of NO<sub>2</sub> adsorption at 100 °C shows that  $Pd^{2+}$  and  $Pd^{+}$  begin to appear simultaneously, the  $Pd^{2+}(\text{NO})$  band achieving maximum intensity at 9 minutes, while  $Pd^{+}(\text{NO})$  continues to form. Nitrate bands begin to appear at 10 minutes. The spectrum collected prior to

desorption is also included here, this being collected after a 10 min Ar purge to remove gas-phase species before desorption. This spectrum reveals that  $Pd<sup>+</sup>(NO)$  continues to form, as do nitrate features, even as the concentration of  $NO<sub>2</sub>$  in the feed decreased as the adsorbate gas was cleared from the cell. Figure S5 shows the OH region during this experiment, these results showing no strong interaction with the P-HAl(OH) band until nitrates begin to form.

To conclude this section, the reactivity of NO and  $NO<sub>2</sub>$  with Pd species on BEA zeolite have been explored. NO apparently has a substantially reducing effect on ionexchanged Pd at 500 °C. The consistent intensity of the  $Pd^{2+}(\text{NO})$  IR band under differing treatment gases suggests that not all  $Pd^{2+}$  species participate in  $NO_x$  adsorption in the case of BEA, otherwise the loss of super-electrophilic  $Pd^{2+}$  would be evident as a decrease in the intensity of the  $Pd^{2+}(NO)$  band. Meanwhile, the intensity of the  $Pd^{+}(CO)$  and  $Pd^{+}(NO)$ IR bands increases after pretreatment in  $NO$  and  $NO<sub>2</sub>$ , the  $NO<sub>2</sub>$ -treated material showing a greater preference for  $Pd^+$  formation than even the material pretreated in Ar. This indicates that NO<sub>2</sub> serves to re-oxidize Pd metal to Pd<sup>+</sup> at 50 °C, while reoxidation to  $Pd^{2+}$  is less extensive. The effect of NO and NO<sub>2</sub> on the H<sub>2</sub>-reduced 1.4% Pd-BEA was examined at various temperatures, NO adsorption failing to re-oxidize Pd to form nitrosyl bands at any of the evaluated temperatures, indicating that NO does not have an oxidizing effect on Pd. NO<sub>2</sub> adsorption at 25  $\degree$ C similarly fails to produce nitrosyl bands, while at 50 °C Pd<sup>+</sup>(NO) species are formed preferentially and at 100 °C additional Pd<sup>2+</sup>(NO) species form. Nitrate bands can be used to gauge the extent of re-oxidation as their appearance coincides with the endpoint of  $NO<sub>2</sub>$  consumption by Pd oxidation, these data also serving to establish temporal coordination between the appearance of nitrates and disappearance of the P-HAl(OH) band.

## 5.4 Effect of NO and NO<sub>2</sub> on Pd Speciation of Pd-CHA

Similar experiments to those presented in the previous section were performed on  $0.7\%$  Pd-CHA to assess the impact of NO and NO<sub>2</sub> on the Pd speciation of this material. Pd speciation after treatment was initially examined by CO adsorption, the results presented in Figure 5.10. In this case, the Pd species primarily affected by NO treatment

is the super-electrophilic  $Pd^{2+}$ , these CO bands being greatly reduced in intensity upon treatment at 500 °C in both NO and NO<sub>2</sub>. Notably, the  $[{\rm Pd}^{2+}({\rm OH})]^+$  bands are unaffected in either gas, while  $Pd^+$  and the less electrophilic  $Pd^{2+}(CO)$  bands increase in intensity after treatment. These observations suggest that  $[{\rm Pd}^2^+({\rm OH})]^+$  is more stable on Pd-CHA than on Pd-BEA. Examination of the metallic Pd(CO) region again reveals the presence of the band at 1790 cm–1 after NO treatment, corresponding to a site involved in the agglomeration of large Pd particles. Meanwhile this band is absent on the  $NO<sub>2</sub>$ -treated material as observed on Pd-BEA. The increased presence of pore-confined Pd particles is also observed after treatment in either gas, further demonstrating the reducing effect of NO. Aside from the absence of the band at  $1790 \text{ cm}^{-1}$ , the NO- and NO<sub>2</sub>-treated CHA material show remarkably similar CO adsorption spectra, indicating the oxidizing effect of NO2 is less pronounced than observed on Pd-BEA.



Figure 5.10 0.7% Pd-CHA: DRIFT spectra of CO adsorption after pretreatment at 500 °C for 1 h with rigorously dried Ar, 1000 ppm NO in Ar or 1000 ppm NO<sub>2</sub> in Ar. CO was adsorbed for 10 minutes at 25 °C. Top: ionic Pd range. Bottom: metallic Pd range.



Figure 5.11 0.7% Pd-CHA: DRIFT spectra of  $NO<sub>2</sub>$  adsorption after pretreatment at 500 °C for 1 h with rigorously dried Ar, 1000 ppm NO in Ar or 1000 ppm  $NO<sub>2</sub>$ in Ar. NO<sub>2</sub> was adsorbed for 10 minutes at 50 °C.

Given the similarity of the CO adsorption spectra of the Pd-CHA material, it could also be expected to exhibit similar  $NO<sub>2</sub>$  adsorption spectra; however, this is not the case. Figure 5.11 shows  $NO<sub>2</sub>$  adsorption after pretreatment in NO or  $NO<sub>2</sub>$ , these results again being compared to the Ar-pretreated material. Unlike the CO adsorption spectra, NO treatment leads to little change in the  $NO<sub>2</sub>$  adsorption spectrum, nitrosyl bands being of similar intensity to those observed after Ar treatment, a result that suggests much of the super-electrophilic  $Pd^{2+}$  (nearly eliminated in CO DRIFTS after NO treatment) is converted to less electrophilic  $Pd^{2+}$  and  $Pd^+$ , these species still participating in NO<sub>x</sub> adsorption. Indications of Pd reduction can still be observed in the increased intensity of  $NO<sup>+</sup>$  bands, again indicating greater availability of Brønsted acid sites after NO and  $NO<sub>2</sub>$ treatment at 500 °C. However, the  $NO<sub>2</sub>$ -pretreated material shows a dramatically different spectrum, a new nitrosyl band appearing at  $1825 \text{ cm}^{-1}$  in conjunction with a nitrate band at 1627 cm<sup>-1</sup>. This result is surprising given the similarity of the CO adsorption spectra and suggests that  $PdO_x$  may be involved in the observed changes, as  $PdO_x$  does not readily adsorb CO. A further indication of the involvement of  $PdO<sub>x</sub>$  is the lack of a corresponding increase in the intensity of the P-HAl(OH) band in the presence of nitrates, suggesting that P-HAl is not the adsorption site playing host to nitrates under this

circumstance (Figure S6). Further analysis will be required to establish the true nature of this site, including examination of a CHA material with higher initial Pd dispersion. These data also suggest an alternative explanation for the observed changes in the CHA  $NO<sub>x</sub>$  desorption behavior upon reoxidation from  $H<sub>2</sub>$  reduction, the formation of nitratesupporting sites leading to the observed increase in  $NO<sub>x</sub>$  desorption temperature, an inference supported by the similar desorption temperatures of this event and the desorption of nitrates from BEA. However, as with the absence of low-temperature  $NO<sub>x</sub>$ desorption in that case, the absence of the typical  $Pd^{n+}(NO)$  bands in this case is not readily explained, formation of high abundances of  $Pd<sup>+</sup>$  being yet another alternative explanation for these results.

## 5.5 Conclusions

To summarize the results of this chapter, microreactor-MS studies of  $NO<sub>x</sub>$ desorption were carried out with and without oxygen to facilitate association of  $NO<sub>x</sub>$ desorption events in both MS and DRIFTS studies. These experiments show that the presence of  $O_2$  is important in determining the Pd speciation during treatment by maintaining  $Pd^{2+}$  in ionic form. Meanwhile, the adsorbate speciation during adsorption is also shown to be affected by the presence of  $O_2$  while  $O_2$  also serves to lower the desorption temperature of most species during the desorption step. The presence of  $O<sub>2</sub>$ during pretreatment of Pd-BEA is shown to decrease  $NO<sub>x</sub>$  storage due to the formation of  $Pd^{2+}$  species that don't participate in NO<sub>x</sub> adsorption, while Pd-CHA shows an increase in  $NO<sub>x</sub>$  storage after air treatment due to the participation of additional types of  $Pd<sup>2+</sup>$  on this material. The presence of  $O_2$  during adsorption facilitates formation of  $NO_2$ , this being stored as nitrates at P-HAl sites, while the presence of  $O<sub>2</sub>$  during desorption generally decreases the desorption temperature of all species. By examining the effect of Pdloading on  $NO<sub>x</sub>$  desorption, inferences about the Pd-related  $NO<sub>x</sub>$  desorption features can be derived. As Pd-loading increases, the desorption event at 300 °C gradually disappears, an indication that this feature is representative of nitrates adsorbed at P-HAl. The shift of this event to lower temperatures is also indicative of interaction of these sites with Pd. This supposition is further corroborated by DRIFTS  $NO<sub>2</sub>-TPD$ , these spectra showing a decrease in the nitrate desorption endpoint temperature as Pd-loading increases.

Conversely, the Pd-CHA material shows no evidence of a desorption event at 300 °C, in agreement with its weak expression of the P-HAl(OH) band. DRIFTS NO-TPD under rigorously dried conditions was employed to assess the desorption of nitrosyl and  $NO<sup>+</sup>$ species, both  $NO^+$  and  $Pd^+(NO)$  bands remaining to 500 °C. Previous studies have shown lower desorption temperatures for  $NO^+$ , though these species have also been shown to be highly sensitive to the presence of water. Therefore, the higher desorption temperature of these species can be inferred to be a result of the rigorous removal of trace water during these experiments.<sup>95, 102</sup> From these data, it can be inferred that most  $Pd^{2+}(\text{NO})$  desorbs below 300 °C, the maximum desorption occurring at  $\sim$ 200 °C. NO is proposed to desorb from Pd<sup>+</sup> in two discreet events, one in a similar range to  $Pd^{2+}(NO)$  while the other desorbs at ~450 °C, this result supported by the changing frequency of the relevant  $Pd<sup>+</sup>(NO)$  IR band as desorption proceeds. The CHA material exhibits similar behavior aside from the absence of nitrates and P-HAl, however desorption from the various  $Pd^{2+}$ species cannot be easily differentiated. In fact, most of these desorption events overlap to at least some degree, thus limiting these studies to qualitative assessment.

The effect of NO and  $NO<sub>2</sub>$  at high temperature on these materials was also examined, NO being demonstrated to generally reduce ion-exchanged Pd on BEA, while only super-electrophilic  $Pd^{2+}$  is reduced on Pd-CHA. NO<sub>2</sub> treatment leads to comparatively better retention of ionic Pd, especially  $Pd^+$  and  $[Pd^{2+}(OH)]^+$ . A concerning result is the appearance of the  $(Pd^0)_{3}(CO)$  band at 1790 cm<sup>-1</sup> after NO treatment on both materials, this species being associated with the agglomeration of large Pd particles. This band is absent after  $NO<sub>2</sub>$  treatment, suggesting that  $NO<sub>2</sub>$  may inhibit the formation of this CO adsorption site. Adsorption of  $NO<sub>2</sub>$  after NO or  $NO<sub>2</sub>$  pretreatment was also conducted,  $NO<sub>2</sub>$  pretreatment leading to an increase in the intensity of the  $Pd<sup>+</sup>(NO)$  band on Pd-BEA, while  $NO<sub>2</sub>$  pretreatment on Pd-CHA leads to the appearance of a nitrate band and a single nitrosyl species, an unexpected result given the similarity of the CO adsorption spectra and an indication that the  $NO<sub>x</sub>$  adsorption behavior is substantially altered by treatment in  $NO<sub>2</sub>$ .

## CHAPTER 6. CONCLUSIONS

In this dissertation, two Pd-loaded zeolite materials were assessed for their potential to serve as PNA's. Generally, the addition of Pd to these materials is shown to enhance the  $NO<sub>x</sub>$  adsorption capacity while also leading to higher  $NO<sub>x</sub>$  desorption temperatures. The effects of Pd-loading on the Pd speciation of BEA were also assessed, these results allowing for the development of a self-consistent set of FT-IR band assignments that could be correlated with microreactor MS-desorption events. The most consequential results of these comparisons are the identification of Pd and zeolite sites that actively participate in NO<sub>x</sub> storage and desorb NO<sub>x</sub> near or above 200 °C. For BEA, these NO<sub>x</sub> adsorption sites consist of Pd<sup>+</sup>,  $[{\rm Pd}^{2+}({\rm OH})]$ <sup>+</sup>, P-HAl, and possibly other Pd<sup>2+</sup> species, though the super-electrophilic  $Pd^{2+}$  formed on this material is unlikely to adsorb  $NO<sub>x</sub>$  given its sensitivity to reduction by water that will be unavoidably present under realistic conditions. Further, increasing Pd loading is shown to decrease the desorption temperature of nitrate species, this result, along with an abundance of other evidence, indicating the interaction of Pd with P-HAl sites. Pd-CHA shows differing behavior, the absence of P-HAl and nitrates leading to a lower average  $NO<sub>x</sub>$  desorption temperature from this material, while all or nearly all the  $Pd^{2+}$  species evidently participate in NO<sub>x</sub> storage. The highest temperature desorption event on both CHA and BEA occurs between 400 and 550  $\degree$ C and can be tied to Pd<sup>+</sup>(NO), the abundance of this species being favored when the materials are pretreated under inert conditions.

The stability of the Pd speciation of both materials was also examined, BEA being found to be more sensitive to Pd reduction in all respects due to its larger pore size. CO and NO are shown to be especially problematic for the retention of ion-exchanged Pd, exposure of both materials to these gases at high temperature leading to formation of large Pd particles outside the zeolite framework that are more susceptible to sintering. This result suggests these materials will not be able to achieve sufficient lifetime for realworld applications on stoichiometric automotive systems, whereas the positive effects of  $O_2$  and  $NO_2$  on Pd speciation and  $NO_x$  adsorption suggest these materials may be applicable to lean diesel systems. If a DOC were placed upstream of the PNA, this would

eliminate all the NO and CO in favor of  $NO_2$  and  $CO_2$ , these gases being more benign in their effects on Pd speciation. Both materials have advantages and disadvantages: Pd-CHA exhibits greater  $NO<sub>x</sub>$  storage in the absence of water, but BEA shows better performance than CHA when water is present. However, the BEA material will likely suffer more rapid deactivation and has lower hydrothermal stability than the CHA material; therefore, CHA is still more likely to be applicable under realistic conditions. The dramatically different behavior of these materials despite their similar Si/Al ratios serves to highlight the fact that framework structure and pore diameter are key parameters in the control of Pd-speciation and thus  $NO<sub>x</sub>$  adsorption. This also indicates further improvement of the PNA behavior of Pd zeolites is likely possible through the control of zeolite Al-O(H)-Si site proximity, the preferential formation of particular  $Pd^{n+}$ species being possible under those circumstances. While this work serves to highlight many problematic aspects of long-term PNA stability, identifying these issues is the first step in resolving them and further developments may still reveal a viable PNA material. Even if such a system only achieves limited implementation it could still lead to the elimination of millions of tons of  $NO<sub>x</sub>$  emissions each year, and thus the potential impact of this work cannot be understated.

## APPENDICES

#### APPENDIX 1. EXPERIMENTAL PROCEDURES

#### A 1.1 In-Situ DRIFTS

In-situ DRIFTS studies were carried out in a Harrick Scientific Praying Mantis DRIFTS cell equipped with a high-temperature reaction chamber. In each case a gas flow rate of 50 sccm was employed, and approximately 50 mg of catalyst material was loaded into the sample cup of the DRIFTS cell such that gas must flow through the catalyst bed. This system was also equipped with a custom-built feed-bypass manifold to allow for simultaneous blending of up to eight gases, as well as the rapid switching between different gas feeds. Gas flow control is achieved with mass flow controllers operated with Brooks instrument control interfaces, the gas flow being calibrated periodically to achieve precise gas blend concentrations. To purify feed gases of trace species such as water and NO<sub>2</sub>, a dry ice-acetone cold trap was employed. This device consisted of a long copper coil packed with glass beads and submerged in the dry ice-acetone mixture contained in a Dewar. A bypass of the trap was also added so that higher boiling adsorbates and treatment gases could be employed at will and each loop of the manifold was equipped with a backpressure valve to maintain a consistent positive pressure of 1 psi throughout the system. The IR spectrometer used for these experiments was a Nicolet 6700 FT-IR equipped with a liquid nitrogen-cooled MCT-A detector. The FT-IR collection parameters were kept constant across all experiments, 116 scans being collected for each spectrum at a resolution of  $4 \text{ cm}^{-1}$ . The collection time for each spectrum was approximately 1 minute, these parameters allowing sufficient temporal resolution to examine gas adsorption over time.

Each DRIFTS experiment consisted of three phases: pretreatment, adsorption, and desorption steps. In each case the pretreatment was carried out at 500 °C for 1 h to remove adsorbed water from the zeolite, this being necessary due to the inhibiting effects of water on the adsorption of other species. If a strongly adsorbing species such as  $NO<sub>2</sub>$ or NH3 was employed during pretreatment, the treatment gas was switched off after 1h and the temperature was increased to 550 °C for 10 min to achieve desorption of these

species. The final part of the pretreatment is a period of cooling to the desired adsorption temperature, background FT-IR spectra being collected every 20 °C to compensate for thermal effects on the zeolite FT-IR spectrum during desorption. During cooling, the adsorption gas was prepared on the manifold bypass loop to achieve stable concentration of the adsorbate prior to the beginning of the adsorption step. Adsorption was then carried out by switching the bypass and reactor gas feeds, and the cold trap was bypassed if necessary. Adsorption was carried out at a constant temperature for 10 min followed by a switch back to the gas employed during cooling. This was followed by a 10 min purge to remove any remaining adsorption gas from the system prior to desorption. Desorption was achieved by ramping the temperature of the cell from the adsorption temperature to 500 °C at a rate of approximately 10 °C per min. Spectra were collected every 20 °C at the same temperatures as those collected during cooling.

## A 1.2 Microreactor-MS

Microreactor-MS studies were carried out using a Micromeritics Autochem-II microreactor system equipped with a Thermo-QMS 200 MS to measure gas concentrations at the outlet. This system employs a U-shaped quartz tube as a microreactor, supported inside a furnace. Approximately 0.1 g of material was supported on quartz wool inside the reactor tube such that gas must flow through the material. This system requires a flow rate of 100 sccm.  $NO<sub>x</sub>$  concentrations were calibrated by measuring the background response of m/z 30 (NO) and the response at a NO concentration of 1000 ppm. The steps of these experiments are like those employed in DRIFTS studies, the primary difference being the need to purge the material for 1 h after adsorption to remove weakly adsorbed species. Otherwise, these experiments were intentionally conducted in as similar a manner as possible to the DRIFTS experiments to facilitate comparison between the two.

## A 1.3 Atomic Absorption Spectroscopy

A Perkin Elmer AAnalyst 300 AAS system was employed for this analysis, these experiments being conducted by Mr. Trevor Lardinois at Purdue University. Prior to analysis, 20-50 mg of the Pd-loaded zeolite materials were digested with 2.5 g of HF (48 wt%, Sigma Aldrich) followed by dilution with 50 g 18.2 MΩ de-ionized water.

Following analysis, the Si/Al ratio was estimated using a molar composition of  $Si<sub>(1)</sub>$  $x)O_2Al_x$ .

## $A$  1.4  $H_2$ -TPR

H2-TPR was carried out using a Micromeritics 2920 chemisorption analyzer equipped with a Cryocooler II accessory for low-temperature analysis. Materials were pretreated in flowing air (30 sccm) to 200-550 °C for 1 h, then cooled in flowing Ar to – 70 °C, then switched to a flow of 5.01%  $H_2$  in Ar (10 sccm) A cold trap was used to remove water from the gas prior to detection with a thermal conductivity detector. After baseline stabilization ( $\sim 30$  min), the sample was heated to 300 °C followed by a 30 min isothermal hold at 300 °C. TCD signals were calibrated similarly to the MS signals.

#### A 1.5 In-situ near-ambient pressure XPS

These experiments were performed by Dr. Olivier Heintz at the University of Burgundy, France. A PHI 5000 Versaprobe XPS system was employed for this study, this system equipped with a monochromatic Al K 1 X-Ray source with an energy of 1486.6 eV, spot diameter of 200  $\mu$ m, acceleration voltage of 15 kV and power of 50 W. Highresolution scans were collected in the Pd 3d region with a pass energy of 58.7 eV. The Al 2p signal at 74.4 eV was used for energy calibration, measurements being performed with a neutralization system. CasaXPS was used for data processing and Landau ionization cross-sections were employed to quantify the semi-empirical relative sensitivity factors. The pretreatment chamber and Steatite sample holders employed were constructed inhouse. Pretreatment was carried out in Ar, air or  $10\%$  H<sub>2</sub>, at either 500 or 750 °C for 1 h, pretreatment being conducted such that no exposure to atmosphere occurred prior to analysis.

## APPENDIX 2. MATERIAL SYNTHESIS AND CHARACTERIZATION

Zeolite synthesis and preparation was performed by Mr. Trevor Lardinois at Purdue University. The NH4-BEA zeolite was obtained from Zeolyst International  $(CP814E, Si/A1 = 12.5)$ . The CHA zeolite was hydrothermally synthesized according to previously published procedures (SSZ-13,  $Si/Al<sub>gel</sub> = 15$ ). In summary, the reagents used in these preparations were Ludox HS-40 (40 wt%, Sigma Aldrich), Al(OH)<sub>3</sub> (98 wt%, SPI Pharma), trimethyladamantylammonium hydroxide (TMAdaOH, 25 wt%, Sachem), NaOH pellets (98 wt%, Alfa Aesar), and deionized water (18.2 M $\Omega$ ) added in the molar ratios 1  $SiO_2/0.0667$  Al/0.25 TMAdaOH/0.25 Na/44 H<sub>2</sub>O to form the precursor gel. In a PFA container with a Teflon(CO)ated magnetic stir bar, TMAdaOH and water were mixed for 15 min prior to the addition of  $Al(OH)$ <sub>3</sub> and NaOH pellets. Upon dissolution of the solids, the Si precursor was added and the mixture was stirred for an additional 2 h. The synthesis gel was then transferred to sealed, Teflon-lined stainless-steel autoclaves (Parr Instrument Company) and placed in a convection oven at 160 °C with daily quenching and stirring (5 min).

The silicious zeolite BEA was synthesized according to literature procedures with molar gel ratios of 0.54 tetraethylammonium hydroxide (TEAOH, 40 wt%, Sigma Aldrich), 1.0 tetraethyl orthosilicate (TEOS, 98 wt%, Sigma Aldrich), 7.25 deionized water (18.2 M $\Omega$ ), and 0.54 hydrofluoric acid (48 wt%, Sigma Aldrich). TEAOH and TEOS were combined in excess water contained in a PFA vessel with Teflon stir bar. The resulting mixture was stirred under flowing air to evaporate EtOH, additional water added as necessary to achieve the final molar ratio. HF was then added dropwise with intermittent stirring, and then loaded into Teflon-lined stainless-steel autoclaves and heated in an oven to 140 °C.

After 7 days, all the zeolite materials were recovered using a centrifuge and washed a total of seven times alternating between water and acetone (99.9 wt%, Sigma-Aldrich), water comprising the last wash. The resulting solids were dried in a static oven at 100 °C prior to treatment in flowing air (100 cm<sup>3</sup> min<sup>-1</sup>, Air Zero, Indiana Oxygen) to 580 °C with a ramp rate of 1 °C per min for 10 h. At this point, powder XRD was performed with a Rigaku Smart-Lab X-ray diffractometer (XRD) using a Cu K $\alpha$  ( $\lambda$  =

0.154 nm) radiation source, these results shown in Figure A2.1. The resulting diffraction patterns serve to confirm the framework structure of each batch of the various materials prior to combination. The CHA material was ion-exchanged into the NH4-form through ion-exchange with 1 M NH<sub>4</sub>NO<sub>3</sub> at a ratio of 150 g of solution per g of solid under ambient conditions for at least 24 h. The solutions were centrifuged and washed with water 4 times, followed by centrifugation to recover the solids. The recovered solids were then dried for at least 12 h at 100 °C. N<sub>2</sub> physisorption was performed with a Micromeritics ASAP 2020 Surface Area and Porosity Analyzer, the results being presented in Figure A2.2. The pore volumes were found to be consistent with those observed in literature reports.



Figure A2.1. Ex-situ X-ray diffraction (XRD) patterns normalized to maximum intensity. (Left) XRD of Si-Beta (gray, bottom), NH4-Beta (black, offset +1 a.u.), and 1.36 wt% Pd-Beta (gray, offset +2 a.u.). Instrument artifact (labeled) is from sample holder. (Right) XRD of a reference Si-CHA (black, bottom), 8 independent CHA syntheses  $(Si/Algel = 15)$  of similar zeolite gel composition before mixing into a larger batch (gray for synthesis aliquots, black for CHA samples that were washed and airtreated to 580 °C for 10 hours, offset range  $+1$  to  $+8$  a.u.), and a 0.7 wt% Pd-CHA sample (red, offset  $+9$  a.u.).



Figure A2.2 N<sub>2</sub> (-196 °C) adsorption isotherm for H-form Beta (black circles) and Ar (-186 °C) adsorption isotherm for 0.7 wt% Pd-CHA (gray circles, offset +200 cm<sup>3</sup> g<sup>-1</sup> at STP).

Pd-loaded materials were prepared via incipient wetness impregnation. A dilute aqueous  $Pd(NH_3)_{4}(NO_3)_{2}$  (10 wt% solution, Sigma Aldrich) solution was added dropwise to the NH4-form material while stirring, until adequate solution was added to completely fill the zeolite pore volume. The resulting solids were dried in static air at 120 °C for at least 12 h before treatment in flowing air at 100 sccm to 550 °C at a ramp rate of 2 °C per min. XRD was again used to confirm the framework topology of the Pd exchanged materials, indicating no significant structural changes occurred during impregnation, these results also being shown in Figure A2.1.

To establish the extent of Pd ion-exchange,  $H_2$ -TPR was employed as described above. Figure A2.3 shows the results of these experiments on both the Pd BEA and CHA materials, the integrated area of  $H_2$  consumption events being used to calculate the  $Pd_{iso}/Pd_{tot}$  ratio. This is possible due to the differing reduction temperatures of  $PdO_x$ particles and ion-exchanged Pd, these species reducing at  $-10-35$  °C and 35–200 °C, respectively. This calculation assumes a 1:1  $H_2$ : $Pd^{2+}$  stoichiometry to quantify the number of divalent Pd cations in each reduction event. Apparent H<sub>2</sub> consumption is also observed at –40 °C, though in reality this feature corresponds to desorption of Ar from the framework pores, this feature also being observed on the H-form materials.



Figure A2.3. H2 temperature-programmed reduction profiles of BEA (top) and CHA (bottom) zeolites pretreated in flowing air to 550 °C, unless otherwise indicated.


Figure S1. Deconvolution of in-situ XPS Pd 3d spectra for 1.4% Pd-BEA pretreated at 500 °C for 1 h in H<sub>2</sub> (top), pretreated at 500 °C in Ar for 1 h (2<sup>nd</sup>), pretreated at 750 °C in air for 1 h (3<sup>rd</sup>), pretreatment in air at 500 °C for 1 h (4<sup>th</sup>), material as-prepared (bottom).



Figure S2. H-BEA: DRIFT spectra of the NH<sub>4</sub><sup>+</sup> region after pretreatment at 500 °C for 1 h with Ar rigorously dried by cold trap. The material was cooled to 25 °C prior to the adsorption of 1% NH3 for ten minutes.



Figure S3. H-BEA: DRIFT spectrum of the carbonate region after pretreatment at 500 °C for 1 h with Ar rigorously dried by cold trap. The material was cooled to 25 °C prior to the adsorption of  $1\%$  CO<sub>2</sub> for ten minutes.



Figure S4 1.4% Pd-BEA: DRIFT spectra of NO<sub>2</sub> adsorption after pretreatment at 500 °C for 1 h with rigorously dried Ar, 1000 ppm NO in Ar or 1000 ppm  $NO<sub>2</sub>$  in Ar. NO<sub>2</sub> was adsorbed for 10 minutes at 50 °C.



Figure S5. DRIFTS spectra of NO-NO2 sequential adsorption after pretreatment at 500 °C for 1 h with 10% H<sub>2</sub>. NO was adsorbed for 10 minutes at 100 °C followed by NO<sub>2</sub> adsorption for 10 minutes. Time-resolved spectra of  $NO<sub>2</sub>$  adsorption at 100 °C are shown, the black spectrum being that collected after Ar purge for 10 min.



Figure S6. 0.7% Pd-CHA: DRIFT spectra of NO<sub>2</sub> adsorption after pretreatment at 500 °C for 1 h with rigorously dried Ar, 1000 ppm NO in Ar or 1000 ppm NO2 in Ar. NO2 was adsorbed for 10 minutes at 50 °C.

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### VITA

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## EDUCATION

B.S. CHEMISTRY • MAY 2011 • EASTERN KENTUCKY UNIVERSITY Richmond, Kentucky. American Chemical Society Certified Degree.

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# PROFESSIONAL POSITIONS HELD

ASSISTANT RESEARCH SCIENTIST • UNIVERSITY OF KENTUCKY CENTER FOR APPLIED ENERGY RESEARCH • 2013– 2016

TECHNICIAN • UNIVERSITY OF KENTUCKY CENTER FOR APPLIED ENERGY RESEARCH • 2011 – 2013

# **HONORS**

2018 University of Kentucky College of Arts and Sciences Outstanding Teaching Assistant Award

2018 National Science Foundation Graduate Research Fellowship Honorable Mention

## PUBLICATIONS

1. Mazzotta, M. G.; Pace, R. B.; Wallgren, B. N.; Morton, S. A.; Miller, K. M.; Smith, D. L., Direct Analysis in Real Time Mass Spectrometry (DART-MS) of Ionic Liquids. *J. Am. Soc. Mass Spectrom.* **2013**, *24*, 1616-1619.

2. Harman-Ware, A.; Crocker, M.; Pace, R.; Placido, A.; Morton, S., III; DeBolt, S., Characterization of Endocarp Biomass and Extracted Lignin Using Pyrolysis and Spectroscopic Methods. *Bioenergy Res.* **2014**, *8*, 1-19.

3. Santillan-Jimenez, E.; Perdu, M.; Pace, R.; Morgan, T.; Crocker, M., Activated Carbon, Carbon Nanofiber and Carbon Nanotube Supported Molybdenum Carbide Catalysts for the Hydrodeoxygenation of Guaiacol. *Catalysts* **2015**, *5*, 424.

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5. Santillan-Jimenez, E.; Pace, R.; Marques, S.; Morgan, T.; McKelphin, C.; Mobley, J.; Crocker, M., Extraction, characterization, purification, and catalytic upgrading of algae lipids to fuel-like hydrocarbons. *Fuel* **2016**, *180*, 668-678.

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10. Mohler, D.; Wilson, M. H.; Kesner, S.; Schambach, J. Y.; Vaughan, D.; Frazar, M.; Stewart, J.; Groppo, J.; Pace, R.; Crocker, M. Beneficial re-use of industrial  $CO<sub>2</sub>$ emissions using microalgae: Demonstration assessment and biomass characterization. *Biores. Technol.* **2019**, 293, 122014.

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